LLAMP: LARGE LANGUAGE MODEL MADE POWER FUL FOR HIGH-FIDELITY MATERIALS KNOWLEDGE RETRIEVAL

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ABSTRACT

Reducing hallucination of Large Language Models (LLMs) is imperative for use in the sciences, where reliability and reproducibility are crucial. However, LLMs inherently lack long-term memory, making it a nontrivial, ad hoc, and often biased task to fine-tune them on domain-specific literature and data. Here we introduce LLaMP, a multimodal retrieval-augmented generation (RAG) framework of hierarchical reasoning-and-acting (ReAct) agents that can dynamically and recursively interact with computational and experimental data from the Materials Project (MP) and run atomistic simulations via high-throughput workflow interface. Without fine-tuning, LLaMP demonstrates strong tool-usage ability to comprehend and integrate various modalities of materials science concepts, fetch relevant data stores on the fly, process higher-order data (such as crystal structure and elastic tensor), and streamline complex tasks in computational materials and chemistry. We propose a metric combining uncertainty and confidence estimates to evaluate the self-consistency of responses by LLaMP and vanilla LLMs. Our benchmark shows that LLaMP effectively mitigates the intrinsic bias in LLMs, counteracting the errors on bulk moduli, electronic bandgaps, and formation energies that seem to derive from mixed data sources. We also demonstrate LLaMP's capability to edit crystal structures and run annealing molecular dynamics simulations using pre-trained machine-learning interatomic potentials. The framework offers an intuitive and nearly hallucination-free approach to exploring and scaling materials informatics, and paves the way for future agentic scientific workflows and knowledge-grounded LLMs.

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1 INTRODUCTION

The generation of convincing yet unreliable information poses a pressing challenge to large language 037 model (LLMs), particularly to their application in the sciences. LLMs are prone to hallucinationproviding outright false information with high confidence (Bang et al., 2023; Xu et al., 2024). This issue is particularly concerning for knowledge-intensive tasks, where users rely on chatbots and 040 other AI systems to provide accurate guidance (Lewis et al., 2020). LLMs often lack up-to-date 041 factual knowledge on topics outside their training data, requiring rigorous verification against trusted 042 external sources (Mallen et al., 2023). In the scientific community, where the integration of insights 043 and data accuracy is already complex, the proliferation of generative models may exacerbate the risk 044 of misinformation. This trend accentuates the importance of scrutinizing and ensuring the reliability 045 of information sources.

Current approaches to enhance LLM accuracy in domain-specific knowledge often involve finetuning pre-trained models (Dagdelen et al., 2024; Gupta et al., 2022) or tailored prompt engineering techniques (Yang et al., 2023; Zheng et al., 2023). While these models are easy to deploy, they suffer from diminished reproducibility and data adherence due to the absence of a memory base, untraceable fine-tuning history, or opaque extraction processes. Even though fine-tuning can encode a certain amount of domain-specific knowledge into LLMs, it is constrained by scalability and intrinsic memory capacity. Fine-tuned LLMs struggle to retain in the long term the knowledge they were trained on as the training progresses, nor can they be aware of the recent events and data beyond pretraining. Prompt engineering, while effective, also compromises the generalizability, thus limiting the overall power and flexibility of LLMs. Therefore, a more sensible approach involves equipping
 LLMs with external data sources, allowing them to generate holistic responses via few-shot adapta tion to factual information (Lewis et al., 2021) that can reliably support real-world scientific research
 and decision-making.

058 In this work, we propose LLaMP, a multimodal retrieval-augmented generation (RAG) framework leveraging hierarchical reasoning-and-acting (ReAct) agents to interact with Materials Project (MP), 060 arXiv, Wikipedia, and atomistic simulation tools. The framework serves as a safeguard against 061 LLM hallucination by grounding them in high-fidelity material informatics from large-scale mate-062 rial databasederived from various sources, including computational data from quantum-mechanical 063 first-principles calculations and expert-curated material synthesis recipes, and further enables the 064 capabilities of complex downstream tasks. The hierarchical planning of supervisor and assistant ReAct agents improves self-correcting tool-usage performance and enhances the self-consistency 065 in final responses. The new capabilities emerge-such as multi-modal searching, tensor and 3D 066 crystal structure retrieval and operation, and language-driven simulation. The frameworkThrough 067 hierarchical planning of multiple ReAct agents, we demonstrate that LLaMP not only can correctly 068 retrieve high-fidelity, higher-order materials datahigher-order materials data such as tensors and 069 3D crystal structures but also can combine different modalities to perform complex, knowledgeintensive inferences and operations essential for real-world materials science applications. 071

Our contributions are as follows: (1) we introduce a multimodal RAG framework employing hierar-072 chical ReAct agents that dynamically interact with the Materials Project, enabling LLMs to access 073 high-fidelity materials informatics; (2) we propose a statistical metric to assess the self-consistency 074 of LLM responses in high-precision, reproducibility-critical settings; (3) we evaluate the perfor-075 mance of LLaMP and standard LLMs in predicting key material properties, including bulk moduli, 076 electronic bandgaps, formation energies, and magnetic orderings; (4) we showcase real-world appli-077 cations in materials science, such as inorganic synthesis and crystal structure generation and editing; 078 (5) we enhance LLaMP with high-throughput atomistic simulation workflows and pre-trained uni-079 versal ML force fields, lowering the entry barriers to computational materials and chemistry.

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2 BACKGROUND

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Materials Project (MP) The Materials Project is a multi-institution effort to explore and com-084 pute the properties of all known inorganic materials (Jain et al., 2013) and molecules (Spotte-Smith 085 et al., 2023). The initiative leverages high-throughput electronic structure calculations (Kresse and 086 Furthmüller, 1996; Shao et al., 2015) based on density functional theory (DFT), providing large-087 scale open-source database and analysis algorithms, with the ultimate goal to drastically reduce the 088 time and cost required for materials discovery by focusing experiments on the promising candidates 089 from computational screening. Most of the atomic structures are selected from the Inorganic Crystal 090 Structure Database (ICSD) (Zagorac et al., 2019) and undergo standardized relaxation procedures, 091 followed by post-processing or additional calculations for higher-order material properties such as 092 electron and phonon bandgaps, elastic tensors, dielectric tensors, and more. MP provides these 093 calculated material properties through API endpoints.

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NLP and LLM in materials science Natural language processing (NLP) has found extensive 096 application in extracting valuable information from scientific publications, with notable instances 097 involving text-to-text or more recent image-to-text summarization techniques (Gupta et al., 2022; 098 Radford et al., 2021; Tshitoyan et al., 2019). For summarizing crystal structures in textual form, Ganose and Jain (2019) introduced the robocrystallographer, a toolkit designed for the analysis 099 and generation of descriptions for crystalline materials. Their method condenses atomic structures 100 into descriptive JSON representations that encompass coordination statistics, connectivity motifs, 101 geometric features, and dimensionality. MP leverages robocrystallographer to generate human-level 102 descriptions for 130K compounds which are accessible through MP website and API. 103

Recent efforts have curated datasets (Zaki et al., 2023) and benchmarks (Song et al., 2023) to better
evaluate the limitations of LLMs in question answering within the materials science domain. Zhang
et al. (2024) further curated instruction data to fine-tune Llama for material science-specific tasks.
These works focus on general (undergraduate-level) question answering instead of factual grounding on expert-curated database and downstream agentic workflow. In a complementary aspect, other

works address the challenges of extracting complex materials informatics from diverse formats such as tables and unstructured texts (Hira et al., 2024; Schilling-Wilhelmi et al., 2024). This motivates us to augment LLM's knowledge base with MP—one of the most authoratative materials database of stable crystal structures, high-fidelity DFT calculations, inorganic solid-state synthesis recipes, *etc.*

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3 RELATED WORK

116 **Prompting and fine-tuning in domain science** Prompt-based methods have been used as ef-117 fective tools for automating data extraction process from the literature. Polak and Morgan (2023) 118 employ a prompt workflow to extract the cooling rates of metallic glasses and yield strengths of high 119 entropy alloys. Zheng et al. (2023) implement a ChatGPT metal-organic framework (MOF) synthe-120 sis assistant through embedding and searching on preselected papers. StructChem (Ouyang et al., 121 2024) leverages step-by-step reasoning, and iteratively refines results to solve college-level chem-122 istry questions. Yang et al. (2023) use GPT-4 to extract experimentally measured bandgaps to train 123 a graph neural network for accurate bandgap prediction from crystal structures. Despite the success in the specific data extraction tasks, prompt-based methods face challenges in reproducibility when 124 the used prompts are fine-grained to work for specific edge cases. They are also still prone to hal-125 lucination and less generalizable to combine different data sources due to the deliberately designed 126 prompt. 127

128 Several other knowledge-grounded, domain-specific language models lean on the fine-tuning approach against pre-selected data and literature. For instance, ChemGPT (Frey et al., 2022) in-129 volves fine-tuning GPT-neo on self-referencing embedded strings (SELFIES) representations of 130 small molecules. Jablonka et al. (2024) demonstrated GPT-3 fine-tuned against online corpora could 131 outperform purpose-trained models on classification, regression, and inverse design of high-entropy 132 alloys and molecules. Dagdelen et al. (2024) fine-tuned GPT-3 on \sim 500 prompt-completion pairs to 133 enhance LLM's capability to extract useful information on materials chemistry from text paragraphs. 134 However, the fine-tuned models without augmentation inherently lack awareness of the up-to-date 135 results and any data only available after their training. Moreover, fine-tuned LLMs still suffer from 136 limited memory retention and are prone to forget during continual training (Wang et al., 2023).

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138 **LLM function calling and tool usage** An emerging class of LLM applications, including this 139 work, take advantage of LLM text completion and instruction following capability for function 140 calling. This approach extends LLMs with expert-curated tools to improve the quality of control for 141 downstream applications. Coscientist (Boiko et al., 2023) combines tools such as search engines, 142 Python, and document index for autonomous chemical research. ChemCrow (M. Bran et al., 2024) 143 gathers multiple molecule and safety tools to enhance organic chemistry experiment and molecule design. Concurrently, Zhang et al. (2024) develop retrieval based agentic framework on their curated 144 dataset. Ghafarollahi and Buehler (2024) propose AtomAgents for alloy design and analysis. 145

However, most prior works adopt *flat planning* strategy, where a single agent accesses all the available tools, resulting in a lack of self-correcting tool usage capabilities. This often leads to premature
reasoning stop and summarization when the agent encounters tool usage errors. We mitigate this
through *hierarchical planning* of multiple ReAct agents (see Section 4.1).

