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# Generating ideal synthetic data for 3D reconstruction of FIB tomography data using generative adversarial networks

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## Abstract

1 Accurate 3D reconstruction of nanomaterials is essential for studying their physical  
2 properties. Focused Ion Beam (FIB) tomography is a preferred method for  
3 creating 3D image stacks of micrometer-sized material volumes at nanometer resolu-  
4 tion. To achieve valid 3D reconstructions, it is crucial to segment these images  
5 using machine learning-based methods, as they help mitigate artifacts in the data.  
6 However, supervised machine learning requires a large amount of training data  
7 and ground truth, which is challenging because FIB tomography is a destructive  
8 technique. While training machine learning models on synthetic data and applying  
9 this to real data is possible, it is only partially accurate due to differences in data  
10 distributions. Moreover, generating synthetic training data is time-consuming, even  
11 with modern computing, because of the complex physical Monte Carlo modeling.  
12 This study proposes a machine learning pipeline that reduces the difference in FIB  
13 tomography data distribution using domain adaptation techniques and introduces a  
14 novel method for quickly generating synthetic data by considering physical effects  
15 without Monte Carlo simulations.

## 16 1 Introduction

17 Nanoporous materials have significant potential in fields like materials science and biochemistry  
18 due to their unique properties. To understand these properties, accurate 3D reconstruction of their  
19 structure is often required. When studying nanoporous materials, such as hierarchical nanoporous  
20 gold (HNPG), electron microscopy (EM) is one of the few imaging methods that can provide the  
21 necessary resolution, as pore sizes can be smaller than 20 nm. Focused ion beam (FIB) combined  
22 with a scanning electron microscope (SEM) allows for high-resolution volumetric data collection of  
23 such nanomaterials, with in-plane (xy) resolution of 1 nm and depth resolution of 10 nm or less. This  
24 is achieved by removing material and imaging the newly exposed cross-sections consecutively (1).  
25 However, because FIB tomography is a destructive technique, obtaining ground-truth values for the  
26 structure being studied is impossible.

27 Additionally, these high-resolution image stacks often contain artifacts, such as the *shine-through*  
28 *effect* and intensity ambiguities (2). The *shine-through effect* occurs when structures from deeper  
29 layers become visible in the current milling plane, introducing extra information to the images. As a  
30 result, it is very challenging to uniquely map intensity and structural information in gray-scale FIB  
31 tomography images.

32 These challenges make it hard to semantically segment FIB tomography images using traditional  
33 methods like thresholding or k-means clustering, which rely mainly on intensity values. However,  
34 as demonstrated in (3), combining FIB tomography with machine learning can lead to accurate 3D

35 reconstructions of nanomaterials. In their study, they trained machine learning models on synthetic  
36 data and then applied this knowledge to extract structural information from real FIB tomography  
37 data.

38 Training deep learning models with synthetic data is especially valuable in electron microscopy,  
39 where acquiring real data is difficult and expensive. In FIB tomography, the sample is destroyed,  
40 making it impossible to obtain ground-truth data. Synthetic data, however, can be easily reproduced  
41 and often include ground truth values, making them highly useful. Researchers have successfully used  
42 synthetic data in various fields, such as training machine learning models for autonomous vehicles  
43 to handle rare scenarios (4; 5). In electron microscopy, (6) generated simulated images of basic  
44 geometries for training purposes.

45 However, generating synthetic electron microscopy data presents challenges. It is time-consuming and  
46 resource-intensive, and creating realistic synthetic data requires comprehensive physical knowledge.  
47 The best synthetic data in electron microscopy are generated using Monte Carlo simulations, as  
48 suggested by (7). However, simulating 512 slices of 512 x 512 pixels using the Monte Carlo plugin in  
49 Dragonfly software (8) can take time in days on a high-performance CPU due to the need to calculate  
50 each electron trajectory.

51 Monte Carlo simulations are generally slow because they require iterative calculations for each event.  
52 Recently, generative adversarial networks (GANs) have shown promise in generating these events  
53 more efficiently. For instance, (9) demonstrated the use of GANs in particle physics, while (10)  
54 suggested a GAN-based approach for simulating electron-proton scattering events. Additionally,  
55 (11) proposed a 3D GAN based on StyleGAN2 (12) to generate realistic MR images. In electron  
56 microscopy, (13) utilized CycleGAN to create realistic scanning transmission electron images.  
57 However, to effectively replace Monte Carlo simulations, which incorporate physics-based knowledge,  
58 generative networks must also be provided with the necessary information to accurately simulate data  
59 using machine learning.

