EFFICIENT APPROXIMATIONS OF COMPLETE INTERATOMIC POTENTIALS FOR CRYSTAL PROPERTY PREDICTION

Anonymous authors
Paper under double-blind review

ABSTRACT

We study the problem of crystal material property prediction. A crystal structure consists of a minimal unit cell that is repeated infinitely in 3D space. How to accurately represent such repetitive structures in machine learning models remains unresolved. Current methods construct graphs by establishing edges only between nearby nodes, thereby failing to faithfully capture infinite repeating patterns and distant interatomic interactions. In this work, we propose several innovations to overcome these limitations. First, we propose to model physics-principled interatomic potentials directly instead of only using distances as in existing methods. These potentials include the Coulomb potential, London dispersion potential, and Pauli repulsion potential. Second, we propose to model the complete set of potentials among all atoms, instead of only between nearby atoms as in prior methods. This is enabled by our approximations of infinite potential summations with provable error bounds. We further develop efficient algorithms to compute the approximations. Finally, we propose to incorporate our computations of complete interatomic potentials into message passing neural networks for representation learning. We perform experiments on the JARVIS and Materials Project benchmarks for evaluation. Results show that the use of complete interatomic potentials leads to consistent performance improvements with reasonable computational costs.

1 INTRODUCTION

The past decade has witnessed a surge of interests and rapid developments in machine learning for molecular analysis (Duvenaud et al., 2015). These initial studies mainly focus on the prediction and generation problems of small molecules. To enable computational analyses, molecules need to be featurized in an appropriate mathematical representation form. Recently, with the advances of graph neural networks (GNNs) (Gilmer et al., 2017; Battaglia et al., 2018), molecules are more commonly represented as graphs in which each node corresponds to an atom, and each edge corresponds to a chemical bond (Stokes et al., 2020; Wang et al., 2022c). A variety of molecular graph prediction (Stokes et al., 2020; Wang et al., 2022c) and generation (Shi et al., 2019; Jin et al., 2018; Luo et al., 2021) methods have been developed based on 2D molecular graph representations. A key limitation of the 2D graph representations is that the 3D geometries of molecules are not captured, but such information may be critical in many molecular property prediction problems (Hu et al., 2021). To enable the encoding of 3D molecular geometries in GNNs, a series of 3D GNN methods have been developed for prediction (Schütt et al., 2017; Gasteiger et al., 2019; Liu et al., 2022c; Wang et al., 2022b) and generation (Liu et al., 2022a; Luo & Ji, 2022; Hoogeboom et al., 2022) problems. In these 3D graph representations, each node is associated with the corresponding atom's coordinate in 3D space. Geometric information, such as distances between nodes and angles between edges, is used during message passing in GNNs. Recently, these methods have been extended to learn representations for proteins (Jing et al., 2020; Wang et al., 2022a).

Inspired by the success of GNNs on small molecules, Xie & Grossman (2018) developed the crystal graph convolutional neural network (CGCNN) for crystal material property prediction. Different from small molecules and proteins, crystal materials are typically modeled by a minimal unit cell (similar to a small molecule) that is repeated in 3D space with certain directions and step sizes. In theory, the unit cell is repeated infinitely in 3D space, but any real-world material has finite size.

However, given that our modeling is at the atomic level, modeling crystal materials as infinite repetitions of unit cells is approximately accurate. Therefore, a key challenge in crystal material modeling is how to accurately capture the infinite-range interatomic interactions resulted from the repetitions of unit cells in 3D space. Current GNN-based crystal property prediction methods construct graphs by creating edges only between atoms within a pre-specified distance threshold (Xie & Grossman, 2018; Chen et al., 2019; Louis et al., 2020; Schmidt et al., 2021; Choudhary & DeCost, 2021). Thus, they fail to capture interactions between distant atoms explicitly.

In this work, we propose a new graph deep learning method, **PotNet**, with several innovations to significantly advance the field of crystal material modeling. First, we propose to model interatomic potentials directly as edge features in PotNet, instead of using distance as in prior methods. These potentials include the Coulomb potential (West, 1988), London dispersion potential (Wagner & Schreiner, 2015), and Pauli repulsion potential (Krane, 1991). Second, a distinguishing feature of PotNet is to model the **complete** set of potentials among all atoms, instead of only between nearby atoms as in prior methods. This is enabled by our approximations of infinite potential summations with provable error bounds. We further develop efficient algorithms to compute the approximations. Finally, we propose to incorporate our computations of complete interatomic potentials into message passing neural networks for representation learning. We performed comprehensive experiments on the JARVIS and Materials Project benchmarks to evaluate our methods. Results show that the use of complete interatomic potentials in our methods leads to consistent performance improvements with reasonable computational costs.

2 Background and Related Work

2.1 Crystal Representation and Property Prediction

A crystal structure can be represented as the periodic repetition of unit cells in the three-dimensional (3D) Euclidean space, where the unit cell contains the smallest repeatable structure of a given crystal. Specifically, let n be the number of atoms in the unit cell, a crystal can be represented as $\boldsymbol{M} = (\boldsymbol{A}, \boldsymbol{L})$. Here, $\boldsymbol{A} = \{\boldsymbol{a}_i\}_{i=1}^n = \{(\boldsymbol{x}_i, \boldsymbol{p}_i)\}_{i=1}^n$ describes one of the unit cell structures of $\boldsymbol{M}$, where $\boldsymbol{x}_i \in \mathbb{R}^b$ and $\boldsymbol{p}_i \in \mathbb{R}^3$ denote the b-dimensional feature vector and the 3D Cartesian coordinates of the i-th atom in the unit cell, respectively. $\boldsymbol{L} = [\boldsymbol{l}_1, \boldsymbol{l}_2, \boldsymbol{l}_3] \in \mathbb{R}^{3\times 3}$ is the lattice matrix describing how a unit cell repeats itself in the 3D space. In the complete crystal structure, every atom in a unit cell repeats itself periodically in the 3D space. Specifically, from an arbitrary integer vector $\boldsymbol{k} \in \mathbb{Z}^3$ and the unit cell structure $\boldsymbol{A}$, we can always obtain another repeated unit cell structure $\boldsymbol{A}^{\boldsymbol{k}} = \{\boldsymbol{a}_i^{\boldsymbol{k}}\}_{i=1}^n = \{(\boldsymbol{x}_i^{\boldsymbol{k}}, \boldsymbol{p}_i^{\boldsymbol{k}})\}_{i=1}^n$, where $\boldsymbol{x}_i^{\boldsymbol{k}} = \boldsymbol{x}_i$, $\boldsymbol{p}_i^{\boldsymbol{k}} = \boldsymbol{p}_i + \boldsymbol{L}\boldsymbol{k}$. Hence, the complete crystal structure $\widetilde{\boldsymbol{A}}$ of $\boldsymbol{M}$ with all unit cells can be described as

$$\widetilde{\boldsymbol{A}} = \bigcup_{\boldsymbol{k}\in\mathbb{Z}^3} \boldsymbol{A}^{\boldsymbol{k}}. \tag{1}$$

In this work, we study the problem of crystal property prediction. Our objective is to learn a property prediction model $f : \boldsymbol{M} \to y \in \mathbb{R}$ that can predict the property y of the given crystal structure $\boldsymbol{M}$. We will focus on predicting the total energy, or other energy-related properties of crystals.

2.2 Crystal Property Prediction with Interatomic Potentials

Most of the classical crystal energy prediction methods are based on interatomic potentials. According to the studies in physics (West, 1988; Daw et al., 1993; Brown, 2016), the total energy of a crystal structure can be approximated by the summation of interatomic potentials in the crystal. Particularly, the three following categories of interatomic potentials are widely used in crystals, and they are considered as sufficient for accurate energy approximation.

- **Coulomb potential**, which is caused by the electrostatic interaction of two atoms with charges. Coulomb potentials are closely related to ionic bonding and metallic bonding in crystals (West, 1988). For any two atoms $\boldsymbol{a}$ and $\boldsymbol{b}$, let $z_{\boldsymbol{a}}$ and $z_{\boldsymbol{b}}$ denote the number of charges in the atom $\boldsymbol{a}$ and $\boldsymbol{b}$, and let $d(\boldsymbol{a}, \boldsymbol{b})$ be the Euclidean distance between the atom $\boldsymbol{a}$ and $\boldsymbol{b}$. The Coulomb potential V_{Coulomb} is defined as $V_{\text{Coulomb}}(\boldsymbol{a}, \boldsymbol{b}) = -\frac{z_{\boldsymbol{a}} z_{\boldsymbol{b}} e^2}{4\pi\epsilon_0 d(\boldsymbol{a},\boldsymbol{b})}$. Here e is the elementary charge constant, and ϵ_0 is the permittivity constant of free space.