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- 4 Method
- 4.1 HIERARCHICAL ORCHESTRATION

Overviews Flat planning, where an agent see all the available tools and related API schemas, quickly exceeds LLM context window and incurs huge cost for large-scale database like MP. To manage heterogeneous data sources and diverse types of queries, we introduce hierarchical planning, featuring a supervisor ReAct agent overseeing multiple assistant ReAct agents that have access to the tools (Figure 1). This design offers three major advantages over flat planning commonly implemented in previous works (Boiko et al., 2023; M. Bran et al., 2024): (1) modularity of the system ensures that each assistant agent can focus on domain-specific queries while the supervisor agent handles higher-level reasoning and task allocation; (2) the hierarchical structure improves



Figure 1: Hierarchical ReAct agent planning in LLaMP. Two levels of agents are deployed using a standardized LangChain interface (Chase, 2022). Supervisor ReAct agent oversees assistant ReAct agents at the bottom-level, each equipped with distinct toolkits and data/document stores to accomplish various tasks, including high-fidelity materials information retrieval, atomistic modeling and simulations, and literature search. For a detailed example, refer to Figure A.1.

the overall accuracy and efficiency by reducing the cognitive load on any individual agent; (3) by offloading specific functions to specialized agents, we minimize the context window consumption and schema parsing.

Supervisor agent The supervisor agent acts as a router and decision-maker, handling abstract logic between user requests and assistant agents. Here, we adopt ReAct on GPT-4 (Yao et al., 2023) to augment the agent's action space \mathcal{A} with a language space \mathcal{L} to create an expanded action space of $\hat{\mathcal{A}} = \mathcal{A} \cup \mathcal{L}$. This expanded action space empowers the agent to take action $\hat{a}_t \in \mathcal{L}$ in language space that facilitate the collaboration with assistant agents to retrieve domain specific information and achieve complex downstream tasks such as molecular dynamics simulations.

Assistant agent The efficient function calling in LLMs is often hindered by the need to process complex API schemas, which can consume a significant portion of the context window. To address this, we assign a specialized ReAct agent for each specific tool or API endpoint. It reduces context window consumption, as each agent handles only the relevant schema for its task, avoiding unnecessary schema parsing. Additionally, the use of ReAct agents enables them to refine their API calls based on feedback, significantly improving task completion rates through ReAct's iterative self-correcting mechanism.

The full list of agents and tools are defined in A.1. Each MP assistant agent employs a self-correcting
ReAct mechanism, enabling agents to refine their API calls and improve task completion rates.
The framework's modularity enable a seamless integration of new assistant agents, allowing for
extensibility to various materials discovery methods and experimental techniques (Luo et al., 2023;
Pilania et al., 2017; Wen et al., 2023; 2024; Zeni et al., 2024).

4.2 Self-consistency of response (SCoR)

When LLMs are integrated in scientific workflows and deployed in high-stakes settings (*i.e.* selfdriving labs), it is important for these models to have consistent and predictable behaviors (Liang et al., 2023). For numeric knowledge retrieval tasks, we define the following metrics:

Precision (sample standard deviation) measures the uncertainty in the model's responses where n is the number valid responses from N trials and $\hat{\sigma}$ is the standard deviation of valid response:

$$\text{Precision} = \frac{\hat{\sigma}}{\sqrt{n}} \geq$$

Coefficient of Precision (CoP) maps the precision to (0, 1]:

$$\operatorname{CoP} = \exp\left(-\operatorname{Precision}\right) = \exp\left(-\frac{\hat{\sigma}}{\sqrt{n}}\right) \in (0, 1].$$

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Confidence measures the ratio of generating n valid responses in N trials:

Confidence =
$$\frac{n}{N}$$

Self-consistency of Response (SCoR) is then defined as

 $SCoR = CoP \times Confidence \in [0, 1]$.

The limit of SCoR = 1 is reached when the model yields the same response to a given query every time. At the limit of SCoR = 0, the model is either very inconsistent (with large variance across the responses) or very reluctant (with low confidence) to answer the query. Despite the simplicity in definition, SCoR effectively reflects the reproducibility and practical usability of the method, which is important when the method is incorporated into broader systems where the stable and expected behaviors are prioritized. Refer to Appendix A.2 for the detailed procedure of metric calculation.

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5 EXPERIMENTS

5.1 MULTIMODAL REACT AUGMENTATION

237 Materials design often involves multi-objective property optimization. These properties span a 238 Pareto front where optimizing one factor incurs deterioration in others. To succeed in such tasks, 239 combining different modalities of materials properties is necessary. LLaMP achieves this through 240 the hierarchical orchestration of multiple ReAct agents (Yao et al., 2023). For the example ques-241 tion "What's the stiffest material with the lowest formation energy in Si-O system?" (Figure A.1), 242 when a query requires multimodal information and compound logic, the supervisor agent decomposes the query into multiple subtasks, delegates them to assistant agents (MPThermoExpert and 243 MPElasticityExpert) for information retrieval, and in the final stage of reasoning integrates in-244 formation from both modalities, drawing on the context in episodic memory retrieved from the 245 assistant agents (Figure 1). This enables LLaMP to achieve various tasks step-by-step by combining 246 multiple data sources from the Materials Project (MP) (e.g. 3D crystal structures, thermodynamic, 247 mechanical, magnetic properties, and more listed in Appendix A.1) in a single query. 248

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 5.2

 Performance Benchmarks

251 **Response quality and consistency** We evaluate the performance of LLaMP, StructChem (Ouyang 252 et al., 2024), Darwin (Xie et al., 2023), and vanilla LLMs (gpt-4, llama3-8b, gemini-1.0-pro) 253 on material properties such as bulk modulus, formation energy, and bandgap (Figure 2, Table 1). 254 Performance is assessed through Precision, CoP, SCoR, and MAE metrics, as defined in Section 4.2. 255 We argue that any useful LLM agents to be included in the scientific workflow should have high SCoR and low error on the materials properties. Notably, LLaMP consistently outperforms other 256 models, achieving the highest SCoR and the lowest errors across material properties, making it 257 highly suitable for scientific workflows. StructChem, despite extensive prompting strategies, often 258 fails due to a lack of necessary domain knowledge, resulting in high refusal rates when it cannot 259 validate outputs. 260

For bulk modulus prediction, vanilla LLMs, particularly Llama 3-8b, frequently rely on low-fidelity online data, leading to significant deviations for elements like Cr, Mn, and Fe, compared to MP theoretical values. Interestingly, Llama 3-8b usually cites spurious reference in the responses despite largest response variance but occasionally agrees with MP values. In contrast, LLaMP outperforms vanilla LLMs and reduces the MAE from around 40 to 14.57 GPa.

Our results demonstrate that vanilla LLMs fail to provide accurate formation energy predictions,
 with low SCoR and high MAE ranging from 1.5 to 5.5 eV, which is impractical for material discovery requiring meV-level precision. This is not unexpected, since accurate formation energy prediction requires the computation of multiple energetics (energies of the compound itself and its elemental constituents).



Figure 2: Boxplot of LLaMP RAG responses, baseline methods, and LLM intrinsic knowledge on material properties. (a) Bulk moduli, K, of 3d transition metals. (b) Formation energies, ΔH_f , of common compounds. (c) Electronic bandgaps, E_g , of common intrinsic semiconductors. (d) Electronic bandgaps of multi-element (ternary or quaternary) materials. Missing predictions are marked by shaded areas. Fliers (Outliers) are marked in circles. Horizontal lines represent the MP reference data. All LLaMP results use GPT-4 as backend language provider. Method with higher SCoR has narrower distribution. LLaMP is effectively grounded on MP reference across different tasks and materials.

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In evaluating bandgaps, we query 10 common compounds and 10 multi-element materials that 305 are less commonly encountered in the literature. Vanilla LLMs perform surprisingly well on the 306 bandgaps of common semiconductors (Figure 2c), with expected systematic deviation from MP 307 values retrieved by LLaMP¹. This is likely due to the extensive literature on experimental semi-308 conductor bandgaps, which have been studied and reported for decades. On the contrary, vanilla 309 LLMs lack intrinsic knowledge of the bandgaps for the queried multi-element materials and exhibit 310 low confidence or refuse to make predictions (Figure 2d, Table B6.8), whereas LLaMP retrieves 311 accurate data with a SCoR of 0.938 and correctly identifies the stable polymorph's bandgap when 312 multiple forms are present.

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Ablation study Our frameworks relies on two principal components: first, factual material informatics on MP database; second, stable function calling mechanism that allows assistant agent to interact with tools. In Table 5, we examine three variants: (1) LLaMP: ReAct with MP tools; (2) GPT-4+ReAct with SerpAPI for internet browsing; (3) vanilla GPT-4. LLaMP achieved the best performance when using the complete set of MP tools, highlighting the importance of grounding in up-to-date, high-fidelity materials databases. In Section 4.1, we mentioned the importance of hierarchical planning for robust function call. Evaluating several backbone models on bulk moduli and formation energy prediction, we found LLaMP's grounding performance correlates with the

 ¹Bandgaps calculated from generalized gradient approximation (GGA) functional are known to underestimate the experimental values by 40-50% (Borlido et al., 2020). Strategies to improve bandgap prediction at moderate or low computational cost will be included in MP in the future.

Table 1: Performance metrics of LLaMP and LLM baselines on the prediction of material properties. The metrics from left to right are precision (sample standard deviation), coefficient of precision (CoP), confidence, self-consistency of response (SCoR), and mean absolute error (MAE), where Materials Project are taken as the ground truth. All the tabulated values are the average metrics over five runs and the sampled materials. All LLaMP and StructChem results use GPT-4 as backend language provider. Better method has high SCoR and MAE simultaneously.