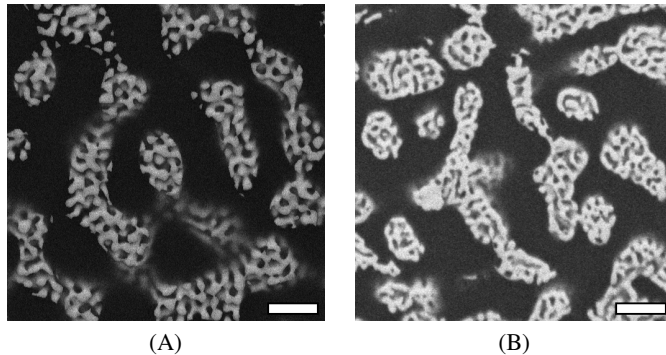


Figure 1: Slice of a (A) synthetic hierarchical nanoporous gold (HNPg) generated using MCXray plugin and (B) real HNPg structure - scale bar: 200 nm

60 When generating synthetic data, it is not always guaranteed that it will match the distribution of real  
61 data. This mismatch often occurs because synthetic data typically follow strict physical formulas,  
62 making them too *ideal*. As a result, they lack the randomness and unique characteristics found  
63 in real-world data (see Figure 1). Due to these differences, models trained on synthetic data may  
64 underperform when applied to real data (14; 15).

65 In this study, we make two key contributions. First, we introduce a synthetic data pipeline that  
66 replaces the time-intensive Monte Carlo simulations with generative networks. This pipeline includes  
67 an additional artificial neural network (ANN) block designed to incorporate critical characteristics,  
68 such as realistic structural details and the physical relationships between variables like voltage,  
69 penetration depth, and atomic number. Second, we enhance the semantic segmentation of hierarchical  
70 nanoporous gold (HNPg) by applying domain adaptation techniques using generative networks.  
71 Our results demonstrate that our novel machine learning-based synthetic data pipeline performs on  
72 par with Monte Carlo-based simulation methods. Additionally, by applying domain adaptation to  
73 synthetic data before training, we achieve, on average, a 20% improvement in 3D reconstruction  
74 accuracy for semantic segmentation tasks.

## 75 **2 Method**

### 76 **2.1 Acquiring Imaging Data**

77 In this study, we analyzed real HNPG samples and generated synthetic samples using various methods.

#### 78 **2.1.1 Synthetic Samples**

79 We began by generating binarized structures using the leveled wave method (LWM) (16), as described  
80 in (17). These binarized structures served as the first step in our synthetic FIB tomography data  
81 generation pipeline and provided the ground-truth values for our synthetic data.

82 Next, we used the Monte Carlo plugin in Dragonfly software (8) to generate *realistic* synthetic FIB  
83 tomography images, which incorporate *nearly* all relevant electron microscope physics (7). We  
84 created three synthetic datasets with different voltages (1kV, 2kV, and 4kV) using the same initial  
85 structure. These multi-voltage datasets (sMC-1kV, sMC-2kV, and sMC-4kV) are suitable for training  
86 multimodal machine learning models. Additionally, we generated three more synthetic datasets (sML-  
87 1kV, sML-2kV, and sML-4kV) using the same binarized structure but with the machine learning  
88 method outlined in Section 2.3 instead of Monte Carlo simulations. To evaluate the impact of domain  
89 adaptation, we also created domain-adapted versions of the Monte Carlo datasets (sMCDA-1kV,  
90 sMCDA-2kV, and sMCDA-4kV).

91 Furthermore, we prepared a dataset, sMC-BS, using Monte Carlo simulations to capture *realistic*  
92 material contrasts. We generated 50 virtual structures, each with an epoxy layer of varying thickness  
93 (0 to 190 nm in 10 nm increments) over a bulk material. These structures were simulated at different  
94 accelerating voltages and with various materials. By calculating the mean voxel intensities for each  
95 structure, we derived *realistic* backscattered coefficients. These coefficients, along with atomic  
96 number, accelerating voltage, and penetration depth, were used to train our artificial neural network  
97 (ANN) in a supervised manner.

#### 98 **2.1.2 Real Samples**

99 An HNPG sample with a uniform random network structure and ligament sizes of 15 and 110 nm was  
100 prepared using the dealloying-coarsening-dealloying method (17). To enhance SEM imaging for solid-  
101 pore phase differentiation, the sample was infiltrated with epoxy resin (18). Following the approach  
102 in (19), multi-voltage FIB tomography was conducted using a Dual Beam FEI HeliosNanoLab G3  
103 system with ASV4 software for automated control (20), which also monitored milling progress and  
104 compensated for drift (21).

105 To optimize HNPG tomography, two fiducial markers with intersecting trenches and a ruler system  
106 were used for drift compensation and precise slice thickness determination (22; 23). Three datasets  
107 at different accelerating voltages (1kV, 2kV, and 4kV) were prepared as described in (19) and are  
108 referred to as r-1kV, r-2kV, and r-4kV.

### 109 **2.2 Machine Learning Architecture**

110 We employed an encoder-decoder model based on cycle-consistent adversarial networks (CycleGAN)  
111 (24). This architecture utilizes two GANs with identical structures, each focusing on different tasks:  
112 the first maps data from the source domain to the target domain, while the second maps data from the  
113 target domain back to the source domain. This design enables the model to learn without requiring  
114 paired image data, making it suitable for unsupervised tasks.

115 For the encoder, we used a customized U-Net (25) with residual connections, and for the decoder, we  
116 adopted the architecture proposed in (24).

117 The cycle loss concept was key in addressing two primary challenges in this study: generating  
118 synthetic data and minimizing differences in data distribution.

### 119 **2.3 Generating Synthetic Data Using Machine Learning**

120 To efficiently generate Monte Carlo-like images, we employed a CycleGAN model combined with an  
121 artificial neural network (ANN) trained on physics-based data.