- **London dispersion potential**, which describes the Van der Waals interaction between atoms. London dispersion potential is often considered in energy estimation because its contribution is cumulative over the volume of crystals (Wagner & Schreiner, 2015) and sometimes very strong in bulk crystals, such as the crystals containing sulfur and phosphorus. The mathematical form of this potential can be described as $V_{\text{London}}(\boldsymbol{a}, \boldsymbol{b}) = -\epsilon/d^6(\boldsymbol{a}, \boldsymbol{b})$, where ϵ is a hyperparameter.
- **Pauli repulsion potential**, which results from the generally existing Pauli exclusion principle in all crystal structures. The Pauli exclusion principle forces any two atoms to be sufficiently far away from each other so that the electron orbits of them do not overlap. Such exclusion interactions lead to Pauli repulsion potential with the form of $V_{\text{Pauli}}(\boldsymbol{a}, \boldsymbol{b}) = e^{-\alpha d(\boldsymbol{a}, \boldsymbol{b})}$, where α is a hyperparameter (Buckingham, 1938; Slater, 1928).

2.3 Crystal Property Prediction with Deep Learning

While physics-based methods have been used for predicting the crystal energy for a long time, these methods are usually crystal-specific, i.e., one method can only achieve accurate approximation for one certain type of crystals. Recently, thanks to the advances of deep learning, many studies have been done to develop a general crystal property predictor for a variety of different crystals with powerful deep neural network models. Some studies (Wang et al., 2021; Jha et al., 2018; 2019; Goodall & Lee, 2020) represent crystals as chemical formulas, and adopt sequence models to predict properties from these string representations. However, more recent studies consider crystals as 3D graphs and employ expressive 3D GNNs (Schütt et al., 2017; Klicpera et al., 2020b; Liu et al., 2022b), a family of deep neural networks specifically designed for 3D graph-structured data, to crystal representation learning. CGCNN (Xie & Grossman, 2018) is the first method that proposes to represent crystals with radius graphs and adopts a graph convolutional network to predict the property from the graph. Based on the pioneering exploration of CGCNN, a lot of subsequent studies (Schmidt et al., 2021; Louis et al., 2020; Chen et al., 2019; Choudhary & DeCost, 2021; Batzner et al., 2022) propose various 3D GNN architectures to achieve more effective crystal representation learning. Particularly, by enhancing the input features with angle information, ALIGNN Choudhary & DeCost (2021) develops the currently most powerful 3D GNN architecture for crystals and achieves the best crystal property prediction performance.

3 Method

Although existing GNN-based methods have achieved impressive performance in crystal property prediction, they struggle in further boosting the performance due to the approximation of interatomic interactions using functional expansions based on distances and failure in capturing complete interatomic interactions. In this section, we present PotNet, a novel crystal representation model that can overcome these limitations of prior methods. Based on the physical modeling of crystal energy, PotNet explicitly uses infinite potential summations as input features to capture complete interatomic interactions. The infinite potential summations are incorporated into the message passing mechanism of graph neural networks and efficiently approximated by a fast algorithm. To the best of our knowledge, PotNet is the first work that bridges the classical crystal energy computation methods based on potentials and the data-driven methods based on deep neural networks.

3.1 Approximating Crystal Energy with Complete Interatomic Potentials

According to the density functional theory (DFT) in physics, for any crystal $\boldsymbol{M} = (\boldsymbol{A}, \boldsymbol{L})$ with the complete structure $\widetilde{\boldsymbol{A}}$ defined in Eqn. (1), its total energy $E(\boldsymbol{M})$ can be accurately approximated by the embedded atom method (Daw & Baskes, 1984; Daw et al., 1993; Baskes, 1987; Lee et al., 2016; Riffe et al., 2018) in the form of

$$E(\boldsymbol{M}) = \frac{1}{2} \sum_{\boldsymbol{a} \in \boldsymbol{A}} \sum_{\boldsymbol{b} \neq \boldsymbol{a}, \boldsymbol{b} \in \widetilde{\boldsymbol{A}}} V(\boldsymbol{a}, \boldsymbol{b}) + \sum_{\boldsymbol{a} \in \boldsymbol{A}} F(\rho_{\boldsymbol{a}}), \tag{2}$$

where $V(\boldsymbol{a}, \boldsymbol{b})$ denotes the interatomic potentials between the atoms $\boldsymbol{a}$ and $\boldsymbol{b}$, capturing the magnitude of interactions; $\rho_{\boldsymbol{a}}$ is the local electron density of the atom $\boldsymbol{a}$, determined by the coordinate and number of charges of the atom $\boldsymbol{a}$ according to the Hohenberg-Kohn theorem; $F(\cdot)$ is a nonlinear function to embed the electron density $\rho_{\boldsymbol{a}}$. Actually, existing studies (Jalkanen & Müser, 2015)

show that $\rho_{\boldsymbol{a}}$ can be considered as a function of $\sum_{\boldsymbol{b}\neq\boldsymbol{a},\boldsymbol{b}\in\widetilde{\boldsymbol{A}}} V(\boldsymbol{a},\boldsymbol{b})$ mathematically. Hence, the Eqn. (2) can be rewritten in the following form as

$$E(\boldsymbol{M}) = \sum_{\boldsymbol{a}\in\boldsymbol{A}} \left[\frac{1}{2} \sum_{\boldsymbol{b}\neq\boldsymbol{a},\boldsymbol{b}\in\widetilde{\boldsymbol{A}}} V(\boldsymbol{a},\boldsymbol{b}) + G\left(\sum_{\boldsymbol{b}\neq\boldsymbol{a},\boldsymbol{b}\in\widetilde{\boldsymbol{A}}} V(\boldsymbol{a},\boldsymbol{b}) \right) \right], \tag{3}$$

where $G(\cdot)$ is a nonlinear function. Eqn. (3) can be considered as a way to compute the energy from the complete interatomic potential summation $\sum_{\boldsymbol{b}\neq\boldsymbol{a},\boldsymbol{b}\in\widetilde{\boldsymbol{A}}} V(\boldsymbol{a},\boldsymbol{b})$ of every atom $\boldsymbol{a}$ in the unit cell $\boldsymbol{A}$. However, in practice, the function G is computationally expensive if not infeasible. Hence, more and more studies have turned to the powerful learning capability of modern deep neural network models to effectively approximate it.

3.2 Limitations of Existing Deep Learning Methods

Currently, most of the existing graph deep learning methods (Xie & Grossman, 2018; Chen et al., 2019; Louis et al., 2020; Choudhary & DeCost, 2021) for crystals takes the radius graph representation and the distance-based features as inputs to predict the crystal energy in Eqn. (3). Specifically, for a crystal $\boldsymbol{M} = (\boldsymbol{A}, \boldsymbol{L})$, the radius graph is constructed by adding edges between any atom $\boldsymbol{a}$ in the unit cell $\boldsymbol{A}$ and any other atom $\boldsymbol{b}$ in the complete crystal structure $\widetilde{\boldsymbol{A}}$ whose distances are smaller than a pre-specified distance threshold r. In addition, some functional expansions of distances, e.g., radial basis functions (RBF), are used to model interatomic interactions and form the input edge features to the 3D GNN model. Hence, let $\boldsymbol{a} = (\boldsymbol{x}_{\boldsymbol{a}}, \boldsymbol{p}_{\boldsymbol{a}}), \boldsymbol{b} = (\boldsymbol{x}_{\boldsymbol{b}}, \boldsymbol{p}_{\boldsymbol{b}})$, the crystal energy prediction $\hat{E}(\boldsymbol{M})$ of these methods can be generally described as

$$\hat{E}(\boldsymbol{M}) = \sum_{\boldsymbol{a}\in\boldsymbol{A}} \sum_{\boldsymbol{b}\in\mathcal{N}_r(\boldsymbol{a})} H\left(\phi\left(||\boldsymbol{p}_{\boldsymbol{a}} - \boldsymbol{p}_{\boldsymbol{b}}||_2\right)\right), \tag{4}$$

where $\mathcal{N}_r(\boldsymbol{a}) = \{\boldsymbol{b} : \boldsymbol{b} \neq \boldsymbol{a}, \boldsymbol{b} \in \widetilde{\boldsymbol{A}}, ||\boldsymbol{p}_{\boldsymbol{a}} - \boldsymbol{p}_{\boldsymbol{b}}||_2 < r\}$, $\phi(\cdot)$ denotes the functional expansions, and $H(\cdot)$ is a non-linear function based on 3D GNN models.