		Bulk	Modulus K (O	GPa)			Formati	on Energy ΔH	H_f (eV)	
	Precision↓	CoP	Confidence	SCoR↑	MAE↓	Precision↓	CoP	Confidence	SCoR↑	MAE↓
LLaMP (GPT-4)	2.698	0.900	1.000	0.900	14.574	0.006	0.994	0.940	0.934	0.007
LLaMP (Sonnet)	1.816	0.562	1.000	0.562	15.104	0.000	1.000	1.000	1.000	0.000
LLaMP (Gemini)	5.178	0.053	1.000	0.053	16.251	0.076	0.932	0.620	0.576	0.166
LLaMP (Llama3)	12.993	0.036	0.800	0.029	50.308	0.000	1.000	0.250	0.250	1.377
StructChem	0.000	1.000	0.200	0.200	41.017	0.000	1.000	0.200	0.200	3.146
Darwin	0.001	0.999	0.500	0.499	156.266	0.003	0.997	1.000	<u>0.997</u>	2.245
GPT-4+Serp	2.221	0.833	0.300	0.433	29.937	0.025	0.977	0.560	0.791	11.669
GPT-4	0.186	0.910	1.000	<u>0.910</u>	41.225	0.000	1.000	0.180	0.200	1.680
Sonnet	0.009	0.992	1.000	0.992	41.033	0.022	0.979	1.000	0.979	294.360
Gemini-Pro	6.065	0.169	1.000	0.169	43.429	0.467	0.657	1.000	0.657	1.412
Llama 3	11.222	0.010	1.000	0.010	41.874	2.346	0.139	0.960	0.137	4.657
	Elec	tronic Ba	andgap E_g - C	ommon (e'	V)	Electro	nic Band	lgap E_g - Mul	ti-element	(eV)
	Precision↓	CoP	Confidence	SCoR↑	MAE↓	Precision↓	CoP	Confidence	SCoR↑	MAE↓
LLaMP (GPT-4)	0.000	1.000	0.800	0.800	0.000	0.047	0.958	0.960	0.918	0.167
LLaMP (Sonnet)	0.145	0.870	0.600	0.522	0.298	0.046	0.962	1.000	0.962	0.304
LLaMP (Gemini)	0.627	0.571	0.600	0.343	1.327	0.003	0.997	0.500	<u>0.997</u>	0.637
LLaMP (Llama3)	0.051	0.952	0.800	0.761	1.038	0.169	0.848	0.800	0.678	1.094
StructChem	0.017	0.984	1.000	0.984	0.986	0.000	1.000	0.200	0.200	0.973
Darwin	0.002	0.998	1.000	<u>0.998</u>	1.224	0.000	1.000	1.000	1.000	1.951
GPT-4+Serp	0.040	0.963	1.000	0.963	1.012	0.000	1.000	0.660	0.660	0.576
GPT-4	0.032	0.970	1.000	0.970	0.959	-	-	0.000	0.000	-
Sonnet	0.000	1.000	1.000	1.000	0.938	0.000	1.000	0.500	1.000	0.644
Gemini-Pro	0.034	0.968	1.000	0.968	0.994	0.168	0.849	0.600	0.509	0.989
	0.040	0.070	1 000	0.070	1 0 5 2	0 100	0.000	0.070	0 710	1 00 1

function-calling capability of backbone LLM: Claude-3.5-Sonnet (#1) > Gemini-1.5-Flash (#24) > and Llama3-8B (#46). The number following each model refers to its ranking on the Berkeley Function-Calling Leaderboard at the time of the experiment (Yan et al., 2024).

356 High-fidelity and higher-order data retrieval The challenge for LLMs in excelling at 357 knowledge- and data-intensive tasks is well-documented (Cobbe et al., 2021; Hendrycks et al., 2021; 358 Liang et al., 2023). Figure 3 shows the prediction of LLaMP, GPT-3.5, and GPT-4 on the magnetic orderings and total magnetization of 800 materials randomly selected from all unary, binary, and 359 ternary compounds in MP. Our result indicates that without RAG, vanilla LLMs suffer from hallu-360 cinations and misclassify the magnetic orderings of materials. LLaMP with GPT-4 as backend can 361 counteract the intrinsic bias of GPT models, increasing the classification accuracy to 0.98 and R^2 of 362 magnetization prediction to 0.992 (Table 2). We note that GPT-3.5 as backend, while effective for 363 classification and other information retrieval tasks, struggles to distinguish total_magnetization 364 from magnetization_per_formula_unit in magnetism API schema and often requests the wrong 365 field and forgets to normalize the values. In the magnetic orderings queries, LLaMP with GPT-3.5 as 366 backend fails to distinguish ferromagnetic (FM) and ferrimagnetic (FiM) orderings, while LLaMP 367 with GPT-4 as backend gracefully separates the two classes (Figure 3a, d).

368 We further test the capability of LLaMP and LLMs 369 for higher-order data (such as tensors, 3D crystal struc-370 tures, curves). As shown in Table B6.2, GPT-3.5 hal-371 lucinates the values for the components in the elastic 372 tensor of NaCl, with serious erroneous values such as 373 $C_{11} = 289.2 \,\text{GPa}$ —a significant deviation from DFT-374 calculated values (76 GPa). It also omits the values for 375 $C_{22}, C_{33}, C_{55}, C_{66}$ and fails to represent the full elastic 376 tensor in a matrix format, despite the query explicitly requesting the full elastic tensor. This highlights the lim-377 itation of intrsinic knowledge in LLMs to recall higher-

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Table 2: Prediction performance of LLaMP, GPT-3.5, and GPT-4 on magnetic orderings and magnetization. LLaMP with GPT-4 and GPT-3.5 as backend LLM are compared.

	Magnetic C	Ordering	Magnetization		
	Accuracy	F1	MAE	\mathbb{R}^2	
LLaMP (GPT-4)	0.98	0.89	0.045	0.992	
GPT-4	0.48	0.26	1.611	-0.201	
LLaMP (GPT-3.5)	0.96	0.88	1.896	0.407	
GPT-3.5	0.23	0.18	1.988	-0.024	



Figure 3: Prediction of LLaMP, GPT-3.5, and GPT-4 on (a,b,d,e) magnetic orderings and (c,f) total magnetization per formula unit of randomly selected materials. Confusion matrix presents the number of entries in each class. Colormap represents the percentage of correct classification.

order, more complex data for more comprehensive, holistic response.

5.3 REAL-WORLD APPLICATIONS

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Inorganic synthesis recipes Empowered by the MP synthesis endpoint (Kononova et al., 2019), LLaMP can extract synthesis recipes and summarize detailed step-by-step procedures grounded on real experimental papers with associated DOI references, as demonstrated in the example queries (Table B6.9 and B6.10).

419 Vanilla LLMs often give seemingly correct and ver-420 bose synthesis procedures but pull irrelevant com-421 pounds into the recipes and overlook more optimal 422 or efficient reactions. In the example of YMnO₃ (Table B6.9), GPT-3.5 suggests the possible reaction 423 pathways from two common oxide precursors (Y_2O_3) 424 and MnO₂). However, it pulls irrelevant lithium 425 compounds (Li₂CO₃ and LiOH) into the recipe and 426 overlooks the fact that metathesis reactions (Li et al., 427 2015; Todd et al., 2021) require less applied energy

Table 3: Positive-unlabeled (PU) classification of LLaMP and baseline methods on inorganic material synthesizablity. (*) Evaluations on 352,236 positive and 40,817 unlabeled compounds by Kim et al. (2024).

	Accuracy	F1	Precision	Recall
LLaMP (GPT-4)	0.800	0.773	0.895	0.680
LLaMP (Sonnet)	0.818	0.812	0.848	0.780
GPT-4	0.600	0.649	0.578	0.740
Sonnet	0.530	0.230	0.636	0.140
Llama3	0.480	0.623	0.489	0.860
Gemini	0.590	0.388	0.765	0.260
GPT-4*	-	-	0.151	0.620
GPT-3.5 (FT)*	-	-	0.558	0.951
stoi-CGNF*	-	-	0.541	0.942

than high-temperature sintering, which relies on solid-state diffusion (Maximenko and Olevsky, 2004).

431 Vanilla LLMs also exhibit uncertainty about specific synthesis details, such as heating temperature, duration, cooling rate, *etc.* In some edge cases such as LiFePO₄ presented in Table B6.10, the cited

432 references are associated with the real papers but the contents are dissociated from the cited titel and 433 hallucinated from the pre-training corpus. 434

We further compare the performance of LLaMP on synthesizability prediction with stoichiometric convolutional graph neural fingerprint (stoi-CGNF) (Jang et al., 2024) and fine-tuned LLMs (Kim et al., 2024). We follow the positive-unlabeled (PU) classification task proposed in (Kim et al., 2024) by randomly selecting a subset of positive (probable) and unlabeled (unlikely) inorganic compounds and compare the classification performances of LLaMP with different backend LLMs and baselines. As presented in Table 3, LLaMP effectively enhances the performances of backbone GPT-4 and Sonnet LLMs by a significant margin of 20%, with the classification precision of LLaMP (GPT-4) up to 0.895.



(a) MP (DC Si, mp-149)

hexagonal interstitial)



(c) GPT-3.5 (distorted Si with Li tetrahedral interstitial)

Figure 4: Generation and manipulation of crystal structures using LLMs to insert an additional lithium atom at the interstitial site in diamond cubic silicon structure. Blue: Si. Green: Li. Questionanswer pairs are listed in Table B6.11. Additional atoms extended through bonds are visualized.

Table 4: Structural parameters of the generated crystals compared with diamond cubic (DC) silicon. From left to right are fractional coordinates of inserted Li atom $(x, y, z)_{Li}$, total cell volume V, average Si-Si bond lengths ℓ_{SiSi} , Si-Si-Si angles θ_{SiSiSi} , and Si-Li-Si angles θ_{SiLiSi} . GPT-4 refuses to respond due to their safeguard against the lack of atomic structure information.

	$(x,y,z)_{\rm Li}$	ℓ_{SiSi} (Å)	Error (%)	$V(\text{\AA}^3)$	Error (%)	$\theta_{\rm SiSiSi}$ (°)	Error (%)	$\theta_{\rm SiLiSi}$ (°)
LLaMP	(0.5, 0.5, 0.5)	2.36	0.0	40.33	0.0	109.47	0.0	62.96
GPT-3.5	(0.5, 0.5, 0.5)	2.71	+15.0	67.05	+66.3	98.28	-10.2	67.69
GPT-4	-	-	-	-	-	-	-	-
DC Si (mp-149)		2.36		40.33		109.47		

RAG-assisted crystal generation and editing Fine-tuned LLMs for text-encoded atomistic infor-470 mation have shown the capability to generate stable crystals under the constraints of atomic positions 471 and charges (Gruver et al., 2023). In this context, we delve into the examination and comparison 472 of the crystal generation capabilities between LLaMP and GPT-3.5, without resorting to fine-tuning 473 or tailored prompt messages in previous work. Figure 4 showcases the structures generated by 474 LLaMP and vanilla GPT-3.5 without RAG, both instructed to insert one lithium atom at the tetrahe-475 dral site of the diamond cubic silicon structure (Table B6.11). Notably, both LLaMP and GPT-3.5 476 place an additional Li atom at fractional coordinate (0.5, 0.5, 0.5). However, the Si structure re-477 trieved by LLaMP adheres to the MP convention, positioning two Si bases at (0.125, 0.125, 0.125)and (0.875, 0.875, 0.875). This causes the inserted Li atom to be *hexagonal interstitial* instead of 478 tetrahedral interstitial. 479

480 GPT-3.5 locates the Li atom at the tetrahedral site given the "luckily chosen" Si bases at (0,0,0) and 481 (0.25, 0.25, 0.25); however, the resulting cell volume and shape are highly distorted, and the Si–Si 482 bond length and Si–Si–Si angle deviate significantly from the ground truth (Table 4), highlighting the limitations in the intrinsic encoding of LLMs for atomistic information and the challenges 483 associated with zero-shot generation of crystal structures. In contrast, the LLaMP-retrieved MP 484 structure serves as a robust prior, anchoring the lattice parameters of the generated structure to the 485 correct values.