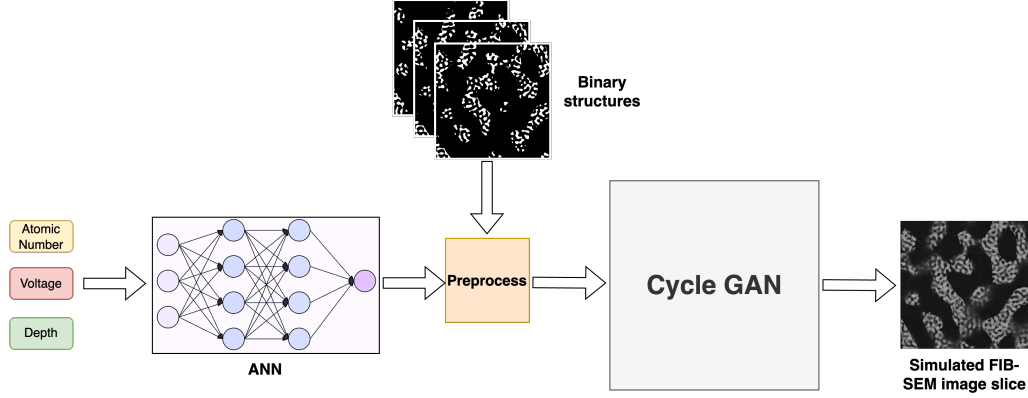


Figure 2: Block diagram of the experimental setup to obtain synthetic images using machine learning based on physical effects

122 In electron microscopy, the relationship between the backscattered electron coefficient ( $\eta$ ), acceler-  
 123 ating voltage ( $E$ ), and atomic number ( $Z$ ) is well-established (Equation 1) (26). However, a direct  
 124 relation between  $\eta$  and penetration depth has not been explicitly defined in existing literature. To  
 125 address this, we defined a basic ANN model, which had three input features, followed by four  
 126 hidden layers containing 16, 32, 32, and 32 neurons, respectively, and a single output to map the  
 127 backscattered electron coefficient with accelerating voltage, atomic number, and penetration depth.

$$\eta(Z, E) = E^{m(Z) \cdot C(Z)} \quad (1)$$

128 Where

$$m(Z) = 0.1382 - \frac{0.9211}{Z^{0.5}} \quad (2)$$

$$C(Z) = 0.1904 - 0.2236 \cdot \ln(Z) + 0.1292 \cdot (\ln(Z))^2 - 0.01491 \cdot (\ln(Z))^3 \quad (3)$$

129 Then, we trained this ANN model in a supervised manner using the sMC-BS dataset (see Section  
 130 2.1.1) to obtain the necessary backscattered coefficients to preprocess the leveled wave method  
 131 (LWM) data before feeding it into the CycleGAN model. These backscattered coefficients were  
 132 normalized across groups of atomic numbers and accelerating voltages and then applied as weights  
 133 to generate weighted grayscale images from the binarized LWM data, typically using 10-15 slices to  
 134 create the final preprocessed dataset.

135 These preprocessed gray-scale images served as the source domain, while the corresponding Monte  
 136 Carlo simulation data acted as the target domain. The CycleGAN model was trained in an unsuper-  
 137 vised manner, similar to domain adaptation, to enhance the robustness of our pipeline. The complete  
 138 pipeline for this approach is illustrated in Figure 2.

## 139 2.4 Reducing Data Distribution Differences Using Machine Learning

140 To address the data distribution discrepancies between synthetic (source domain) and real FIB  
 141 tomography data (target domain), we utilized a CycleGAN-based approach for unpaired image style  
 142 transfer. This process, known as domain adaptation, was applied to synthetic datasets (sMC-1kV,  
 143 sMC-2kV, sMC-4kV, sML-1kV, sML-2kV, and sML-4kV) to produce the final training datasets. The  
 144 full pipeline for generating domain-adapted synthetic data and performing semantic segmentation on  
 145 real HNPG data is illustrated in Figure 3.

## 146 2.5 Semantic Segmentation Using Machine Learning

147 After domain adaptation, we conducted semantic segmentation on all datasets following the approach  
 148 outlined in (3). We trained three different models to evaluate performance: one using domain-adapted

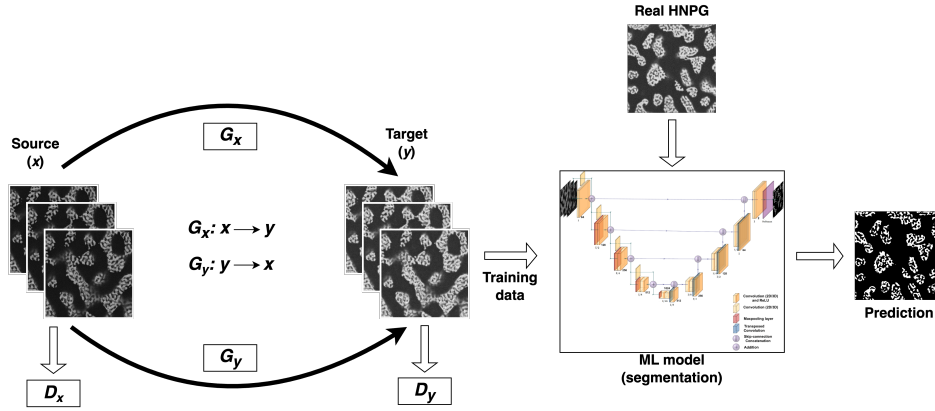


Figure 3: Block diagram of the experimental setup to reduce data distribution difference between synthetic and real HNPG images.  $G_x$  and  $G_y$  are CNN-based generators, and  $D_x$  and  $D_y$  are discriminators for the source domain ( $x$ ) and the target domain ( $y$ ) respectively