However, we argue that predicting or approximating the energy with Eqn. (4) is a suboptimal solution. Actually, compared with Eqn. (3) that is physics-principled, there exist non-negligible approximation errors in Eqn. (4). First, Eqn. (4) captures the interatomic interactions based on interatomic distances, while the energy can be more accurately approximated by a function of interatomic potentials as in Eqn. (3). Though according to Sec. 2.2, interatomic potentials themselves are also functions of distances, we argue that directly using functional expansions of distances is not the best solution to crystal energy prediction. The commonly used functional expansions in existing methods, such as RBF $\phi(\cdot)$, have different mathematical forms from potentials defined in Sec. 2.2. Intuitively, this will pose more challenges and burdens to 3D GNN models since they need to learn a mapping from $\phi(\cdot)$ to the energy $\boldsymbol{E}$ while the energy $\boldsymbol{E}$ is not the direct function of $\phi(\cdot)$. Hence, we argue that directly employing the physics-principled potential functions instead of $\phi(\cdot)$ as input features is more suitable for crystal energy prediction.

Second, different from Eqn. (3), Eqn. (4) does not capture the complete set of interatomic interactions because the summation set $\mathcal{N}_r(\boldsymbol{a})$ of atoms $\boldsymbol{b}$ is constrained to be the atoms whose distances to the atom $\boldsymbol{a}$ are smaller than r. This can lead to a significant approximation error due to ignoring long-range interatomic potentials, i.e., interatomic potentials between distant atoms. Different from molecules with finite structures, long-range interatomic potentials cannot be ignored for crystals with infinite structures. By the first principles in physics, interatomic potentials decay algebraically when pairwise interatomic distances become larger. Hence, for a finite structure like molecules, the potentials from atoms that are far away from a given atom is limited and can be ignored. However, long-range interatomic potentials have significant influence on a given atom in the infinite crystal structure. Taking Coulomb potentials as an example, assuming that we are given a 1D crystal structure where there is only one atom repeating itself with Euclidean distance of 1, each atom has only one unit of charge, and the total energy is simply the sum of all interatomic Coulomb potentials. As defined in Sec. 2.2, Coulomb potential energy $V(\boldsymbol{a},\boldsymbol{b})$ between atoms $\boldsymbol{a}$ and $\boldsymbol{b}$ satisfies $V(\boldsymbol{a},\boldsymbol{b}) \propto 1/d$, where d is the distance between atoms $\boldsymbol{a}$ and $\boldsymbol{b}$. Considering the Coulomb potentials between a given atom and all other atoms, the total potential $\widetilde{V}$ of them satisfies $\widetilde{V} \propto \sum_{n=1}^{\infty} 1/n$. If only the pairwise atoms within the distance threshold r are considered,

an infinite error of energy calculation is introduced. Specifically, the smallest possible prediction error ΔV satisfies $\Delta V \propto \sum_{n=\lfloor r+1 \rfloor}^{\infty} 1/n$, which is infinite. In other words, ignoring interatomic Coulomb potentials between atoms with distances larger than r will cause a significant prediction error of the total energy. From this example, we can easily find that the failure to capture complete interatomic potentials due to using radius graphs is a key factor that prevents accurate energy prediction in existing GNN-based methods.

3.3 Message Passing with Complete Interatomic Potentials

It follows from the analysis in Sec. 3.2 that the major limitations of existing deep learning methods for crystal representation learning lie in (1) not making the prediction from physics-principled interatomic potentials, and (2) not considering complete interatomic interactions. To overcome these limitations, we propose to explicitly use complete interatomic potential summations in GNN models. Since our proposed method is tightly related to potentials, we name it PotNet.

By reformulating Eqn. (3), our PotNet incorporates the crystal energy computation with complete interatomic potentials into the message passing scheme of GNN models. For any material structure $\boldsymbol{M} = (\boldsymbol{A}, \boldsymbol{L})$, we can rewrite the definition of its complete structure $\widetilde{\boldsymbol{A}}$ in Eqn. (1) as

$$\widetilde{\boldsymbol{A}} = \bigcup_{\boldsymbol{k} \in \mathbb{Z}^3} \boldsymbol{A}^{\boldsymbol{k}} = \bigcup_{\boldsymbol{k} \in \mathbb{Z}^3} \bigcup_{\boldsymbol{b} \in \boldsymbol{A}} \{\boldsymbol{b}^{\boldsymbol{k}}\} = \bigcup_{\boldsymbol{b} \in \boldsymbol{A}} \bigcup_{\boldsymbol{k} \in \mathbb{Z}^3} \{\boldsymbol{b}^{\boldsymbol{k}}\} = \bigcup_{\boldsymbol{b} \in \boldsymbol{A}} \boldsymbol{A}_{\boldsymbol{b}}, \tag{5}$$

where $\boldsymbol{A}_{\boldsymbol{b}} = \bigcup_{\boldsymbol{k} \in \mathbb{Z}^3} \{\boldsymbol{b}^{\boldsymbol{k}}\}$ denotes the set of atoms containing the atom $\boldsymbol{b}$ from the unit cell $\boldsymbol{A}$ and all its periodically repeated duplicates in the complete crystal structure. With Eqn. (5), we can reformulate Eqn. (3) as

$$\begin{aligned} E(\boldsymbol{M}) &= \sum_{\boldsymbol{a} \in \boldsymbol{A}} \left[\frac{1}{2} \sum_{\boldsymbol{b} \in \boldsymbol{A}} \sum_{\boldsymbol{c} \neq \boldsymbol{a}, \boldsymbol{c} \in \boldsymbol{A}_{\boldsymbol{b}}} V(\boldsymbol{a}, \boldsymbol{c}) + G\left(\sum_{\boldsymbol{b} \in \boldsymbol{A}} \sum_{\boldsymbol{c} \neq \boldsymbol{a}, \boldsymbol{c} \in \boldsymbol{A}_{\boldsymbol{b}}} V(\boldsymbol{a}, \boldsymbol{c}) \right) \right] \\ &= \sum_{\boldsymbol{a} \in \boldsymbol{A}} \left[\frac{1}{2} \sum_{\boldsymbol{b} \in \boldsymbol{A}} S(\boldsymbol{a}, \boldsymbol{b}) + G\left(\sum_{\boldsymbol{b} \in \boldsymbol{A}} S(\boldsymbol{a}, \boldsymbol{b}) \right) \right], \end{aligned} \tag{6}$$

where the infinite potential summation $S(\boldsymbol{a}, \boldsymbol{b}) = \sum_{\boldsymbol{c} \neq \boldsymbol{a}, \boldsymbol{c} \in \boldsymbol{A}_{\boldsymbol{b}}} V(\boldsymbol{a}, \boldsymbol{c})$ denotes the sum of the interatomic potentials from the atom $\boldsymbol{b}$ together with its all periodic duplicates to the atom $\boldsymbol{a}$. Eqn. (6) can be integrated into the message passing scheme of GNN models. Specifically, we can create a graph $\boldsymbol{G}$ for $\boldsymbol{M} = (\boldsymbol{A}, \boldsymbol{L})$, where each atom in the unit cell $\boldsymbol{A}$ corresponds to a node in the graph. For any two nodes $\boldsymbol{u}, \boldsymbol{v}$ in the graph, there is an edge connecting them, and every node $\boldsymbol{u}$ in the graph is also connected to itself by a self-loop edge. If we consider the infinite potential summation $S(\boldsymbol{a}, \boldsymbol{b})$ as the feature of the edge from the node $\boldsymbol{b}$ to $\boldsymbol{a}$, we can use the message passing based non-linear neural network model in GNN to fit the non-linear function $\frac{1}{2} \sum_{\boldsymbol{b} \in \boldsymbol{A}} S(\boldsymbol{a}, \boldsymbol{b}) + G\left(\sum_{\boldsymbol{b} \in \boldsymbol{A}} S(\boldsymbol{a}, \boldsymbol{b})\right)$.