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486 Language-driven simulation LLaMP equipped with Python REPL and atomistic simulation 487 workflow package atomate2 performs well out of the box for complex multi-step simulations us-488 ing pre-trained universal machine learning interatomic potential MACE-MP-0 (Batatia et al., 2023) 489 through language instruction. As demonstrated in Appendix C.1 and Appendix C.2, LLaMP is able 490 to follow multi-step instruction to fetch stable crystal structure from MP, generate a supercell of atomic structure, and run annealing molecular dynamics simulation with varying temperature from 491 300K to 800K and back to 300K. After the simulation is finished, LLaMP can read the simulation 492 trajectories and plot the temperature profile over time (Appendix C.1). 493

494 We further test the robustness of our language-driven workflow on running MD simulations (Fig-495 ure 5). A subset of 50 supercell structures were randomly created from up-to-quinary compounds 496 in MP. Each MD simulation runs 0.1 ps with timestep of 2 fs. The timeout was set to 90 seconds. 96% workflows (SUCCESS+TIMEOUT) were successfully initiated, with 62% finished and 34% of sys-497 tems exceeding 90 seconds timeout due to slow or stalled MACE-MP-0 runs (the simulation is still 498 running without error but runs slowly). 4% simulations ran into unspecified status (UNKNOWN). We 499 found that during these triggered workflows LLaMP asks user for approval on the precise chemical 500 formula to fetch the structure from MP, rendering the workflow unfinished. 501

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6 DISCUSSION

Robustness The hierarchical ReAct framework implemented 506 here is essentially a graph of agents, or language graph, with one 507 central node (supervisor) in connection with many satellite nodes 508 (assistants). The implementation of ReAct for the assistant agents 509 enables self-correcting tool usages and fortifies the robustness of 510 data retrieval. As presented in Figure A.1c, MPThermoExpert ini-511 tially misunderstood the schema at the first trial and filled in the 512 formula field with Si-O, an invalid input but a valid one for chem-513 ical system (chemsys) field. The observation step (step 4) allows 514 MPThermoExpert to handle exceptions and to refine the correct in-515 put fields after adaptation (step 6). Storing (Retrieving) question-516 answer and query-argument pairs to (from) vector databases could 517 further reduce the number of trial-and-error steps, and the stored pairs can be used to refine foundation LLMs to improve function 518 calling quality. 519

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521 Limitation We recognize the effectiveness of LLaMP's frame-522 work relies on backbone LLM's function calling and reasoning 523 capabilities. Sometimes LLMs misunderstand the description of 524 schemas and therefore yield unexpected behaviors. For example, 525 sort_fields argument allows sorting the returned documents in ascending order or descending order if the field is prefixed with -, 526 but LLMs sometimes mistake the sign and sort in the opposite or-527 der. Other example failure or safeguard modes are presented in Ap-528 pendix A.3. The correctness of LLaMP is also subject to the quality



Figure 5: The final four statuses (SUCCESS, TIMEOUT, FAILURE, UNKNOWN) of trial language-driven MD simulation runs on random MP supercell structures. LLaMP successfully initiated 96% (SUCCESS+TIMEOUT) of all the simulations within 90 second timeout window.

529 of theoretical prediction and the comprehensiveness of the data in MP. Other than the underpredicted 530 bandgaps by GGA functional, MP's ongoing effort to search all possible magnetic configurations is 531 also not complete. Most of the existing calculations in MP start from high-spin ferromagnetic con-532 figurations, which may overlook many antiferromagnetic ground states below the current energy 533 convex hull. While MP is one of the most comprehensive materials databases, the available crys-534 tal structures on MP are not exhaustive but continuously expanding (Merchant et al., 2023), and 535 would be benefited from additional intermetallic compounds and high-entropy materials from other 536 databases such as AFLOW, OQMD, NOMAD, etc. (Curtarolo et al., 2012; Kirklin et al., 2015; 537 Scheidgen et al., 2023). Furthermore, Kohn-Sham DFT theory is insufficient in some cases, and a higher level of theory is needed. Currently LLaMP only supports a few atomate2 workflows with 538 machine learning force fields and VASP calculations. More diverse electronic calculation methods and workflows will be supported in the future work.

540 **Summary** We present a hierarchical agentic framework, LLaMP, based on ReAct to extract and 541 manipulate material informatics through few-shot generalization. By grounding thoughts and ac-542 tions with high-fidelity information, LLaMP showcases the ability to integrate various modalities of 543 material properties and perform logical inferences to accomplish assigned tasks, all without the need 544 for fine-tuning. In essence, the proposed LangChain framework holds the potential to expand its applicability to multiple data sources, encompassing both theoretical computations and experimental data, knowledge graph databases (Venugopal and Olivetti, 2024; Ye et al., 2024), and real-world lab-546 oratories by incorporating additional assistant agents for data retrieval and robot control (Fei et al., 547 2024). LLaMP functions as a knowledge-aware agent, empowering users to navigate and manipulate 548 complex materials informatics. In the context of self-driving labs (Boiko et al., 2023; Szymanski 549 et al., 2023), LLM agents with multimodal data sources, sensors, and actors may improve their de-550 cision making and operation (Miret and Krishnan, 2024). As new tools continue to emerge, there 551 is an exciting avenue for further exploration to ascertain if this framework can effectively facilitate 552 scientific hypothesis generation and guide data-driven experiments. 553

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756	А	SUPPLEMENTARY INFORMATION
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759	A.1 LIST OF IMPLEMENTED ASSISTANT AGENTS AND TOOLS
760	
761	Here we provide the comprehensive list of implemented assistant agents and tools. Note that
762	MP Assistants are highly modular so it is very trivial to support extra API endpoints from
763	https://api.materialsproject.org/docs.
76/	
765	• MPSummaryExpert: summary provides amalgamated data for a material by combining
766	subsets of data from many of the other API endpoints.
767	• MPThermoExpert : thermo provides computed thermodynamic data for a material such
769	as formation energy and energy above hull.
760	
770	 MPElasticityExpert: elasticity provides bulk, shear, and Young's modulus, poisson
771	ratio, and universal anisotropy index.
779	• MPM agnotism Export: magnotism provides computed magnetic ordering related data
773	• In MagnetismExpert. magnetism provides computed magnetic ordering related data.
774	• MPDielectricExpert: dielectric provides computed dielectric data from density func-
775	tional perturbation theory.
776	
777	• MPPiezoelectricExpert : piezoelectric provides computed piezoelectric data from den-
778	sity functional perturbation theory.
770	• MPElectronicExpert : electronic structure provides computed electronic structure
780	related data for a material such as band gap and fermi level. Python objects for line-mode
781	band structures, density of states, and fermi surfaces are also available.
782	
783	• MPSynthesisExpert : synthesis provides a synthesis recipes for materials extracted from
784	literature using text mining and natural language processing techniques.
785	• MPStructureRetriever : MaterialsStructureText fetches and saves pymatgen Struc-
786	ture objects to local JSON files.
787	
788	• MLFFAgent: MLFFMD runs molecular dynamics simulations using pre-trained machine
789	learning force fields; MLFFE1astic calculates the elastic constants of a given material using
790	pre-trained machine learning force fields.
791	• PythonREPI Tool: Python REPL that LLMs could run the generated script.
792	
793	• ArxivQueryRun: LangChain built-in tool that LLMs can use to send API request to ArXiv.
794	· Willingdig Quemplum, I anothein huilt in tool that II Ma and the second ADI
795	• wikipediaQuerykun: Langenain duitt-in tool that LEWIS can use to send API requise to Wikipedia
796	wikipula.
797	
798	A.2 METRIC CALCULATION IN TABLE 1 AND FIGURE 2
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800	The following procedures are adopted to calculate the metrics for material property regression
801	benchmarks presented in Table 1 and Figure 2:
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803	1. Each method was presented with the same query asking for the property of multiple materi-
804	als. Here we ask each method for bulk modulus, formation energy, and electronic bandgap
805	of ten materials. For example, "What are the bulk moduli of the following metals: Sc, Ti, V_{i} Cr. Ma Fa Ca Ni Cr. $7\pi^{2}$ "
806	\mathbf{v} , $\mathbf{U}\mathbf{r}$, $\mathbf{W}\mathbf{I}\mathbf{n}$, $\mathbf{F}\mathbf{e}$, $\mathbf{U}0$, $\mathbf{N}\mathbf{I}$, $\mathbf{U}\mathbf{u}$, $\mathbf{Z}\mathbf{n}\mathbf{r}$.
807	2. Repeat the same query for five times and collect the responses.
808	
809	3. The numerical values are extracted and formatted into 2D arrays with the aid by LLMs. If the range is provided in the response (<i>e.g.</i> Llama 3), the median value was used.

or databases would be necessary.

810		4 Calculate Precision CoP Confidence and SCoR for each method across five trials on all
811		materials. The code is provided and the pseudocode can be written as follows:
812		
813		Input: arr (2D array of five responses for different materials)
814		$n \leftarrow count of valid responses (non-NaN values) in each column of arr$
815		$\operatorname{prec} \leftarrow \frac{\operatorname{hanstd}(\operatorname{arr}, \operatorname{axis}=0)}{\sqrt{n}}$
816		$cop \leftarrow mean(\exp(-prec))$
817		$conf \leftarrow mean\left(\frac{n}{\# of trials N}\right)$
818		if $conf = 0$ for all columns
819		$cop \times conf$ otherwise
820		$prec \leftarrow mean(prec)$
821		Output: prec, cop, conf, scor
822		
823	A.3	SAFEGUARD OR FAILURE MODES
824		
825		• Question: What is the bulk modulus of stainless steel? ambiguous chemical formula
826		LLaMP: Stainless steel is an alloy with a composition that can vary widely, including
827		elements like iron, chromium, nickel, and molybdenum. Because of this variability, it does
828		not have a single chemical formula or a unique Materials Project ID, and its bulk modulus
829		Project detabase focusing on inorganic crystalline materials may not provide specific data
830		on complex alloys like stainless steel. For precise values of the bulk modulus, consulting
831		specific datasheets or literature on the type of stainless steel in question is recommended.
832		
833		• Question: How to synthesize Y2Be2Fe(SiO5)2 data unavailable
834		LLaMP: Currently, there are no synthesis recipes available in the Materials Project
835		database for Y2Be2Fe(SiO3)2.
836		• Question: What's the phonon bandgap of Ac2HgGe computed property unavailable
837		LLaMP: The material Ac2HgGe, with a Materials Project ID of mp-861872, does not have
838		its phonon bandgap information directly available through the Materials Project database.
839		For detailed phonon properties, further analysis using specialized phonon calculation tools

8	4	0
8	4	1

Table 5: Performance of LLaMP with different backbone LLMs and ablation of ReAct agent with MP API and SerpAPI tools.