149 data, another using Monte Carlo simulation data, and a third using data generated with our proposed  
 150 method. This comparative analysis allowed us to measure the improvement in segmentation accuracy  
 151 due to domain adaptation and our innovative data generation technique.

## 152 2.6 Training Procedure

153 All machine learning models were trained on RTX 3090 GPUs. For CycleGAN, images were cropped  
 154 into smaller patches ( $128 \times 128$ ) with a 64-pixel stride using a sliding window technique, and  
 155 training was conducted on these 2D image patches. The loss functions employed were consistent  
 156 with those proposed in (24), with an initial learning rate of 0.001, which decayed by a factor of 10 if  
 157 no improvement in loss was observed for ten consecutive epochs.

158 For semantic segmentation, we adopted a structured training approach, presenting data as individual  
 159 2D slices, 3D volumetric stacks, or 2D slices combined with neighboring slices in smaller patches  
 160 ( $64 \times 64$ ). The models were optimized using Dice loss in conjunction with the Adam optimizer, with  
 161 an initial learning rate of 0.0001, reduced by a factor of 10 after 10 epochs without improvement.

162 The ANN model, used in the synthetic data generation pipeline, was trained using mean squared error  
 163 (MSE) loss, with a learning rate decay scheme similar to that of the segmentation models. Detailed  
 164 training parameters for all models are summarized in Table 1.

Table 1: Summary of parameters used for training ML models

Parameter	GANs	Semantic Segmentation	ANN
Patch size	128	64	-
Stride	0.5	0.5	-
Batch size	1	64	8
Epochs	100 with early stopping with patience=25		
Loss	CycleGAN Loss	Dice loss	MSE
Optimizer	Adam		
Learning rate	0.0001, adapted with a patience of 10 (reduction factor 10)		

165 **2.7 Evaluation Criteria**

166 We used two types of accuracy metrics to evaluate our methods: those based on ground truth values  
 167 for synthetic datasets and anisotropy-based metrics for real data where ground truth is unavailable.  
 168 These metrics are inspired by (3).

169 **2.7.1 Synthetic Data**

170 For synthetic datasets, where ground truth data is available, we used three metrics:

171 First, misplaced pixels (MP) measures the fraction of incorrectly classified pixels compared to the  
 172 ground truth. It is calculated as:

$$MP = \left(1 - \frac{TP + TN}{TP + FP + FN + TN}\right) \times 100 \quad (4)$$

173 where TP, TN, FP, and FN are true positives, true negatives, false positives, and false negatives,  
 174 respectively. Second, misplaced gold pixels (MGP) assesses the fraction of misclassified gold pixels.  
 175 It is computed by:

$$MGP = \left(1 - \frac{TP}{TP + FN}\right) \times 100 \quad (5)$$

176 MGP is useful for evaluating imbalanced data but does not account for false positives. Third, mean  
 177 Dice score (MDS) evaluates the overlap between predicted and ground truth regions (27). Calculated  
 178 for each phase, the Dice score is:

$$DS = \frac{2TP}{2TP + FN + FP}. \quad (6)$$

179 The MDS averages the Dice scores for the solid and pore phases.

180 **2.7.2 Real Data**

181 We used anisotropy-based metrics for real data where ground truth is unavailable, assuming isotropy  
 182 in the structure. This is a valid assumption for hierarchical nanoporous gold (28; 29). The metrics  
 183 include:

184 First, the two-point correlation function (TPCF) error ( $e_{L_2}^{TPCF}$ ) assesses anisotropy by comparing the  
 185 TPCF values in different directions. It is calculated as:

$$e_{L_2}^{TPCF} = \frac{1}{2} \left( \frac{2 \times \sqrt{\sum_{i=1}^n (f_i^x - f_i^z)^2}}{\sqrt{\sum_{i=1}^n (f_i^x)^2} + \sqrt{\sum_{i=1}^n (f_i^z)^2}} + \frac{2 \times \sqrt{\sum_{i=1}^n (f_i^y - f_i^z)^2}}{\sqrt{\sum_{i=1}^n (f_i^y)^2} + \sqrt{\sum_{i=1}^n (f_i^z)^2}} \right) \quad (7)$$

186 where  $f_i^x$ ,  $f_i^y$ , and  $f_i^z$  are the discretized functional values in the  $x$ -,  $y$ -, and  $z$ -directions, respectively.  
 187 A value of zero indicates perfect isotropy; higher values suggest anisotropy.