Based on this design of directly using interatomic potentials as edge features, our PotNet employs a GNN model with multiple message passing layers on the graph $\boldsymbol{G}$ to predict the crystal energy of $\boldsymbol{M}$. The computational process of the ℓ-th message passing layer for the node $\boldsymbol{a}$ can be described as

$$\boldsymbol{h}_{\boldsymbol{a}}^{(\ell)} = g_{\varphi}\left(\boldsymbol{h}_{\boldsymbol{a}}^{(\ell-1)}, \sum_{\boldsymbol{b} \in \boldsymbol{A}} f_{\theta}\left(\boldsymbol{h}_{\boldsymbol{a}}^{(\ell-1)}, \boldsymbol{h}_{\boldsymbol{b}}^{(\ell-1)}, \sum_{\boldsymbol{b} \in \boldsymbol{A}} S(\boldsymbol{a}, \boldsymbol{b}) \right) \right), \tag{7}$$

where $\boldsymbol{h}_{\boldsymbol{a}}^{(\ell)}$ denotes the node embedding vector of the node $\boldsymbol{a}$ outputted from the ℓ-th message passing layer, $\boldsymbol{h}_{\boldsymbol{a}}^{(0)}$ is initialized to the atom feature vector of the atom $\boldsymbol{a}$, and $g_{\varphi}(\cdot), f_{\theta}(\cdot)$ are both neural network models with the trainable parameters φ and θ, respectively. Here, the model f_{θ} plays the role of capturing information from both atomic features and complete interatomic potentials. Detailed information about model architectures of f_{θ} and g_{φ} is provided in Appendix D. Note that our PotNet is actually a 3D GNN model even though 3D geometric information is not explicitly involved in Eqn. (7). This is because the edge feature $S(\boldsymbol{a}, \boldsymbol{b})$ is related to potential functions, and by Sec. 2.2 we know that they are computed from interatomic distances. In other words, PotNet can be considered to encode 3D geometric information with potential functions, though our direct motivation of using potential functions comes from the physical modeling of crystal energy.

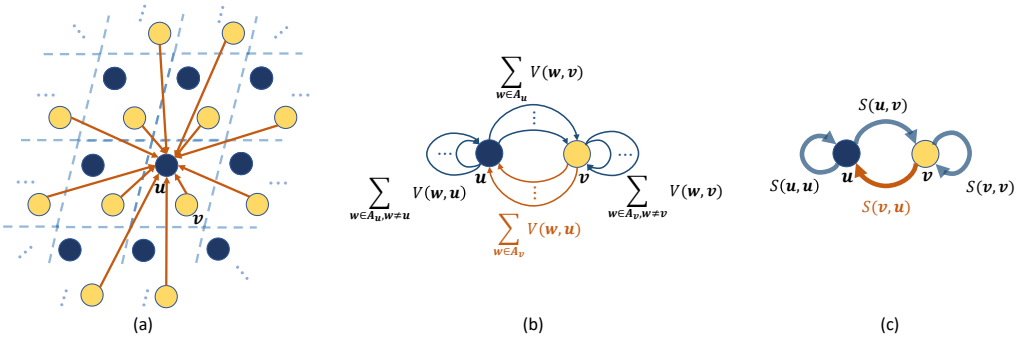

Figure 1: Schematic illustrations of how complete interatomic itneractions are captured in PotNet. Note that PotNet models 3D crystals while we have 2D illustration for simplicity. (a) An example crystal in which each unit cell contains two atoms u and v. In PotNet, the potentials between all pairs of atoms are captured. For simplicity, we only show the potentials from all v atoms to a u atom. (b) The complete set of potentials in (a) can be grouped into four categories, including $u \rightarrow v$, $v \rightarrow u$, $u \rightarrow u$, and $v \rightarrow v$. (c) We propose to compute an approximate summation for each category of potentials.

Intuitively, the message passing process in Eqn. (7) over the graph $\boldsymbol{G}$ can be considered as a general case of employing a radius graph where the distance threshold r goes to infinity, i.e., $r \rightarrow +\infty$. In this case, as shown in Fig. 1(a), for any atom in the crystal, all the other atoms in the complete crystal structure will be included to interact with it. If we follow the radius graph construction process in previous methods (Xie & Grossman, 2018; Chen et al., 2019; Louis et al., 2020; Choudhary & DeCost, 2021), we will obtain a graph $\widetilde{\boldsymbol{G}}$ in which there exist an infinite number of edges between every pair of nodes. However, PotNet simplifies this complicated graph $\widetilde{\boldsymbol{G}}$ to the graph $\boldsymbol{G}$ in which only one edge exists between every node pair. Specifically, PotNet directly models interatomic interactions as potentials and for any two nodes in $\widetilde{\boldsymbol{G}}$, PotNet aggregates all edges between them to a single edge by the use of infinite potential summation $S(\boldsymbol{a}, \boldsymbol{b})$ (see Fig. 1(b)). In other words, PotNet provides an effective solution that enables GNN models to capture complete interatomic interactions through the use of infinite potential summations.

For our implementation of infinite potential summations $S(\boldsymbol{a}, \boldsymbol{b})$ in Eqn. (7), we add up the infinite summations of three potentials, including the Coulomb potentials, London dispersion potentials, and Pauli repulsion potentials described in Sec. 2.2. We simplify the mathematical form of Coulomb potentials $V_{\text{Coulomb}}(\boldsymbol{a}, \boldsymbol{b}) = -\frac{z_{\boldsymbol{a}} z_{\boldsymbol{b}} e^2}{4\pi\epsilon_0 d(\boldsymbol{a}, \boldsymbol{b})}$ to $V_{\text{Coulomb}}(\boldsymbol{a}, \boldsymbol{b}) = -\epsilon_1 / d(\boldsymbol{a}, \boldsymbol{b})$ where ϵ_1 is a hyperparameter, because e, π, ϵ_0 are all known constants, and $z_{\boldsymbol{a}}, z_{\boldsymbol{b}}$ can be learned from atomic features. The detailed hyperparameter settings of London dispersion potentials $V_{\text{London}}(\boldsymbol{a}, \boldsymbol{b}) = -\epsilon / d^6(\boldsymbol{a}, \boldsymbol{b})$ and Pauli dispersion potentials $V_{\text{Pauli}} = e^{-\alpha d(\boldsymbol{a}, \boldsymbol{b})}$ can be found in Appendix D.

3.4 Efficient Computation of Infinite Potential Summation

Although we have effectively incorporated infinite potential summations into the message passing based GNN models, the computation of these infinite potential summations is not easy. Basically, there are two challenges to achieve accurate and efficient computation of the infinite potential summations. For accuracy, the computation algorithm needs to have provable error bounds. For efficiency, it needs to be fast for scalable GNN training and fast crystal property prediction. To tackle these two challenges, we derive a fast approximation algorithm for infinite potential summations based on the Ewald summation method (Ewald, 1921). Specifically, we first unify the summations of three infinite potentials between atom $\boldsymbol{a}$ and all duplicated positions of atom $\boldsymbol{b}$ into an integral form such that the Ewald summation method can be applied for efficient implementation in PotNet (Fig. 1(c)). Based on that, the Ewald summation method is used with the infinite summations split into two parts, including one part that converges fast in Fourier space and another part that converges fast in Euclidean space, to obtain fast approximations with provable error bounds.

To apply the fast approximation algorithm of infinite summations proposed by Ewald (1921), a unified integral view of infinite potential summations is needed. Following notations in Sec. 2 and 3.3, we denote the positions of the atoms in the set $\boldsymbol{A_b}$ as $\boldsymbol{P_b} = \{\boldsymbol{p_b^k} | \boldsymbol{p_b^k} = \boldsymbol{p_b} + \boldsymbol{Lk}, \boldsymbol{k} \in$

Table 1: Comparison between PotNet and other baselines in terms of test MAE on the Materials Project dataset. To make the comparison clear and fair, we retrain baseline methods using the same dataset settings. We also show results from original papers in parentheses and mark missing results as -. The best results are shown in **bold** and the second best results are shown with underlines.