		Bulk Moduli K (GPa)				Formation Energies ΔH_f (eV)				
	Precision↓	CoP	Confidence	SCoR↑	MAE↓	Precision↓	CoP	Confidence	SCoR↑	MAE↓
LLaMP	2.698	0.900	1.000	0.900	14.574	0.007	0.993	0.960	0.953	0.009
LLaMP (Sonnet)	1.345	0.702	0.867	0.608	16.121	0.000	1.000	1.000	1.000	0.000
LLaMP (Gemini-1.5)	21.586	0.217	1.000	0.217	64.324	0.390	0.718	0.560	0.402	0.531
LLaMP (Llama 3)	7.314	0.327	0.800	0.261	47.386	0.413	0.721	1.000	0.721	3.062
GPT-4+Serp	2.222	0.352	1.000	0.352	41.742	5.947	0.745	1.000	0.745	8.214
GPT-4	0.186	0.910	1.000	0.910	41.225	0.000	1.000	0.180	0.180	1.680



Figure A.1: Multimodal retrieval-augmented generation for materials informatics. (a) User query. 908 (b) Supervisor ReAct agent capable of handling multiple assistant agents and high-level reasoning. 909 (c-d) Assistant ReAct agents executing function calling and summarization. (c) MPThermoExpert 910 and (d) MPElasticityExpert have access to the API schemas of thermo and elasticity endpoints 911 on Materials Project, respectively. The selected details are highlighted in red, demonstrating the 912 capabilities of RAG and ReAct implemented in LLaMP. The blue texts show LLaMP assistant ReAct 913 agent can handle API calling errors and self-correct the input query accordingly.

- 914
- 915
- 916
- 917

918 A.4 PROMPT TEMPLATE

920 We use the ReAct template hwchase17/react-multi-input-json from LangChain Hub 921 (https://smith.langchain.com/hub/hwchase17/react-json) as follows: 922 923 Answer the following questions as best you can. You have access to the following 924 tools: 925 {tools} 926 927 The way you use the tools is by specifying a JSON blob. 928 Specifically, this JSON should have an 'action' key (with the name of the tool to 929 use) and an 'action_input' key (with the input to the tool going here). 930 The only values that should be in the "action" field are: {tool_names} 931 932 The \$JSON_BLOB should only contain a SINGLE action, do NOT return a list of multiple 933 actions. Here is an example of a valid \$JSON_BLOB: 934 ... 935 {{ 936 "action": \$TOOL_NAME, 937 "action_input": \$INPUT 938 }} 939 940 ALWAYS use the following format: 941 942 Question: the input question you must answer 943 Thought: you should always think about what to do 944 Action: 945 \$JSON_BLOB 946 ... 947 Observation: the result of the action 948 ... (this Thought/Action/Observation can repeat N times) 949 Thought: I now know the final answer Final Answer: the final answer to the original input question 950 951 Begin! Reminder to always use the exact characters 'Final Answer' when responding. 952 953 The above system prompt message is prepended with the following solicitation to guide the super-954 visor agent: 955 956 You name is LLaMP and you are a helpful agent that can consult materials-related 957 data through Materials Project (MP) database, arXiv, Wikipedia, and run 958 calculations/simulations via a python REPL and MP atomate2 computational 959 workflow. 960 961 If you get an error, debug your code and try again. Only use the output of your code to answer the question. Ask user to clarify their queries if needed. Please 962 note that you don't have direct control over MP but through multiple assistant 963 agents to help you. You need to provide complete context in the input for 964 assistants to do their job. REFINE and CRITQUE the output of the assistants if 965 needed. You can end the conversation by saying "Bye". 966 967 968 969

B SAMPLE QUESTIONS

	Question		Answer
	Question	LLaMP	GPT-3.5
B6.1	What are the bulk mod- uli of the following metals: Sc, Ti, V, Cr, Mn, Fe, Co, Ni, Cu, Zn?	The bulk moduli of the metals you asked for are as follows (in GPa): - Scandium (Sc): Voigt = 45.715 , Reuss = 45.34 , VRH = 45.528 - Titanium (Ti): Voigt = 112.955 , Reuss = 112.807 , VRH = 112.881 - Vanadium (V): Voigt = 181.232 , Reuss = 181.232 , VRH = 181.232 - Chromium (Cr): Voigt = 252.155 , Reuss = 252.155 , VRH = 252.155 - Manganese (Mn): Voigt = 280.959 , Reuss = 280.959 , VRH = 280.959 - Iron (Fe): Voigt = 296.003 , Reuss = 295.207, VRH = 295.605 - Cobalt (Co): Voigt = 200.587 , Reuss = 200.587, VRH = 200.587 - Nickel (Ni): Voigt = 197.399 , Reuss = 197.29, VRH = 197.345 - Copper (Cu): Voigt = 151.394 , Reuss = 151.394, VRH = 151.394 - Zinc (Zn): Voigt = 76.283 , Reuss = 95.46, VRH = 85.872	The bulk modulus (K) is a measure of a material's resistance to uniform compression. It is defined as the ratio of the change in pressure to the fractional volume compression. The values for the bulk modulus can vary somewhat depending on the specific conditions and crystallographic structures of the materials. The values provided here are approximate and given in gigapascals (GPa): - Scandium (Sc): 55 GPa - Titanium (Ti): 110 GPa - Vanadium (V): 160 GPa - Chromium (Cr): 160 GPa - Manganese (Mn): 120 GPa - Iron (Fe): 170 GPa - Cobalt (Co): 180 GPa - Nickel (Ni): 180 GPa - Copper (Cu): 140 GPa - Zinc (Zn): 70 GPa These values represent typical bulk moduli for these metals, and actual values may vary depending on factors such as temperature and pressure.
			Continued on next page

Table 6: QA comparison between LLaMP and GPT-3.5.

	Question		Answer
	Question	LLaMP	GPT-3.5
B6.2	What is the full elastic ten-	The elastic tensor for NaCl is	The elastic tensor describes the relationship between stress and strain in a material.
	sor of NaCl?	available for two different entries	It is a 4th-rank tensor that characterizes the elastic properties of a crystal. In the
		on Materials Project. For	case of NaCl (sodium chloride), the elastic tensor is typically represented by a 6x6
		material id 'mp-22851', the elastic	matrix.
		tensor in IEEE format is [[76.0, 1.0,	The elastic constants of NaCl can vary slightly depending on the specific condi-
		1.0, -0.0, 0.0, -0.0], [1.0, 76.0, 1.0,	tions and crystallographic orientation. Here is an approximate form of the elastic
		-0.0, -0.0, 0.0], [1.0, 1.0, 76.0, 0.0,	tensor for NaCl:
		-0.0, -0.0], [-0.0, -0.0, 0.0, -3.0,	C11 = 289.2 GPa C12 = 97.8 GPa C44 = 81.6 GPa C13 = C14 = C15 = C16 =
		-0.0, -0.0], [0.0, -0.0, -0.0, -0.0,	C23 = C24 = C25 = C26 = 0 GPa
		-3.0, 0.0], [-0.0, 0.0, -0.0, -0.0, 0.0,	Note: The values are given in gigapascals (GPa). The elastic constants Cij rep-
		-3.0]]. For material id 'mp-22862',	resent the components of the elastic tensor. In this case, C11, C12, and C44 are
		the elastic tensor in IEEE format is	the non-zero elastic constants. The off-diagonal terms C13, C14, C15, C16, C23,
		[[47.0, 12.0, 12.0, 0.0, -0.0, -0.0],	C24, C25, and C26 are typically zero due to cubic symmetry.
		[12.0, 47.0, 12.0, 0.0, -0.0, -0.0],	Keep in mind that these values are approximate, and there may be slight varia-
		[12.0, 12.0, 47.0, 0.0, 0.0, -0.0],	tions depending on the source and the specific conditions under which the mea-
		[0.0, 0.0, 0.0, 12.0, -0.0, -0.0], [-0.0,	surements were made. Experimental techniques, such as Brillouin scattering or
		-0.0, 0.0, -0.0, 12.0, 0.0], [-0.0, -0.0,	ultrasonic measurements, are often used to determine the elastic constants of ma-
		-0.0, -0.0, 0.0, 12.0]]	terials.
B6.3	What are the bandgaps of	The ground state bandgaps for the	As of my last knowledge update in January 2022, I can provide you with approx-
	the following materials: Si,	requested materials are as follows:	imate values for the ground state bandgaps of the materials you mentioned. Keep
	Ge, and GaAs, GaN, SiC,	Si: 1.12 eV, Ge: 0.66 eV, GaAs:	in mind that these values can vary depending on factors like temperature and crys-
	BN, InSe, MoS2, BaTiO3,	1.42 eV, GaN: 3.28 eV, SiC: 2.36	tal structure. Additionally, more recent research may have refined these values.
	CsPbI3? Only look for	eV, BN: 6.4 eV, InSe: 1.26 eV,	Here are the approximate ground state bandgaps in electron volts (eV):
	ground-state bandgaps.	MoS2: 1.29 eV, BaTiO3: 3.2 eV,	Silicon (Si): 1.1 eV Germanium (Ge): 0.67 eV Gallium Arsenide (GaAs): 1.42 eV
		CsPbI3: 1.73 eV. Please note that	Gallium Nitride (GaN): 3.4 eV Silicon Carbide (SiC): 3.2 eV (for 4H-SiC) Boron
		these values are typical for bulk	Nitride (BN): 6 eV Indium Selenide (InSe): 1.25 eV Molybdenum Disulfide
		materials and can vary depending	(MoS2): 1.3 eV (indirect bandgap) Barium Titanate (BaTiO3): 3.2 eV Cesium
		on the specific structure and	Lead Iodide (CsPbI3): 1.7 eV Please note that these values are general approx-
		calculation method.	imations and may not represent the specific conditions or variations in different
			forms of these materials. For the most accurate and up-to-date information, I rec-
			ommend checking recent scientific literature or databases.
			Continued on next page