188 Second, lineal path function (LPF) error ( $e_{L_2}^{LPF}$ ) is computed analogous to  $e_{L_2}^{TPCF}$  but based on the  
 189 LPF. This metric evaluates anisotropy based on local correlations between points in the same phase.

190 Third, diameter error ( $e_{L_2}^D$ ) compares the predicted ligament diameters in different directions. It is  
 191 calculated by:

$$e_{L_2}^D = \frac{1}{2} \left( \sqrt{\frac{(D_{xz} - D_{xy})^2}{D_{xy}^2}} + \sqrt{\frac{(D_{yz} - D_{xy})^2}{D_{xy}^2}} \right) \quad (8)$$

192 where  $D_{ij}$  represents the average diameter of the ligaments in the  $ij$ -plane. A value of zero indicates  
 193 a geometrically isotropic structure.

194 All metrics are normalized to a range of [0, 1], where 0 denotes perfect accuracy, and 1 indicates a  
 195 complete mismatch.

196 **3 Results**

197 **3.1 Comparing different simulation techniques**

198 To achieve optimal performance in downstream tasks for real datasets, it is crucial to train machine  
 199 learning models using well-prepared data, including synthetic data. This study compares our novel

200 machine learning-based simulation technique with the state-of-the-art Monte Carlo-based simulation  
 201 method. We trained two machine learning models using two different datasets: sML-2kV (prepared  
 202 using our ML-based method) and sMC-2kV (prepared using the Monte Carlo-based method). We then  
 203 evaluated the models by calculating the overlapping regions of binary structures of r-2kV predicted  
 204 by these segmentation models, using the mean Dice score (MDS) as described in Section 2.7.1. An  
 205 MDS value of 0.83 indicates a high degree of overlap between the segmentations, demonstrating the  
 206 effectiveness of our ML-based simulated data.

Table 2: Segmentation results of ML models trained on sML-1kV, sML-2kV and sML-4kV on respective test data prepared using our ML-based simulation method instead of Monte Carlo-based method

Data	MP ↓	MGP ↓	MDS ↑
sML-1kV	0.310	1.251	0.993
sML-2kV	0.539	2.415	0.987
sML-4kV	0.844	3.583	0.980

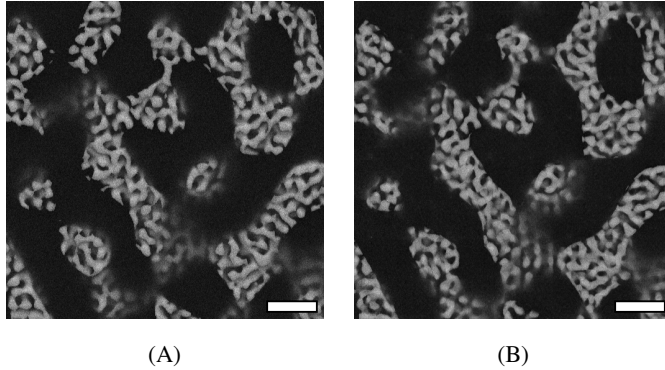


Figure 4: A slice of a synthetic hierarchical nanoporous gold generated using (A) MCXray plugin and (B) our ML-based simulation method - scale bar: 200 nm

207 Additionally, Table 2 demonstrates that models trained on synthetic data prepared using our ML-  
 208 based method exhibit comparable performance in terms of absolute errors. These results are directly  
 209 comparable to those of ML models trained on synthetic data prepared using the Monte Carlo method  
 210 (see Table 3 - only ML). The segmentation results are particularly promising for the real dataset r-2kV,  
 211 with minimal anisotropy-based errors:  $e_{L_2}^{TPCF} = 0.1526$ ,  $e_{L_2}^{LPF} = 0.0359$ , and  $e_{L_2}^D = 0.0222$ . These  
 212 values indicate the strong performance of segmentation models trained using our novel ML-based  
 213 simulation method. Figure 6 in Appendix shows a visual comparison of segmentation performed  
 214 using both ML models.

215 Another notable advantage is the significantly reduced preparation time for simulated data using our  
 216 method, which takes seconds compared to days required for the computation-intensive Monte Carlo  
 217 methods. Figure 4 provides a comparison of a single FIB tomography dataset slice simulated using  
 218 both the Monte Carlo method and our ML-based method.

### 219 3.2 Comparing semantic segmentation after domain adaptation

220 We evaluated the impact of our domain adaptation technique on the segmentation performance of  
 221 machine learning models. Specifically, we compared the segmentation performance on synthetic FIB  
 222 tomography data and real HNPG datasets. The machine learning models were trained on synthetic  
 223 training data both with and without domain adaptation, and the results were predicted on the same  
 224 dataset.

225 **3.2.1 Synthetic Data:**

226 For synthetic datasets, which have ground truth values, we calculated absolute error-based metrics  
 227 as described in Section 2.7.1. Table 3 presents the calculated MP, MGP, and MDS for synthetic test  
 228 datasets (s-1kV, s-2kV, and s-4kV) predicted using machine learning models trained on sMC-1kV,  
 229 sMC-2kV, and sMC-4kV data, and once with domain-adapted sMCDA-1kV, sMCDA-2kV, and  
 230 sMCDA-4kV data. The comparable MP and MGP errors and MDS values for all corresponding  
 231 test datasets indicate that the domain adaptation process does not significantly reduce performance.  
 232 However, to demonstrate the clear advantage of domain adaptation for segmenting real HNPG datasets  
 233 with substantial data distribution differences, we compared the performance of both machine learning  
 234 models on the same real HNPG dataset in the next section. Figure 7 depicts a visual comparison of  
 235 segmentation performed using different techniques.