Method	Formation Energy	Band Gap	Bulk Moduli	Shear Moduli
CGCNN	0.031 (0.039)	0.292 (0.388)	0.047 (0.054)	0.077 (0.087)
SchNet	0.033 (0.035)	0.345 (-)	0.066 (-)	0.099 (-)
MEGNET	0.030 (0.028)	0.307 (0.33)	0.051 (0.050)	0.088 (0.079)
GATGNN	0.033 (0.039)	0.280 (0.31)	0.045	0.075
ALIGNN	0.0221	0.0218	0.051 (-)	0.078 (-)
PotNet	**0.0196**	**0.0204**	**0.042**	**0.069**

$\mathbb{Z}^3\}$. The Euclidean distances between the atom $\boldsymbol{a}$ and all atoms in $\boldsymbol{A_b}$ can be represented as $\{d|d = \|\boldsymbol{p_b} + \boldsymbol{Lk} - \boldsymbol{p_a}\|, \boldsymbol{k} \in \mathbb{Z}^3\}$. As described in Sec. 3.3, we simplify the Coulomb potential function as $V_{\text{Coulomb}}(\boldsymbol{a}, \boldsymbol{b}) = -\epsilon_1/d(\boldsymbol{a}, \boldsymbol{b})$, where ϵ_1 is a hyperparameter scaling the potential. Based on that, we can represent Coulomb potentials from all atoms in $\boldsymbol{A_b}$ to the atom $\boldsymbol{a}$ as $\{-\frac{\epsilon}{d}|d = \|\boldsymbol{p_b} + \boldsymbol{Lk} - \boldsymbol{p_a}\|, \boldsymbol{k} \in \mathbb{Z}^3\}$. Similarly, London dispersion potentials from all atoms in $\boldsymbol{A_b}$ to the atom $\boldsymbol{a}$ can be represented as $\{-\frac{\epsilon'}{d^6}|d = \|\boldsymbol{p_b} + \boldsymbol{Lk} - \boldsymbol{p_a}\|, \boldsymbol{k} \in \mathbb{Z}^3\}$. It is worth noting that Coulomb potentials and London dispersion potentials can be represented in a unified view as $\{\frac{\text{constant}}{d^p}|d = \|\boldsymbol{v_{ab}} + \boldsymbol{Lk}\|, \boldsymbol{v_{ab}} = \boldsymbol{p_b} - \boldsymbol{p_a}, \boldsymbol{k} \in \mathbb{Z}^3\}$, where p is a positive number. And we represent Pauli potentials from all positions of atom j to atom i as $\{e^{-\alpha d}|d = \|\boldsymbol{p_b} + \boldsymbol{Lk} - \boldsymbol{p_a}\|, \boldsymbol{k} \in \mathbb{Z}^3\}$. We then provide detailed proofs in Appendix C.1 that the summations of these three potentials can be unified in an integral form as

$$\mathcal{S}(\boldsymbol{L}, \boldsymbol{v_{ab}}) = D \sum_{\boldsymbol{k} \in \mathbb{Z}^3} \int_0^\infty t^{C-1} e^{-A\pi|\boldsymbol{Lk}+\boldsymbol{v_{ab}}|^2 t - \frac{B}{t}} dt, \tag{8}$$

where A, B, C, D are constants derived from the corresponding specific potential forms.

We then apply the Ewald summation method (Ewald, 1921) to Eqn. (8) and split it into two parts:

$$\begin{aligned}\mathcal{S}(\boldsymbol{L}, \boldsymbol{v_{ab}}) =& D \sum_{\boldsymbol{k} \in \mathbb{Z}^3} \int_0^1 t^{C-1} e^{-A\pi|\boldsymbol{Lk}+\boldsymbol{v_{ab}}|^2 t - \frac{B}{t}} dt + D \sum_{\boldsymbol{k} \in \mathbb{Z}^3} \int_1^\infty t^{C-1} e^{-A\pi|\boldsymbol{Lk}+\boldsymbol{v_{ab}}|^2 t - \frac{B}{t}} dt \\ =& \mathcal{S}_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v_{ab}}) + \mathcal{S}_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v_{ab}}),\end{aligned} \tag{9}$$

where S_{direct} denotes the part that converges fast in Euclidean space, and S_{Fourier} denotes the other part that quickly converges in Fourier space when the total summation converges as shown by Ewald (1921). Based on this, we further prove that S_{direct} and S_{Fourier} can be deduced into summations of incomplete Bessel functions $K_\nu(x, y)$ in Appendix C.2 and the approximation error can be analyzed and proved to be bounded. It is worth noting that for London dispersion potentials and Pauli potentials, the transformed summations of incomplete Bessel functions can be directly approximated. However, for Coulomb potentials, the direct summation diverges. Concretely, when $p > 3$ for potentials of form $\{\frac{\text{constant}}{d^p}|d = \|\boldsymbol{v_{ab}} + \boldsymbol{Lk}\|, \boldsymbol{v_{ab}} = \boldsymbol{p_b} - \boldsymbol{p_a}, \boldsymbol{k} \in \mathbb{Z}^3\}$, the corresponding potential summation converges, and when $p \leq 3$, the corresponding potential summation diverges. To tackle this problem, we follow previous mathematical derivations (Harris, 2008; Slevinsky & Safouhi, 2010; Jones, 2007) and use the analytically continued incomplete Bessel functions to approximate infinite summations of Coulomb potentials as shown in Appendix C.3. We then provide detailed mathematical proofs that the summations of incomplete Bessel functions are convergent and can be approximated with an error bounded by the Gaussian Lattice Sum in Appendix B.2. Detailed implementation of the proposed summation algorithm can be found in Appendix C.4.

4 Experimental Studies

4.1 Experimental Setup

We conduct experiments on two material benchmark datasets, including The Materials Project and JARVIS. Baseline methods include CFID (Choudhary et al., 2018), SchNet (Schütt et al., 2017),

Table 2: Comparison between PotNet and other baselines in terms of test MAE on JARVIS dataset. The best results are shown in **bold** and the second best results are shown with underlines.

Method	Formation Energy	Bandgap(OPT)	Total energy	Bandgap(MBJ)	Ehull
CFID	0.14	0.30	0.24	0.53	0.22
CGCNN	0.063	0.20	0.078	0.41	0.17
SchNet	0.045	0.19	0.047	0.43	0.14
MEGNET	0.047	0.145	0.058	0.34	0.084
GATGNN	0.047	0.17	0.056	0.51	0.12
ALIGNN	0.0331	0.142	0.037	0.31	0.076
PotNet	**0.0308**	**0.136**	**0.034**	**0.28**	**0.050**

CGCNN (Xie & Grossman, 2018), MEGNET (Chen et al., 2019), GATGNN (Louis et al., 2020), and ALIGNN (Choudhary & DeCost, 2021). Unless otherwise specified, we report the results reported by referred papers or provided by original authors. All PotNet models are trained using the Adam (Kingma & Ba, 2014) optimizer with weight decay (Loshchilov & Hutter, 2017) and one cycle learning rate scheduler (Smith & Topin, 2019) with learning rate of 0.001, training epoch of 500, and batch size of 64. We use Pytorch to implement our models. For all tasks on two benchmark datasets, we use one NVIDIA RTX A6000 48G GPU for computing. Other detailed configurations of PotNet for different tasks are provided in Appendix D.

To capture the global infinite-range interactions without losing details of local interactions, PotNet uses both local and infinite crystal graphs. Specifically, for the local crystal graph, we use the radius crystal graph proposed by CGCNN but replace Euclidean distances with interatomic potentials for edge features. To be concrete, because the influences of London dispersion potentials and Pauli potentials are limited and can be ignored when only considering nearby regions, we only use the Coulomb potentials in the radius crystal graph. The infinite crystal graph is constructed as described in Sec. 3.3, where Coulomb potentials, London dispersion potentials and Pauli potentials are used as edge features.

4.2 Experimental Results

The Materials Project. We first evaluate PotNet on The Materials Project-2018.6.1, which is a widely used large-scale material benchmark with 69239 crystals. We follow previous works (Xie & Grossman, 2018; Chen et al., 2019; Choudhary & DeCost, 2021; Louis et al., 2020) and use four crystal properties including formation energy, band gap, bulk moduli, and shear moduli. We notice that previous works (Xie & Grossman, 2018; Chen et al., 2019; Choudhary & DeCost, 2021; Louis et al., 2020; Schütt et al., 2017) compare with each other using different versions of splitting training, evaluation and testing datasets with different random seeds. For instance, *the original CGCNN paper only uses 28046 training samples for formation energy prediction, resulting in the original result of 0.039 as shown in Table 1.* To make the comparisons fair, we follow the settings of the previous state-of-the-art (SOTA) ALIGNN (Choudhary & DeCost, 2021) for tasks including formation energy and band gap prediction, and retrain all other baselines using the same dataset setting for these two tasks. For bulk and shear moduli, we follow the dataset setting of GATGNN (Louis et al., 2020) which has the best prediction performances for these two tasks and retrained all other baselines. We use test mean-absolute-error (MAE) as the evaluation metric following previous studies.

From Table 1, PotNet consistently outperforms other SOTA methods on all four tasks by significant margins. Specifically, for the task of formation energy, PotNet outperforms the previous SOTA ALIGNN by 11.3% and is the first method that achieves test MAE lower than 0.02. Furthermore, the impressive results of PotNet for Bulk Moduli and Shear Moduli tasks with only 4664 training samples demonstrate the robustness and adaptive ability of PotNet to the tasks with small training data.