Table 6 – continued from previous page

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Question		Answer				
	Question	LLaMP	GPT-3.5			
B6.4	Give me the decomposi-	The maximum decomposition	The decomposition enthalpies (in kJ/mol) of the following materials are:			
	tion enthalpies of the fol-	enthalpies for the requested	Ag2O3: -335			
	lowing materials: Ag2O3,	materials are as follows: - Ag2O3:	MgZn2: -82			
	MgZn2, GaAs, GaN, SiC,	0.2565836414999989 eV/atom -	GaAs: -317			
	BN, InSe, MoS2, BaTiO3,	MgZn2: 0.1356 eV/atom - GaAs:	GaN: -505			
	CsPbI3. If there are multi-	0.075 eV/atom - GaN: 0.07162	SiC: -40			
	ple values, please give me	eV/atom - SiC: -0.172 eV/atom -	BN: -248			
	the largest value.	BN: -0.358 eV/atom - InSe:	InSe: -125			
	-	-0.01816 eV/atom - MoS2: 0.267	MoS2: -393			
		eV/atom - BaTiO3: 3.055 eV/atom	BaTiO3: -112			
		- CsPbI3: 0.01753601900000902	CsPbI3: -153			
		eV/atom Please note that the values				
		for SiC, BN, and InSe are negative,				
		indicating that the decomposition is				
		exothermic. All other values are				
		positive, indicating endothermic				
		decompositions.				
B6.5	Give me the formation	The smallest formation energies of	Ag2O3: -2.63 eV/atom MgZn2: -0.27 eV/atom GaAs: -4.13 eV/atom GaN: -3.51			
	energies of the follow-	the materials you asked for are as	eV/atom SiC: -4.98 eV/atom BN: -3.8 eV/atom InSe: -0.89 eV/atom MoS2: -3.31			
	ing materials: Ag2O3,	follows: Ag2O3: -0.3601 eV/atom,	eV/atom BaTiO3: -4.45 eV/atom CsPbI3: -2.32 eV/atom			
	MgZn2, GaAs, GaN, SiC,	MgZn2: -0.1306 eV/atom, GaAs:				
	BN, InSe, MoS2, BaTiO3,	-0.446 eV/atom, GaN: -0.052				
	CsPbI3. If there are multi-	eV/atom, SiC: -0.204 eV/atom, BN:				
	ple values, please give me	-1.41 eV/atom, InSe:				
	the smallest value.	-0.742582154921875 eV/atom,				
		MoS2: -1.202 eV/atom, BaTiO3:				
		-3.493 eV/atom, CsPbI3: -1.296				
		eV/atom.				
	1	1	Continued on next page			

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	Question	Answer	
	Question	LLaMP	GPT-3.5
B6.6	Give me the decomposi-	The smallest decomposition	The decomposition enthalpies of the materials are as follows: - Ag2O3: -1138.5
	tion enthalpies of the fol-	enthalpy values for the requested	kJ/mol - MgZn2: -54.2 kJ/mol - GaAs: -452.2 kJ/mol - GaN: -510.9 kJ/mol -
	lowing materials: Ag2O3,	materials are as follows: Ag2O3:	SiC: -2133.6 kJ/mol - BN: -2551.0 kJ/mol - InSe: -209.3 kJ/mol - MoS2: -739.3
	MgZn2, GaAs, GaN, SiC,	0.019373648083333528 eV/atom,	kJ/mol - BaTiO3: -393.5 kJ/mol - CsPbI3: -65.3 kJ/mol
	BN, InSe, MoS2, BaTiO3,	MgZn2: -0.0197 eV/atom, GaAs:	
	CsPbI3. If there are multi-	-0.446 eV/atom, GaN: -0.05231	
	ple values, please give me	eV/atom, SiC: -0.204 eV/atom, BN:	
	the smallest value.	-1.41 eV/atom, InSe: -0.06336	
		eV/atom, MoS2: -0.669 eV/atom,	
		BaTiO3: -0.063 eV/atom, CsPbI3:	
		-0.018602 eV/atom.	
B6.7	Give me the energy	Here are the lowest energy above	The energy above hull values for the given materials are as follows: - Ba2MnSe3:
	above hull values of	hull values for the requested	0.0000 eV/atom - KAlSiO4: 0.0000 eV/atom - LiCuO: 0.0000 eV/atom -
	the following materials:	materials: Ba2MnSe3: 0.0 eV/atom	H8C3N2O: 0.0000 eV/atom - Dy2ZnS4: 0.0000 eV/atom - Rb2BeF4: 0.0000
	Ba2MnSe3, KAlSiO4,	KAlSiO4: 0.0 eV/atom LiCuO: 0.0	eV/atom - K9Fe2S7: 0.0000 eV/atom - Tb2FeC4: 0.0000 eV/atom - ScAlAu:
	LiCuO, H8C3N2O,	eV/atom H8C3N2O: 0.0735	0.0000 eV/atom - CaCrO4: 0.0000 eV/atom
	Dy2ZnS4, Rb2BeF4,	eV/atom Dy2ZnS4: 0.00966	
	K9Fe2S7, Tb2FeC4,	eV/atom Rb2BeF4: 0.0 eV/atom	
	ScAlAu, CaCrO4. If there	K9Fe2S7: 0.0 eV/atom Tb2FeC4:	
	are multiple values, please	0.0 eV/atom ScAlAu: 0.0 eV/atom	
	give me the lowest one.	CaCrO4: 0.0 eV/atom. Please note	
		that these are the lowest values	
		found in the Materials Project	
		database and actual values may	
		vary based on different entries.	
			Continued on next page

	Question	Answer		
	Question	LLaMP	GPT-3.5	
B6.8	What are the bandgaps of the following materi- als: Ba(PdS2)2, FePO4, DyBi2IO4, NaFe(SO4)2, Eu2B5BrO9, Sr5As3H, TI- GaS2, AgSbS2, Pr3S3N, LiLaO2. If there are poly- morphs, give me only the bandgap of the stable ma- terial.	The bandgaps for the materials are: - Ba(PdS2)2: 0.779 eV - Most stable FePO4: 2.1932 eV - DyBi2IO4: 1.367 eV - Most stable NaFe(SO4)2: 2.094 eV - Eu2B5BrO9: 0.88 eV - Sr5As3H: 0.676 eV - TIGaS2: 1.7885 eV - Most stable AgSbS2: 1.372 eV - Pr3S3N: 1.15 eV - Most stable LiLaO2: 3.7336 eV Please note that these values are obtained from the Materials Project database and might be different in experimental results due to various factors.	I don't have real-time data access, and my training only includes information available up to January 2022. As a result, I may not have specific bandgap values for the materials you mentioned. However, I can provide you with general infor- mation about some of these materials and their known properties up to my last update: Ba(PdS2)2: I don't have specific bandgap information for this compound. The bandgap may depend on factors such as crystal structure and synthesis conditions. FePO4: Iron(II) phosphate, known as ferrous phosphate, typically has a bandgap in the range of 2.5-3.0 eV. DyBi2IO4: Dysprosium bismuth iodate. I don't have specific bandgap informa- tion for this compound. NaFe(SO4)2: Sodium iron sulfate. I don't have specific bandgap informa- tion for this compound. Eu2B5BrO9: Europium borate bromate. I don't have specific bandgap informa- tion for this compound. Sr5As3H: Strontium arsenide hydride. I don't have specific bandgap information for this compound. TIGaS2: Thallium gallium sulfide. I don't have specific bandgap information for this compound. AgSbS2: Silver antimony sulfide. This material is a semiconductor with a bandgap typically around 1.5-2.0 eV. Pr3S3N: Praseodymium sulfide nitride. I don't have specific bandgap information for this compound. LiLaO2: Lithium lanthanum oxide. I don't have specific bandgap information for this compound.	

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1137 1138 1140 1141 1142 1143 1144 1144 1144 1144 1144 1144 1145 1145 1145 1145 1145 1151 1152 1153 1154 1155 1155 1156 1157 1158 1153 1154 1155 1156 1161 1162 1163 1164 1165 1165 1166 1167 1168 1168 1168

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	Question	Answer		
	Question	LLaMP	GPT-3.5	
B6.9	How to synthesize YMnO ₃ ?	To synthesize YMnO ₃ , a solid-state reaction method can be employed. Here is a summarized procedure based on the synthesis details provided in the literature with DOI: 10.1021/ic502027k: 1. Dissolve Y(NO3)3·6H2O, Mn(NO3)2·4H2O, and a source of [OH-] in water to create a homogeneous solution. 2. Add citric acid to the solution to act as a mixing media. 3. Heat the mixture at 250°C for 2 hours to achieve pre-calcination. 4. Finally, heat the pre-calcined mixture at a temperature range of 700°C to 930°C for 10 to 18 hours in an air atmosphere to form YMnO ₃ .	Synthesizing YMnO ₃ (yttrium manganite) typically involves the solid-state reac- tion method. Please note that the details may vary depending on the specific con- ditions and equipment available in your laboratory. Here's a general procedure: Materials : Yttrium oxide (Y ₂ O ₃) Manganese oxide (MnO ₂ or Mn ₂ O ₃) High- purity manganese carbonate (MnCO ₃) Lithium carbonate (Li ₂ CO ₃) or lithium hy- droxide (LiOH) Crucible and pestle High-temperature furnace Mortar and pestle Platinum crucible or alumina crucible Procedure: Step 1: Weighing the Precursors Weigh the appropriate amounts of yttrium oxide (Y ₂ O ₃), manganese oxide (MnO ₂ or Mn ₂ O ₃), and lithium carbonate (Li2CO3) or lithium hydroxide (LiOH) ac- cording to the stoichiometric ratio of YMnO3. Step 2: Mixing the Powders 2. GPT-3.5 starts to fail numbering correctly in this response. Grind the powders thoroughly in a mortar and pestle to ensure a ho- mogenous mixture. Step 3: Heating 3. Place the mixed powder in a crucible or alumina crucible. Heat the mixture in a high-temperature furnace. The specific temperature and heating rate depend on the exact synthesis conditions, but a typical range is 800- 1100°C. Step 4: Sintering 5. Hold the mixture at the target temperature for a specific duration (several hours) to allow for sintering and the formation of YMnO ₃ . Step 5: Cooling 6. Slowly cool the sample to room temperature to avoid thermal stress. Step 6: Characterization 7. Characterize the synthesized YMnO ₃ using tech- niques such as X-ray diffraction (XRD), scanning electron microscopy (SEM), and energy-dispersive X-ray spectroscopy (EDS) to confirm the desired phase and composition. Notes: The choice of manganese precursor (MnO ₂ or Mn ₂ O ₃) and lithium source (carbonate or hydroxide) can impact the synthesis process and may require op- timization. It's crucial to use high-purity starting materials to avoid impurities in the final product. The specific temperatures and times mentioned are general guidelines and m	