Table 3: Comparison of segmentation results based on absolute errors with (ML+DA) and without (only ML) domain adaptation on test datasets

Method \ Measure		MP ↓	MGP ↓	MDS ↑
Original		0.000	0.000	1.000
ML + DA	s-1kV	0.401	1.654	0.991
	s-2kV	0.833	3.294	0.980
	s-4kV	1.809	7.428	0.958
only ML	s-1kV	0.245	0.963	0.994
	s-2kV	0.654	2.512	0.985
	s-4kV	1.407	6.466	0.967

236 **3.2.2 Real Data:**

Table 4: Comparison of segmentation results based on isotropy errors with (ML+DA) and without (only ML) domain adaptation on real datasets

Method \ Measure		$e_{L_2}^{TPCF} \downarrow$	$e_{L_2}^{LPF} \downarrow$	$e_{L_2}^D \downarrow$
ML + DA	r-1kV	0.132	0.047	0.029
	r-2kV	0.121	0.096	0.059
	r-4kV	0.140	0.038	0.017
only ML	r-1kV	0.145	0.059	0.030
	r-2kV	0.169	0.091	0.037
	r-4kV	0.271	0.181	0.138

237 It is crucial to assess the effect of training data prepared using domain adaptation on real HNPG  
 238 datasets, as data distribution shifts are typically observed between synthetic and real FIB tomography  
 239 datasets. Table 4 shows the superior performance of machine learning models trained using domain-  
 240 adapted datasets. Notably, the very low errors for the r-4kV dataset for ML models with domain  
 241 adaptation highlight the method’s effectiveness, even for datasets with large artifacts, such as r-4kV.  
 242 Since capturing microscopy images at lower voltages requires significant effort and expertise, our  
 243 domain adaptation method offers a novel approach for microscopists to capture images at higher  
 244 accelerating voltages while achieving similarly good segmentation results. Figure 5 provides a  
 245 histogram comparison of sMC-2kV, sMCDA-2kV, and r-2kV, illustrating the qualitative improvement



246 due to our domain adaptation technique. Figure 8 describes a visual comparison of segmentation of  
247 real HNPG microstructures performed using different techniques.

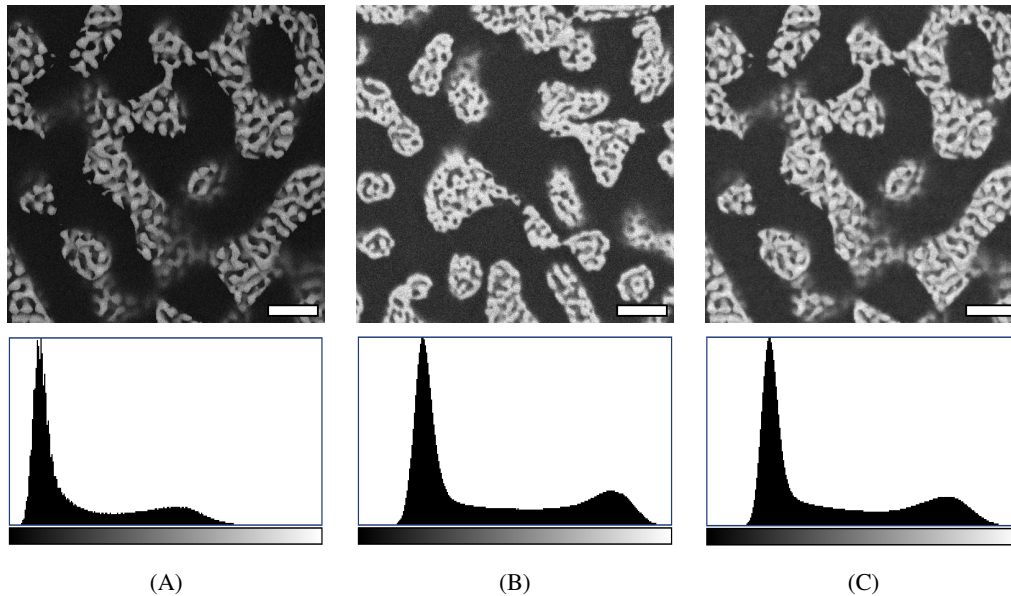


Figure 5: A slice of (A) a synthetic hierarchical nanoporous gold (HNPG) generated using MCXray plugin and (B) a real HNPG structure and (C) a synthetic HNPG structure after domain adaptation. The lower row represents histograms of the respective datasets - scale bar: 200 nm