JARVIS Dataset. We then evaluate PotNet on JARVIS, a newly released benchmark dataset proposed by Choudhary et al. (2020). We evaluate PotNet on five important crystal property prediction tasks including formation energy, bandgap (OPT), bandgap (MBJ), total energy, and Ehull. We follow ALIGNN and use the same training, validation, and test sets for these tasks. Because there are missing results for baseline methods in their original papers for JARVIS, we retrain corresponding baseline methods following the same settings.

As shown in Table 2, PotNet achieves the best performances on all the five tasks consistently by significant margins. Compared with the previous SOTA ALIGNN, PotNet has better prediction performances by explicitly capturing infinite interactions between atoms for crystal structures. Specifically, PotNet outperforms ALIGNN by 7.5%, 4.2%, 8.1%, 9.7%, and 34% for these five tasks, respectively. Overall, the superior performances of PotNet show the effectiveness of interaction modeling using potentials and explicit modeling of infinite interactions for crystal structures.

Table 3: Model complexity and runtime compared with ALIGNN on JARVIS formation energy.

Method	#Params	Time/Epoch	Total Training Time	Inference
ALIGNN	15.4 MB	327 s	27.3 h	156 s
PotNet	7.9MB	48 s	6.7 h	41 s

Efficiency of PotNet. Beyond the superior modeling capacity for crystals, our PotNet is faster and more efficient compared with ALIGNN. To demonstrate the efficiency of PotNet, we compare PotNet with ALIGNN in terms of training time per epoch, total training time, and inference time for the task of JARVIS formation energy prediction. From Table 3, PotNet is four times faster in terms of total training time and inference time compared with ALIGNN. We also analyze the time cost of infinite summations introduced in Sec. 3.3. It can be seen from Table 4 that PotNet takes 603 seconds to process the whole JARVIS dataset and ALIGNN takes 152 seconds. The introduced computational cost of infinite summations is reasonable.

Table 4: Introduced computational cost of infinite potential summations for the whole JARVIS dataset.

Method	Time per Atom	Time per Crystal	Total Calculation Time
ALIGNN	0.27 ms	2.7 ms	152 s
PotNet	1.07 ms	10.8 ms	603 s

4.3 Ablation Studies

In this section, we demonstrate the importance of two core components of PotNet, including interaction modeling using potentials and infinite summation of potentials for crystal prediction. We conduct experiments on the JARVIS formation energy task, and use test MAE as the evaluation metric.

Interaction Modeling using Potentials. We demonstrate the importance of interaction modeling using potentials by replacing potentials with Euclidean distances used by previous works in our PotNet with exactly the same model architecture. Specifically, we denote PotNet with only local crystal graph as the base model. We use 'Base + Euclidean' to represent the base model with Euclidean distances as edge features and 'Base + Potential' to represent the base model using Coulomb potentials as edge features. As shown in Table 5, by replacing Euclidean distances with Coulomb potentials, PotNet without considering infinite potential summation already obtains a significant performance gain from 0.0363 to 0.0318, revealing the importance of interaction modeling using potentials in PotNet.

Table 5: Ablation studies for the effects of adding Coulomb potentials and infinite summation.

Method	JARVIS Formation Energy
Base + Euclidean	0.0363
Base + Potential	0.0318
Base + Potential + Infinite	0.0308

Infinite Summation of Potentials. The importance of infinite summation of potentials is demonstrated by comparing the previous base models with 'Base + Potential + Infinite', denoting the full PotNet model with infinite summation in infinite crystal graph. It can be seen from Table 5 that by using infinite crystal graphs introduced in Sec. 3.3, the global repeating patterns of crystal structures are captured, resulting in a performance gain from 0.0318 to 0.0308 for formation energy prediction in JARVIS.

5 Conclusion and Outlook

We study the problem of how to directly capture infinite-range interatomic potentials in crystal property prediction. As a radical departure from prior methods that only consider nearby atoms, we develop a new GNN, PotNet, with the message passing scheme that takes efficient approximations to capture the complete set of potentials among all atoms. Experiments show that the use of complete potentials leads to consistent performance improvements. Altogether, our work provides a theoretically principled and practically effective framework for crystal modeling. In future, we will further improve our algorithms for approximating the infinite summations with tighter error bounds and faster implementation in training PotNet.

REFERENCES

MI Baskes. Application of the embedded-atom method to covalent materials: a semiempirical potential for silicon. *Physical review letters*, 59(23):2666, 1987.

Harry Bateman. *Tables of integral transforms [volumes I & II]*, volume 1. McGraw-Hill Book Company, 1954.

Peter W Battaglia, Jessica B Hamrick, Victor Bapst, Alvaro Sanchez-Gonzalez, Vinicius Zambaldi, Mateusz Malinowski, Andrea Tacchetti, David Raposo, Adam Santoro, Ryan Faulkner, et al. Relational inductive biases, deep learning, and graph networks. *arXiv preprint arXiv:1806.01261*, 2018.

Simon Batzner, Albert Musaelian, Lixin Sun, Mario Geiger, Jonathan P Mailoa, Mordechai Kornbluth, Nicola Molinari, Tess E Smidt, and Boris Kozinsky. E (3)-equivariant graph neural networks for data-efficient and accurate interatomic potentials. *Nature communications*, 13(1): 1–11, 2022.

Laurent Bétermin, Markus Faulhuber, and Stefan Steinerberger. A variational principle for gaussian lattice sums. *arXiv preprint arXiv:2110.06008*, 2021.

David Borwein, Jonathan M Borwein, and Keith F Taylor. Convergence of lattice sums and madelung's constant. *Journal of mathematical physics*, 26(11):2999–3009, 1985.

JM Borwein and O-Yeat Chan. Uniform bounds for the incomplete complementary gamma function. *Mathematical Inequalities and Applications*, 12:115–121, 2009.

I David Brown. *The chemical bond in inorganic chemistry: the bond valence model*, volume 27. Oxford University Press, 2016.

Richard A Buckingham. The classical equation of state of gaseous helium, neon and argon. *Proceedings of the Royal Society of London. Series A. Mathematical and Physical Sciences*, 168 (933):264–283, 1938.

Chi Chen, Weike Ye, Yunxing Zuo, Chen Zheng, and Shyue Ping Ong. Graph networks as a universal machine learning framework for molecules and crystals. *Chemistry of Materials*, 31(9):3564–3572, 2019.

Kamal Choudhary and Brian DeCost. Atomistic line graph neural network for improved materials property predictions. *npj Computational Materials*, 7(1):1–8, 2021.

Kamal Choudhary, Brian DeCost, and Francesca Tavazza. Machine learning with force-field-inspired descriptors for materials: Fast screening and mapping energy landscape. *Physical review materials*, 2(8):083801, 2018.

Kamal Choudhary, Kevin F Garrity, Andrew CE Reid, Brian DeCost, Adam J Biacchi, Angela R Hight Walker, Zachary Trautt, Jason Hattrick-Simpers, A Gilad Kusne, Andrea Centrone, et al. The joint automated repository for various integrated simulations (jarvis) for data-driven materials design. *npj Computational Materials*, 6(1):1–13, 2020.

Richard E Crandall. Fast evaluation of epstein zeta functions. *manuscript, at http:== www. perfsci. com*, 1998.

Murray S Daw and Michael I Baskes. Embedded-atom method: Derivation and application to impurities, surfaces, and other defects in metals. *Physical Review B*, 29(12):6443, 1984.

Murray S Daw, Stephen M Foiles, and Michael I Baskes. The embedded-atom method: a review of theory and applications. *Materials Science Reports*, 9(7-8):251–310, 1993.

Bernard Deconinck, Matthias Heil, Alexander Bobenko, Mark Van Hoeij, and Marcus Schmies. Computing riemann theta functions. *Mathematics of Computation*, 73(247):1417–1442, 2004.

David Duvenaud, Dougal Maclaurin, Jorge Aguilera-Iparraguirre, Rafael Gómez-Bombarelli, Timothy Hirzel, Alán Aspuru-Guzik, and Ryan P Adams. Convolutional networks on graphs for learning molecular fingerprints. In *Proceedings of the 28th International Conference on Neural Information Processing Systems*, pp. 2224–2232, 2015.

Paul P Ewald. Die berechnung optischer und elektrostatischer gitterpotentiale. *Annalen der physik*, 369(3):253–287, 1921.