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page

Table 6 – continued	from	previous	page	

Question			Answer
Question		LLaMP	GPT-3.5
Question B6.10 How to LiFePO ₄ ? vide a detai step procedu reference.	synthesize Please pro- iled step-by- ire and the	LLaMP LiFePO ₄ can be synthesized using various methods. One method involves a high-temperature solid-state synthesis using precursors such as Fe ₂ O ₃ , LiOH, and NH ₄ H ₂ PO ₄ . The precursors are mixed and heated at temperatures up to 850°C for several days. Another method uses a sol-gel combustion approach with precursors like FePO ₄ and Li ₂ CO ₃ , where the mixture is calcined at 700°C in a nitrogen atmosphere. For detailed procedures, please refer to the following scientific literature: 1. High-temperature solid-state synthesis - DOI: 10.1021/jp407510h 2. Ion-exchange method - DOI: 10.1021/cm402617b 3. Solution method - DOI: 10.1021/jp306936t 4. Solid-state reactions - DOI: 10.1021/cm303259j 5. Sol-gel combustion method - DOI: 10.1007/s10853-011-6139-7	 GPT-3.5 Synthesizing LiFePO₄ (lithium iron phosphate) typically involves a solid-state reaction method. Below is a step-by-step procedure for synthesizing LiFePO₄: Materials: - Lithium carbonate (Li2CO3) - Iron (II) acetate (Fe(CH₃COO)₂) - Ammonium dihydrogen phosphate (NH4H2PO4) - Phosphoric acid (H3PO4) - Deionized water - Ethanol - Mortar and pestle - Crucible - Furnace Procedure: 1. Weigh out the appropriate amounts of lithium carbonate (Li2CO3) and iron (II) acetate (Fe(CH₃COO)₂) in a 1:1 molar ratio. For example, if you want to synthesize 1 mole of LiFePO₄, you would weigh out 74.55 grams of Li₂CO₃ and 173.93 grams of Fe(CH₃COO)₂. Grind the lithium carbonate and iron (II) acetate together using a mortar and pestle to ensure a homogenous mixture. Weigh out the appropriate amount of ammonium dihydrogen phosphate (NH4H2PO4) and add it to the lithium carbonate and iron (II) acetate mixture in a 1:1 molar ratio. For example, if you are synthesizing 1 mole of LiFePO₄, you would weigh out 122.00 grams of NH4H2PO4. Add a few drops of phosphoric acid (H3PO4) to the mixture to adjust the pH to around 7. This will help in the formation of LiFePO₄. Slowly evaporate the water from the slurry by placing it on a hot plate or in a furnace at a low temperature. Once the water has evaporated, heat the mixture in a crucible in a furnace at around 600-700°C for several hours to initiate the solid-state reaction and form LiFePO₄. After the reaction is complete, allow the crucible to cool and collect the LiFePO₄ powder. Wash the LiFePO₄ powder with ethanol to remove any impurities and then dry it in a desiccator. Reference: P. G. Bruce, B. Scrosati, JM. Tarascon, "Nanomaterials for Rechargeable Lithium Batteries," Angew. Chem. Int. Ed. 2008, 47, 2930-2946.
l			Continued on next many
		DOI: 10.1007/s10853-011-6139-7	 9. Wash the LiFePO₄ powder with ethanol to remove any impurities and then dry it in a desiccator. Reference: P. G. Bruce, B. Scrosati, JM. Tarascon, "Nanomaterials for Rechargeable Lithium Batteries," Angew. Chem. Int. Ed. 2008, 47, 2930-2946. DOI: 10.1002/anie 200702505
			DOI: 10.1002/anie.200702505.

Table 6 – continued from previous page

	Question	Answer		
	Question	LLaMP	GPT-3.5	
B6.11	Could you insert one Li atom at the tetrahedral in- terstitial site in diamond cubic Si? Pull out the most stable DC Si structure from MP directly in JSON string, modify the retrieved structure directly, and give me the final pymatgen structure as a JSON string.	<pre>The final structure after inserting a Li atom at the tetrahedral interstitial site in the diamond cubic Si structure is as follows: {'@module': 'pymatgen.core.structure', ' @class': 'Structure', 'charge': 0, 'lattice': {'matrix': [[3.33573, 0.0, 1.924639], [1.111191, 3.142924, 1.924639], [0.0, 0.0, 3.849278]], 'pbc': [True, True, True], 'a': 3.8492784033699095, 'b ': 3.849278, 'alpha': 60.00001213094421, 'beta': 60.00001097545789, 'volume': 40.32952684741405}, 'properties': {}, 'sites': [{'species': [{' element': 'Si', 'occu': 1]}, 'abc': [0.875, 0.875, 0.875], 'xyz': [3.8891685, 2.7500584999999997, 6.7362365], 'properties': {'magmom ': -0.0}, 'label': 'Si'}, {'species ': [{'element': 'Si', 'occu': 1]], 'abc': [0.125, 0.125, 0.125], 'xyz ': [0.5555955, 0.3928655, 0.9623195], 'properties': {'magmom ': -0.0}, 'label': 'Si'}, {'species ': [{'element': 'Li', 'occu': 1]], 'abc': [0.5, 0.5, 0.5], 'xyz': [2.222382, 1.571462, 1.924639], ' properties': {}, 'label': 'Li'}]}</pre>	<pre>Here's the JSON string for the most stable diamond cubic Si structure retrieved directly from MP: "''json ('sites': [('species': [['element': 'Si', 'occu': 1]], 'abc': [0, 0, 0], 'xyz': [0, 0, 0], 'label': 'Si', 'properties': ()), ('species': [['element': 'Si', 'occu': 1]], 'abc': [0.25, 0.25, 0.25], 'xyz': [0.178, 0.178, 0.178], 'label': 'Si', 'properties': ()]], 'lattice', ('matrix': [[2.866002, 0, 2.243049], [0, 3.866002, 2.243049]], 's'.5.431, 'b': 5.431, 'c': 5.431, 'aipha': 109.471219, 'beta': 109.471219, 'gamma': 109.471219, 'volume': 129.5843543803), 'properties': ('space_group': ('number': 227, 'source': 'spglib', 'symbol': 'fd-3m']}) Now, let's insert one Li atom at the tetrahedral interstitial site in this diamond cubic Si structure and retrieve the final pymatgen structure as a JSON string: ''json ''sters': [('species': [['element': 'Si', 'occu': 1]], 'abc': [0, 0, 0], 'xyz': [0, 0, 0], 'zyz': [0, 0, 0], 'label': 'Si', 'properties': ()], ('species': [['element': 'Si', 'occu': 1]], 'abc': [0, 0, 0], 'zyz': [0.78, 0.178, 0.178], 'label': 'Si', 'properties': ()], ('species': [['element': 'Li', 'occu': 1]], 'abc': [0, 0, 0.5, 0.5], 'zyz': [0.5, 0.5, 0.5], 'zatice'; 'matrix': [[2.866002, 0, 2.243049], 'a.566002, 2.3686002, 0], 'a 5.431, 'c': 5.431,</pre>	

Table 6 –	continued	from	previous	page

	Question	Answer		
	Question	LLaMP	GPT-3.5	
B6.12	What is the magnetic or-	The stable FePO ₄ has a	The magnetic ordering of stable $FePO_4$ is antiferromagnetic. Its material id is mp-	
	dering of stable $FePO_4$?	ferromagnetic (FM) magnetic	5405, and its space group is Pnma. The magnetization per formula unit is 4.14	
	Give me its material id,	ordering. Its material ID is	μB.	
	space group, and magneti-	'mp-753395', it belongs to the		
	zation per formula unit.	Triclinic crystal system with space		
		group symbol 'P-1' and number 2.		
		The total magnetization per formula		
		unit is 5.00000015 Bohr		
		magnetons.		

1269 C OTHER CAPABILITIES

1271 C.1 AUTOMATED COMPUTATIONAL WORKFLOW

 The following example shows how LLaMP uses computational workflow atomate2 and job manager jobflow to run automated molecular dynamics simulation using foundation machine learning interatomic potential MACE-MP-0 (Batatia et al., 2023).

```
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1277
          Ouerv:
1278
          I would like you to help me run MD simulation with the follwing steps:
1279
          1. Retrieve the stable LiTaO3 structure from Materials Project
1280
          2. Create a supercell around the size of 100 atoms
1281
          3. Use atomate2 with foundation ML forcefield MACE to run Nose-Hoover NVT simulation
1282
            for 2 ps with annealing temperature scheduling 300 to 800 back to 300 Kelvin.
1283
            Use 2 fs timestep and save the trajectory to a ASE traj file.
          4. Read the trajectory file and plot the system temperature vs time
1284
1285
          > Entering new AgentExecutor chain...
1286
          Thought: Do I need to use a tool? Yes
1287
          Action: MPStructureRetriever
1288
          Action Input: LiTaO3
1289
          > Entering new AgentExecutor chain...
1290
          Action:
1291
          ʻʻʻjson
1292
          {
1293
            "action": "search_materials_structure__get",
1294
            "action_input": {
             "formula": "LiTaO3",
1295
             "limit": 5,
1296
              "fields": "material_id,structure"
1297
            }
          }
1299
1300
1301
          All retrieved structures are saved as Pymatgen Structure JSON files to the following
               paths: mp-3666.json, mp-1105280.json, mp-754345.json, mp-1105216.json, mp
              -1105326.json
1303
1304
          > Finished chain.
1305
          {'input': 'LiTaO3', 'output': 'All retrieved structures are saved as Pymatgen
1306
              Structure JSON files to the following paths: mp-3666.json, mp-1105280.json, mp
1307
              -754345.json, mp-1105216.json, mp-1105326.json'}
          Do I need to use a tool? Yes
1308
          Action: MPSummaryExpert
1309
          Action Input: mp-3666
1310
1311
          > Entering new AgentExecutor chain...
1312
          Action:
1313
          '''json
          {
1314
            "action": "search_materials_summary__get",
1315
            "action_input": {
              "material_ids": "mp-3666"
1317
            }
1318
          }
          ```Error on search_materials_summary__get: 'fields' must be specified in the query.
1319
 Please revise arguments or try smaller request by specifying 'limit' in request
1320
1321
 ʻʻʻjson
1322
 {
```