## 248 4 Conclusion

249 In this study, we proposed a novel method for rapidly generating synthetic data by leveraging  
250 available physics knowledge, thereby bypassing the time-consuming Monte Carlo methods. This  
251 approach addresses the large data requirements for machine learning models, achieving performance  
252 comparable to traditional Monte Carlo based methods. Our method also lays the groundwork for  
253 more sophisticated techniques that can accommodate various materials and microscopy conditions.  
254 Furthermore, we demonstrated that reducing data shifts through domain adaptation techniques  
255 significantly improves reconstruction quality. This allows microscopy data acquired at higher  
256 voltages, which requires less effort, to be used for accurate 3D reconstruction when combined with  
257 our domain adaptation technique. Overall, our work provides a robust framework for enhancing  
258 3D reconstruction accuracy in FIB tomography, making it a valuable tool for studying the physical  
259 properties of nanomaterials. However, it is important to note that generating synthetic data using  
260 our ML-based method has only been tested on the HNPG dataset. Future work should extend  
261 this approach to other materials, and consider replacing the current ANN with more sophisticated  
262 physics-based neural networks.

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347 **A Appendix**

348 **A.1 Visual segmentation results of comparing different simulation techniques**

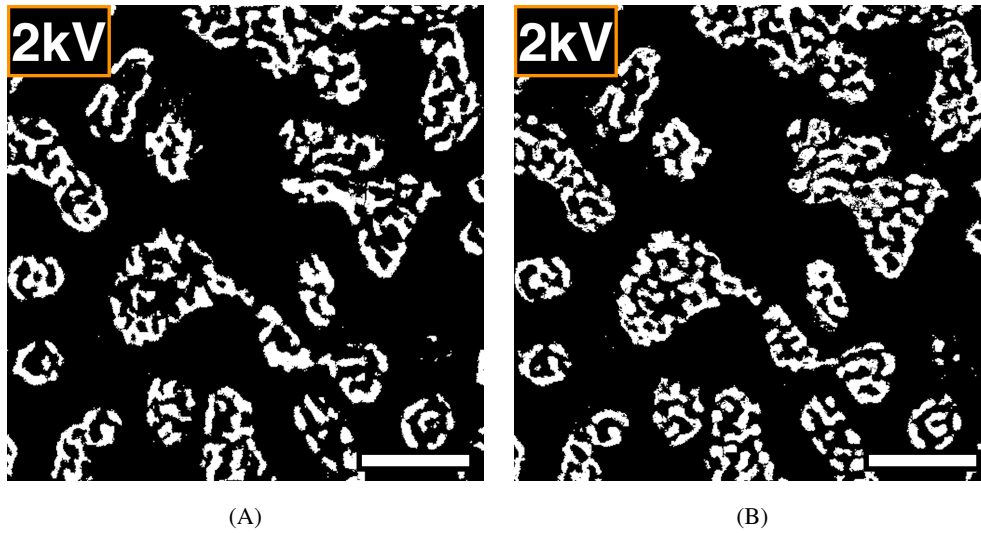


Figure 6: Slice of a real HNPG microstructure segmented using the ML model trained on the synthetic data prepared using (A) MCXray plugin and (B) our ML-based simulation method. Note:  $x$  kV represents the original dataset imaged at an accelerating voltage  $x$  kV - scale bar: 300 nm

349 **A.2 Visual segmentation results of comparing semantic segmentation after domain**  
350 **adaptation**