Mark Galassi, Jim Davies, James Theiler, Brian Gough, Gerard Jungman, Patrick Alken, Michael Booth, Fabrice Rossi, and Rhys Ulerich. *GNU scientific library*. Network Theory Limited Godalming, 2002.

Johannes Gasteiger, Janek Groß, and Stephan Günnemann. Directional message passing for molecular graphs. In *International Conference on Learning Representations*, 2019.

Johannes Gasteiger, Florian Becker, and Stephan Günnemann. Gemnet: Universal directional graph neural networks for molecules. *Advances in Neural Information Processing Systems*, 34:6790–6802, 2021.

Philippe Gaudreau, Richard M Slevinsky, and Hassan Safouhi. Computation of tail probabilities via extrapolation methods and connection with rational and padé approximants. *SIAM Journal on Scientific Computing*, 34(1):B65–B85, 2012.

Justin Gilmer, Samuel S Schoenholz, Patrick F Riley, Oriol Vinyals, and George E Dahl. Neural message passing for quantum chemistry. In *International conference on machine learning*, pp. 1263–1272. PMLR, 2017.

Rhys EA Goodall and Alpha A Lee. Predicting materials properties without crystal structure: Deep representation learning from stoichiometry. *Nature communications*, 11(1):1–9, 2020.

Frank E Harris. Incomplete bessel, generalized incomplete gamma, or leaky aquifer functions. *Journal of Computational and Applied Mathematics*, 215(1):260–269, 2008.

Emiel Hoogeboom, Vıctor Garcia Satorras, Clément Vignac, and Max Welling. Equivariant diffusion for molecule generation in 3D. In *International Conference on Machine Learning*, pp. 8867–8887. PMLR, 2022.

Weihua Hu, Matthias Fey, Hongyu Ren, Maho Nakata, Yuxiao Dong, and Jure Leskovec. OGB-LSC: A large-scale challenge for machine learning on graphs. In *Thirty-fifth Conference on Neural Information Processing Systems Datasets and Benchmarks Track*, 2021.

Jari Jalkanen and Martin H Müser. Systematic analysis and modification of embedded-atom potentials: case study of copper. *Modelling and simulation in materials science and engineering*, 23 (7):074001, 2015.

Dipendra Jha, Logan Ward, Arindam Paul, Wei-keng Liao, Alok Choudhary, Chris Wolverton, and Ankit Agrawal. Elemnet: Deep learning the chemistry of materials from only elemental composition. *Scientific reports*, 8(1):1–13, 2018.

Dipendra Jha, Logan Ward, Zijiang Yang, Christopher Wolverton, Ian Foster, Wei-keng Liao, Alok Choudhary, and Ankit Agrawal. IRNet: A general purpose deep residual regression framework for materials discovery. In *Proceedings of the 25th ACM SIGKDD International Conference on Knowledge Discovery & Data Mining*, pp. 2385–2393, 2019.

Wengong Jin, Regina Barzilay, and Tommi Jaakkola. Junction tree variational autoencoder for molecular graph generation. In *International conference on machine learning*, pp. 2323–2332. PMLR, 2018.

Bowen Jing, Stephan Eismann, Patricia Suriana, Raphael John Lamarre Townshend, and Ron Dror. Learning from protein structure with geometric vector perceptrons. In *International Conference on Learning Representations*, 2020.

DS Jones. Incomplete bessel functions. i. *Proceedings of the Edinburgh Mathematical Society*, 50 (1):173–183, 2007.

Diederik P Kingma and Jimmy Ba. Adam: A method for stochastic optimization. *arXiv preprint arXiv:1412.6980*, 2014.

Johannes Klicpera, Shankari Giri, Johannes T Margraf, and Stephan Günnemann. Fast and uncertainty-aware directional message passing for non-equilibrium molecules. *arXiv preprint arXiv:2011.14115*, 2020a.

Johannes Klicpera, Janek Groß, and Stephan Günnemann. Directional message passing for molecular graphs. *arXiv preprint arXiv:2003.03123*, 2020b.

Kenneth S Krane. *Introductory nuclear physics*. John Wiley & Sons, 1991.

Joseph PS Kung and Chung-Chun Yang. Complex analysis. *Encyclopedia of Physical Science and Technology*, 2003.

Eunkoo Lee, Kwang-Ryeol Lee, MI Baskes, and Byeong-Joo Lee. A modified embedded-atom method interatomic potential for ionic systems: 2 nnmeam+ qeq. *Physical Review B*, 93(14): 144110, 2016.

David Levin and Avram Sidi. Two new classes of nonlinear transformations for accelerating the convergence of infinite integrals and series. *Applied Mathematics and Computation*, 9(3):175–215, 1981.

Meng Liu, Youzhi Luo, Kanji Uchino, Koji Maruhashi, and Shuiwang Ji. Generating 3D molecules for target protein binding. In *Proceedings of The 39th International Conference on Machine Learning*, pp. 13912–13924, 2022a.

Yi Liu, Limei Wang, Meng Liu, Yuchao Lin, Xuan Zhang, Bora Oztekin, and Shuiwang Ji. Spherical message passing for 3D molecular graphs. In *International Conference on Learning Representations*, 2022b.

Yi Liu, Limei Wang, Meng Liu, Yuchao Lin, Xuan Zhang, Bora Oztekin, and Shuiwang Ji. Spherical message passing for 3d molecular graphs. In *International Conference on Learning Representations*, 2022c.

Ilya Loshchilov and Frank Hutter. Decoupled weight decay regularization. *arXiv preprint arXiv:1711.05101*, 2017.

Steph-Yves Louis, Yong Zhao, Alireza Nasiri, Xiran Wang, Yuqi Song, Fei Liu, and Jianjun Hu. Graph convolutional neural networks with global attention for improved materials property prediction. *Physical Chemistry Chemical Physics*, 22(32):18141–18148, 2020.

Youzhi Luo and Shuiwang Ji. An autoregressive flow model for 3D molecular geometry generation from scratch. In *International Conference on Learning Representations*, 2022.

Youzhi Luo, Keqiang Yan, and Shuiwang Ji. GraphDF: A discrete flow model for molecular graph generation. In *Proceedings of The 38th International Conference on Machine Learning*, pp. 7192–7203, 2021.

Franziska Nestler, Michael Pippig, and Daniel Potts. Fast ewald summation based on nfft with mixed periodicity. *Journal of Computational Physics*, 285:280–315, 2015.

DM Riffe, Jake D Christensen, and RB Wilson. Vibrational dynamics within the embedded-atom-method formalism and the relationship to born–von-kármán force constants. *Journal of Physics: Condensed Matter*, 30(45):455702, 2018.

Jonathan Schmidt, Love Pettersson, Claudio Verdozzi, Silvana Botti, and Miguel A. L. Marques. Crystal graph attention networks for the prediction of stable materials. *Science Advances*, 7(49): eabi7948, 2021. doi: 10.1126/sciadv.abi7948.

Kristof Schütt, Pieter-Jan Kindermans, Huziel Enoc Sauceda Felix, Stefan Chmiela, Alexandre Tkatchenko, and Klaus-Robert Müller. Schnet: A continuous-filter convolutional neural network for modeling quantum interactions. *Advances in neural information processing systems*, 30, 2017.

Kristof Schütt, Oliver Unke, and Michael Gastegger. Equivariant message passing for the prediction of tensorial properties and molecular spectra. In *International Conference on Machine Learning*, pp. 9377–9388. PMLR, 2021.

Chence Shi, Minkai Xu, Zhaocheng Zhu, Weinan Zhang, Ming Zhang, and Jian Tang. GraphAF: a flow-based autoregressive model for molecular graph generation. In *International Conference on Learning Representations*, 2019.

John C Slater. The normal state of helium. *Physical Review*, 32(3):349, 1928.

Richard M Slevinsky and Hassan Safouhi. A recursive algorithm for the g transformation and accurate computation of incomplete bessel functions. *Applied numerical mathematics*, 60(12): 1411–1417, 2010.

Leslie N Smith and Nicholay Topin. Super-convergence: Very fast training of neural networks using large learning rates. In *Artificial intelligence and machine learning for multi-domain operations applications*, volume 11006, pp. 369–386. SPIE, 2019.

Jonathan M Stokes, Kevin Yang, Kyle Swanson, Wengong Jin, Andres Cubillos-Ruiz, Nina M Donghia, Craig R MacNair, Shawn French, Lindsey A Carfrae, Zohar Bloom-Ackermann, et al. A deep learning approach to antibiotic discovery. *Cell*, 180(4):688–702, 2020.