1323 "action": "search\_materials\_summary\_\_get", 1324 "action\_input": { 1325 "material\_ids": "mp-3666", "fields": "material\_id,formula\_pretty,composition,nsites,symmetry" 1326 } 1327 } 1328 1329 1330 [{'nsites': 10, 'composition': {'Li': 2.0, 'Ta': 2.0, 'O': 6.0}, 'formula\_pretty': ' 1331 LiTaO3', 'symmetry': {'crystal\_system': 'Trigonal', 'symbol': 'R3c', 'number': 1332 161, 'point\_group': '3m', 'symprec': 0.1, 'version': '2.0.2'}, 'material\_id': ' 1333 mp-3666'}] 1334 Action: ʻʻʻjson 1335 { 1336 "action": "Final Answer" 1337 "action\_input": "Material ID mp-3666 corresponds to Lithium Tantalate (LiTaO3). It 1338 has a trigonal crystal system with space group R3c (number 161) and point group 3m. The structure consists of 10 sites, composed of Li, Ta, and O." 1339 } 1340 ... 1341 Finished chain. 1342 {'input': 'mp-3666', 'output': 'Material ID mp-3666 corresponds to Lithium Tantalate 1343 (LiTaO3). It has a trigonal crystal system with space group R3c (number 161) 1344 and point group 3m. The structure consists of 10 sites, composed of Li, Ta, and 0.'} 1345 Do I need to use a tool? Yes 1346 Action: MPStructureRetriever 1347 Action Input: mp-3666 1348 1349 Entering new AgentExecutor chain... Action: 1350 '''json 1351 { 1352 "action": "search\_materials\_structure\_\_get", 1353 "action\_input": { "material\_ids": "mp-3666", 1354 "limit": 1, 1355 "fields": "material\_id,structure" 1356 } 1357 } 1358 ... 1359 1360 1361 All retrieved structures are saved as Pymatgen Structure json files to the following 1362 paths: mp-3666.json 1363 Do I need to use a tool? Yes 1364 Action: Python\_REPL 1365 Action Input: 1366 1367 1368 1369 from pymatgen.core import Structure 1370 1371 # Load the structure from the JSON file 1372 structure = Structure.from\_file('mp-3666.json') 1373 # Create a supercell to approximately reach 100 atoms 1374 num\_atoms = len(structure) 1375 multiplier = int((100 / num\_atoms) \*\* (1/3)) + 1 # Calculate the multiplier for each 1376 dimension

supercell = structure \* (multiplier, multiplier, multiplier)

```
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1428
```

```
Save the supercell structure to a file and print the number of atoms
supercell.to(fmt="poscar", filename="supercell.vasp")
print({"supercell_path": "supercell.vasp", "num_atoms": len(supercell)})
• • •
{'supercell_path': 'supercell.vasp', 'num_atoms': 270}
Do I need to use a tool? Yes
Action: MLFFMDAgent
Action Input: {"structure_path": "supercell.vasp", "forcefield": "MACE",
 simulation_type": "NVT", "temperature_schedule": [300, 800, 300], "
 simulation_time": 2, "timestep": 2, "trajectory_file": "trajectory.traj"}Do I
 need to use a tool? Yes
> Entering new AgentExecutor chain...
Action:
'''json
{
 "action": "MLFF MD",
 "action_input": {
 "run_mode": "local",
 "project": "llamp-atomate2",
 "atom_path_or_dict": "supercell.vasp",
 "force_field_name": "MLFF.MACE",
 "time_step": 2,
 "n_steps": 1000,
 "ensemble": "NVT",
 "dynamics": "langevin"
 "temperature": [300, 800, 300],
 "traj_file": "trajectory.traj",
 "traj_file_fmt": "ase",
 "traj_interval": 1
 }
}
''Action:
2024-05-12 14:42:31,797 INFO Started executing jobs locally
2024-05-12 14:42:31,837 INFO Starting job - Forcefield MD (040866d1-fbcb-4e0b-9044-0
 ed3cf9d0eef)
Using Materials Project MACE for \texttt{MACECalculator} with /global/homes/c/
 cyrusyc/.cache/mace/5yyxdm76
Using \texttt{float32} for \texttt{MACECalculator}, which is faster but less
 accurate. Recommended for MD. Use float64 for geometry optimization.
Default dtype float32 does not match model dtype float64, converting models to
 float32.
2024-05-12 14:43:58,683 INFO Finished job - Forcefield MD (040866d1-fbcb-4e0b-9044-0
 ed3cf9d0eef)
INFO:jobflow.core.job:Finished job - Forcefield MD (040866d1-fbcb-4e0b-9044-0
 ed3cf9d0eef)
```

```
2024-05-12 14:43:58,710 INFO Finished executing jobs locally
INFO:jobflow.managers.local:Finished executing jobs locally
{'energy': -2262.60595703125, 'n_steps': 1001}
```



1484

## • • •

plt.legend()

plt.show()

plt.savefig('temperature\_vs\_time.png')

print({"temperature\_vs\_time\_plot": "temperature\_vs\_time.png"})

```
import matplotlib.pyplot as plt
from ase.io import read
Load the trajectory
trajectory = read('trajectory.traj', index=':')
Extract temperatures and time steps
temperatures = [step.get_temperature() for step in trajectory]
time_steps = [i * 2 for i in range(len(temperatures))] # Calculate time steps
manually since each step is 2 fs
Plotting
plt.figure(figsize=(10, 5))
plt.plot(time_steps, temperatures, label='Temperature_(K)', marker='o')
plt.title('Temperature_vs_Time_during_MD_Simulation')
plt.xlabel('Time_(fs)')
plt.ylabel('Temperature_(K)')
plt.grid(True)
```



#### C.2 SIMULATION CODE GENERATION AND EXECUTION

Below we present how LLaMP can generate, self-correct, and execute ASE simulation code to calculate the bulk modulus of aluminum.

Generate a script and run ASE EMT calculator to relax 2x2x2 Al supercell and calculate its bulk modulus # Entering new AgentExecutor chain... To generate a script and run the ASE (Atomic Simulation Environment) EMT (Effective Medium Theory) calculator to relax a 2x2x2 Al supercell and calculate its bulk modulus, we need to write a Python script that uses ASE's functionalities. Here 's a step-by-step breakdown of what the script will do: 1. Import necessary modules from ASE. 2. Create an aluminum (Al) supercell. 3. Set up the EMT calculator. 4. Relax the structure using the EMT calculator. 5. Calculate the bulk modulus. Let's write the script: from ase import Atoms from ase.build import bulk from ase.calculators.emt import EMT from ase.constraints import StrainFilter, ExpCellFilter from ase.optimize import BFGS from ase.eos import EquationOfState # Create an Al supercell al = bulk('Al', 'fcc', a=4.05, cubic=True) al = al.repeat((2, 2, 2)) # Repeat the unit cell to create a 2x2x2 supercell # Set up the EMT calculator al.set\_calculator(EMT()) # Relax the structure strain\_filter = StrainFilter(al) optimizer = BFGS(strain\_filter) optimizer.run(fmax=0.01) # Calculate the bulk modulus

```
volumes = []
energies = []
for x in [0.98, 0.99, 1.00, 1.01, 1.02]:
 al_copy = al.copy()
 al_copy.set_calculator(EMT())
 cell = al_copy.get_cell()
 al_copy.set_cell(cell * x, scale_atoms=True)
 volumes.append(al_copy.get_volume())
 energies.append(al_copy.get_potential_energy())
eos = EquationOfState(volumes, energies)
v0, e0, B = eos.fit()
eos.plot('al-eos.png')
```

print(f'Bulk\_modulus:\_{B\_/\_kJ}\_GPa')

Now, let's execute this script using the Python\\_REPL tool.

Action: Python\\_REPL Action Input:

```
from ase import Atoms
 from ase.build import bulk
 from ase.calculators.emt import EMT
 from ase.constraints import StrainFilter, ExpCellFilter
 from ase.optimize import BFGS
 from ase.eos import EquationOfState
 from ase.units import kJ
 # Create an Al supercell
 al = bulk('Al', 'fcc', a=4.05, cubic=True)
 al = al.repeat((2, 2, 2)) # Repeat the unit cell to create a 2x2x2 supercell
 # Set up the EMT calculator
 al.set_calculator(EMT())
 # Relax the structure
 strain_filter = StrainFilter(al)
 optimizer = BFGS(strain_filter)
 optimizer.run(fmax=0.01)
 # Calculate the bulk modulus
 volumes = []
 energies = []
 for x in [0.98, 0.99, 1.00, 1.01, 1.02]:
 al_copy = al.copy()
 al_copy.set_calculator(EMT())
 cell = al_copy.get_cell()
 al_copy.set_cell(cell * x, scale_atoms=True)
 volumes.append(al_copy.get_volume())
 energies.append(al_copy.get_potential_energy())
 eos = EquationOfState(volumes, energies)
 v0, e0, B = eos.fit()
 print(f'Bulk_modulus:_{B_/_kJ}_GPa')
1592
```

1593	
1594	
1595	Step Time Energy fmax
1590	BFGS: 0 01:37:31 -0.048066 8.967521 BFGS: 1 01:37:31 1 542253 33 456077
1500	BFGS: 2 01:37:31 -0.154186 1.250352
1500	BFGS: 3 01:37:31 -0.156157 0.259261
1099	BFGS: 4 01:37:31 -0.156246 0.000522
1601	Bulk modulus: 3.923492594026645e-23 GPa
1602	
1602	The output of the bulk modulus calculation scores to be incorrect because the value
1604	is extremely small and not in the expected range for aluminum. This is likely
1605	due to a unit conversion error. The ASE package uses eV and Angstroms for
1606	energy and length, respectively, and the bulk modulus should be in units of eV/
1607	Angstrom <sup>3</sup> , which can be converted to GPa. The conversion factor from eV/
1608	Angstrom <sup>33</sup> to GPa is approximately 160.21/66208. Let's correct the script to
1609	
1610	Action: Python\_REPL
1611	Action Input:
1612	
1613	
1614	
1615	from ase import Atoms
1616	from ase calculators emt import EMT
1617	from ase.constraints import StrainFilter. ExpCellFilter
1618	from ase.optimize import BFGS
1619	<pre>from ase.eos import EquationOfState</pre>
1620	from ase.units import kJ
1621	# Create an Al supercell
1622	al = bulk('Al', 'fcc', a=4.05, cubic=True)
1623	al = al.repeat((2, 2, 2)) # Repeat the unit cell to create a 2x2x2 supercell
1624	
1625	# Set up the EMI calculator
1626	
1627	# Relax the structure
1628	<pre>strain_filter = StrainFilter(al)</pre>
1629	optimizer = BFGS(strain_filter)
1630	optimizer.run(tmax=0.01)
1631	# Calculate the bulk modulus
1632	volumes = []
1633	energies = []
1634	for x in [0.98, 0.99, 1.00, 1.01, 1.02]:
1635	al_copy = al.copy() al_copy_set_calculator(EMI())
1030	cell = al_copy.get_cell()
103/	al_copy.set_cell(cell * x, scale_atoms=True)
1030	volumes.append(al_copy.get_volume())
1640	<pre>energies.append(al_copy.get_potential_energy())</pre>
16/1	eos = EquationOfState(volumes, energies)
16/12	v0, e0, B = eos.fit()
1643	
1644	# Convert the bulk modulus from eV/Angstrom^3 to GPa
1645	
1646	<pre>print(f'Bulk_modulus:_{B_GPa}_GPa')</pre>

1648 1649 1650 1651 1652 1653 1654 1655 1656	StepTimeEnergyfmaxBFGS:0 01:37:48-0.0480668.967521BFGS:1 01:37:481.54225333.456077BFGS:2 01:37:48-0.1541861.250352BFGS:3 01:37:48-0.1561570.259261BFGS:4 01:37:48-0.1562460.000522Bulk modulus:39.23492594026644GPa
1657 1658 1659 1660 1661	Do I need to use a tool? No Final Answer: The corrected bulk modulus of the relaxed 2x2x2 Al supercell calculated using the ASE EMT calculator is approximately 39.23 GPa. # Finished chain.
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