351 **A.2.1 Synthetic data**

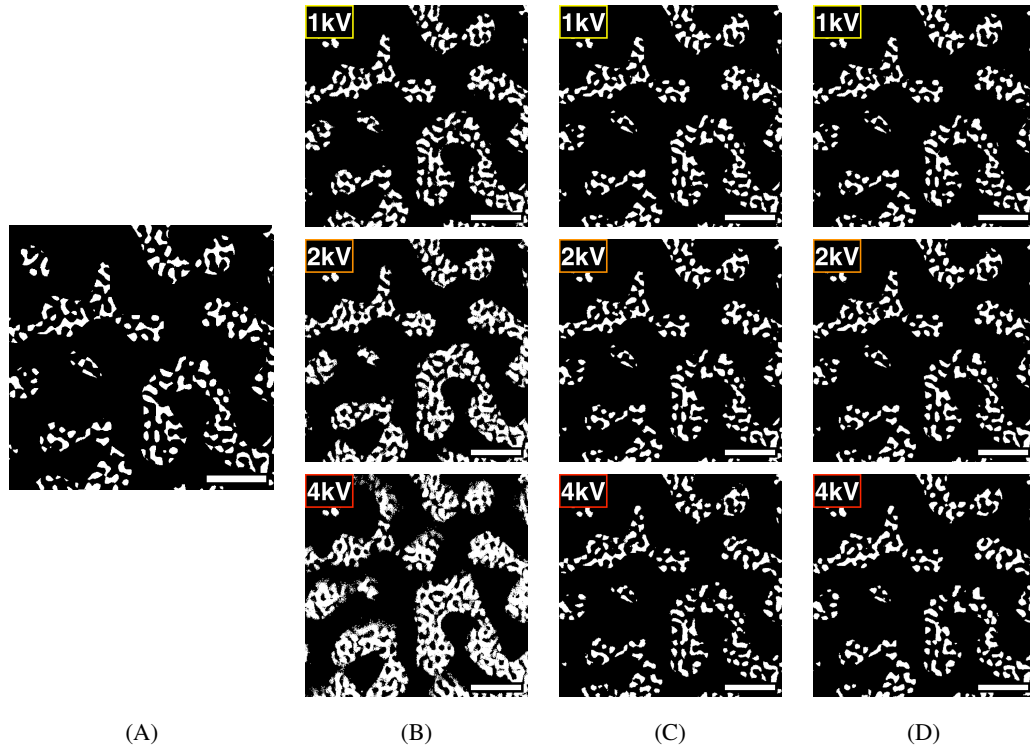


Figure 7: (A) Slice of a synthetic microstructure (ground truth) and segmentation results of Monte Carlo-simulated BSE images using (B) k-means clustering, (C) ML model trained on data without domain adaptation, and (D) ML model trained on data with domain adaptation. Note:  $x$  kV represents the original dataset imaged at an accelerating voltage  $x$  kV - scale bar: 300 nm

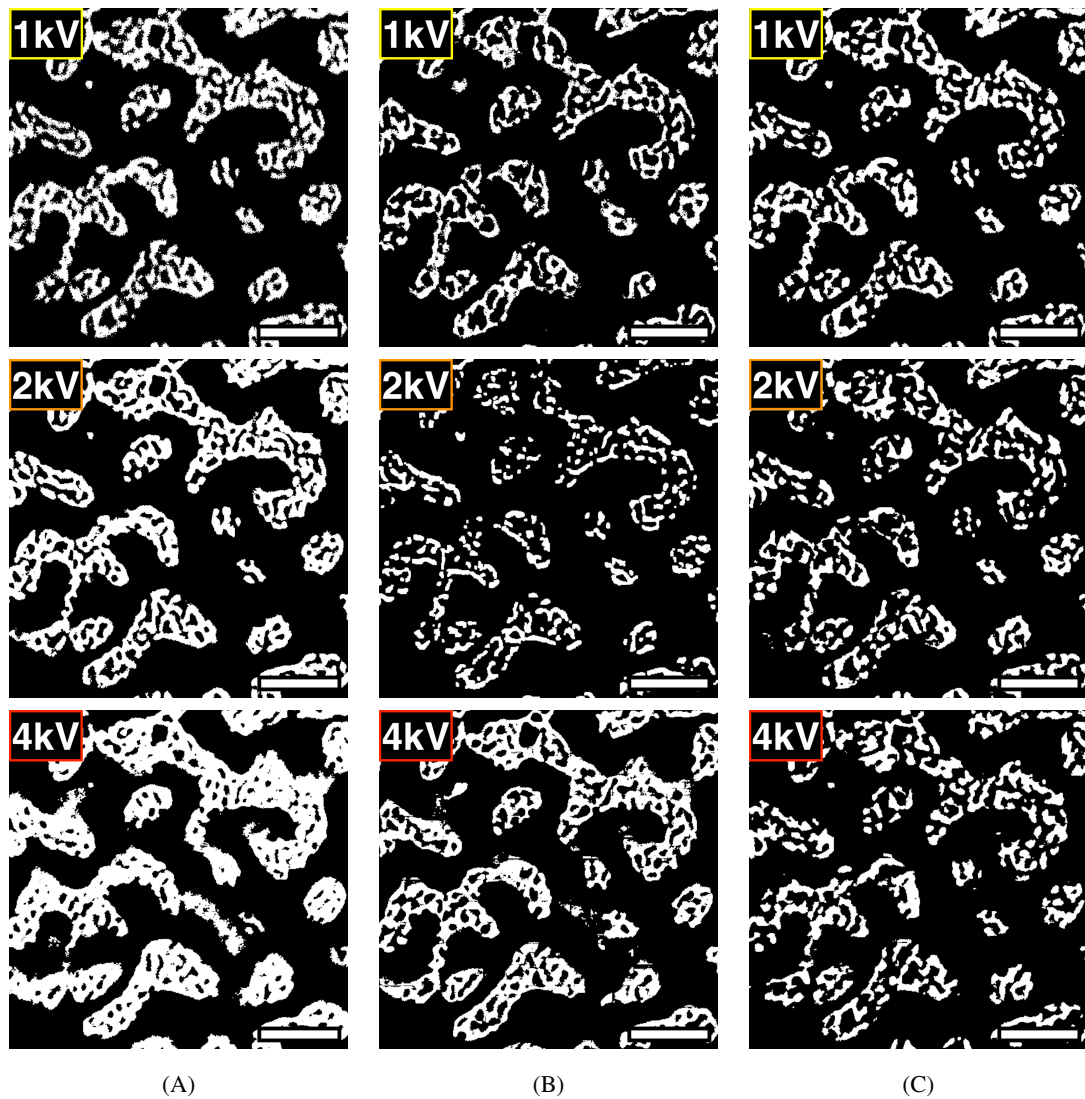


Figure 8: Slice of a real HNP microstructure segmented using (A) k-means clustering, (B) ML model trained on data without domain adaptation, and (C) ML model trained on data with domain adaptation. Note:  $x$  kV represents the original dataset imaged at an accelerating voltage  $x$  kV - scale bar: 300 nm