J Philipp Wagner and Peter R Schreiner. London dispersion in molecular chemistry—reconsidering steric effects. *Angewandte Chemie International Edition*, 54(42):12274–12296, 2015.

Anthony Yu-Tung Wang, Steven K Kauwe, Ryan J Murdock, and Taylor D Sparks. Compositionally restricted attention-based network for materials property predictions. *Npj Computational Materials*, 7(1):1–10, 2021.

Limei Wang, Haoran Liu, Yi Liu, Jerry Kurtin, and Shuiwang Ji. Learning protein representations via complete 3D graph networks. *arXiv preprint arXiv:2207.12600*, 2022a.

Limei Wang, Yi Liu, Yuchao Lin, Haoran Liu, and Shuiwang Ji. ComENet: Towards complete and efficient message passing for 3D molecular graphs. In *The 36th Annual Conference on Neural Information Processing Systems*, 2022b.

Zhengyang Wang, Meng Liu, Youzhi Luo, Zhao Xu, Yaochen Xie, Limei Wang, Lei Cai, Qi Qi, Zhuoning Yuan, Tianbao Yang, and Shuiwang Ji. Advanced graph and sequence neural networks for molecular property prediction and drug discovery. *Bioinformatics*, 38(9):2579–2586, 2022c.

Anthony R West. *Basic Solid State Chemistry*. Wiley, 1988.

Tian Xie and Jeffrey C Grossman. Crystal graph convolutional neural networks for an accurate and interpretable prediction of material properties. *Physical review letters*, 120(14):145301, 2018.

A Gaussian Lattice Sum

The Gaussian Lattice Sum (Bétermin et al., 2021) computes the summation of Gaussian functions centered at the points given by a shifted lattice, formally defined as

$$G_{\Omega}(\boldsymbol{L}, \boldsymbol{v}, c) = \sum_{\boldsymbol{k} \in \mathbb{Z}^d} e^{-c|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2}, \tag{10}$$

where $\boldsymbol{L} \in \mathbb{R}^{d \times d}$ is the lattice matrix, $\boldsymbol{v} \in \mathbb{R}^d$ is a vector inside a unit cell and $c \in \mathbb{R}^+$ is a prefixed constant. One of the characteristics of the Gaussian Lattice Sum is that the term $e^{-c|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2}$ rapidly decays as $\boldsymbol{k}$ becomes large, resulting fast convergence of $G_{\Omega}(\boldsymbol{L}, \boldsymbol{v}, c)$. As shown by Deconinck et al. (2004), for $R \in \mathbb{R}^+$, we have

$$\sum_{\boldsymbol{k} \in \mathbb{Z}^d, \sqrt{c}|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}| \geq R} e^{-c|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2} \leq \frac{d}{2}(\frac{2}{\rho})^d \Gamma(\frac{d}{2}, (R - \frac{\rho}{2})^2), \tag{11}$$

where $\Gamma(z,x) = \int_x^\infty t^{z-1}e^{-t}dt$ is the incomplete Gamma function and $\rho = \min\{\sqrt{c}|\boldsymbol{L}\boldsymbol{k}| \; |\boldsymbol{k} \in \mathbb{Z}^d, \boldsymbol{k} \neq \boldsymbol{0}\}$. We can obtain the upper bound of the Gaussian Lattice Sum by setting $R = 0$ such that

$$\sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{-c|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2} \leq \frac{d}{2}(\frac{2}{\rho})^d \Gamma(\frac{d}{2}, (\frac{\rho}{2})^2). \tag{12}$$

B INCOMPLETE BESSEL FUNCTION

The incomplete Bessel function (Harris, 2008) is defined as

$$K_\nu(x,y) = \int_1^\infty t^{-\nu-1}e^{-xt-y/t}dt, \tag{13}$$

where $x, y \in \mathbb{R}$, $x \geq 0$, $y \geq 0$, and $\nu > 0$. In existing studies (Harris, 2008; Slevinsky & Safouhi, 2010; Jones, 2007), $K_\nu(x,y)$ can be analytically continued to $\nu \in \mathbb{R}$. In this work, we follow these studies and also use the analytical continuation for our calculations, i.e., we also consider $\nu \in \mathbb{R}$.

B.1 FAST APPROXIMATION OF THE INCOMPLETE BESSEL FUNCTION

Computing the incomplete Bessel function is extremely challenging as there does not exist any explicit closed-form solution. In this work, we investigate a fast approximation of the incomplete Bessel function. Specifically, we adopt the algorithm in the work by Slevinsky & Safouhi (2010), where the incomplete Bessel function $K_\nu(x,y)$ is approximated by the $G_n^{(m)}$ transformation with the linear time complexity of $O(n)$, and n is the number of iterations. As shown by Levin & Sidi (1981), the approximation $G_n^{(m)}$ to $\int_0^\infty f(t)dt$ is given as the solution of

$$\frac{d^l}{dx^l}\{G_n^{(m)} - \int_0^x f(t)dt - \sum_{k=0}^{m-1} x^{\sigma_k} f^{(k)}(x) \sum_{i=0}^{n-1} \frac{\beta_{i,k}}{x^i}\} = 0, \tag{14}$$

where $\beta_{i,k}$ and $G_n^{(m)}$ are unknowns, $\frac{d^l}{dx^l}G_n^{(m)} = 0, \forall l > 0$, $\sigma_k = min(s_k, k+1)$, and s_k is the largest of $s \in \mathbb{Z}$ such that $\lim_{x\to\infty} x^s f^{(k)}(x) = 0$ for $k = 0, 1, \cdots, m-1$. Slevinsky & Safouhi (2010) proved that the incomplete Bessel function satisfies Eqn. (14) with $m = 1$, through which we can obtain

$$G_n^{(1)} - \int_0^x f(t)dt = x^{\sigma_0} f(x) \sum_{i=0}^{n-1} \frac{\beta_{i,k}}{x^i}. \tag{15}$$

To eliminate all the unknowns $\beta_{i,k}$, Slevinsky & Safouhi (2010) applied the $(x^2\frac{d}{dx})$ operator n times such that

$$(x^2\frac{d}{dx})^n \left[\frac{G_n^{(1)} - \int_0^x f(t)dt}{x^{\sigma_0} f(x)}\right] = 0. \tag{16}$$

By doing this, we obtain

$$G_n^{(1)} = \frac{(x^2\frac{d}{dx})^n(\frac{\int_0^x f(t)dt}{x^{\sigma_0}f(x)})}{(x^2\frac{d}{dx})^n(\frac{1}{x^{\sigma_0}f(x)})} = \frac{\mathcal{N}_n(x)}{\mathcal{D}_n(x)}, \tag{17}$$

in which we can have

$$\mathcal{N}_n(x) = (x^2\frac{d}{dx})\mathcal{N}_{n-1}(x) \quad \text{and} \quad \mathcal{D}_n(x) = (x^2\frac{d}{dx})\mathcal{G}_{n-1}(x) \tag{18}$$

with

$$\mathcal{N}_0(x) = \frac{\int_0^x f(t)dt}{x^{\sigma_0}f(x)} \quad \text{and} \quad \mathcal{D}_0(x) = \frac{1}{x^{\sigma_0}f(x)}. \tag{19}$$

This leads to a recursive algorithm to approximate $G_n^{(1)}$. To compute the incomplete Bessel function $K_\nu(x, y)$, Slevinsky & Safouhi (2010) investigated the following property

$$K_\nu(x,y) + x^\nu \int_0^x t^{-\nu-1} e^{-t-\frac{xy}{t}} dt = x^\nu \int_0^\infty t^{-\nu-1} e^{-t-\frac{xy}{t}} dt, \tag{20}$$

in which the term $\int_0^\infty t^{-\nu-1} e^{-t-\frac{xy}{t}} dt$ can be approximated by $G_n^{(1)}$. Therefore, to approximate $K_\nu(x, y)$ we have

$$\begin{aligned}
\tilde{G}_n^{(1)} &= x^\nu (G_n^{(1)} - \int_0^x t^{-\nu-1} e^{-t-\frac{xy}{t}} dt) \\
&= x^\nu \frac{\mathcal{N}_n(x) - \mathcal{D}_n(x) \int_0^x t^{-\nu-1} e^{-t-\frac{xy}{t}} dt}{\mathcal{D}_n(x)} \\
&= x^\nu \frac{\sum_{r=1}^n \binom{n}{r} \mathcal{D}_{n-r}(x) (x^2 \frac{d}{dx})^{r-1} (x^{-\nu+1} e^{-x-y})}{\mathcal{D}_n(x)} \\
&= \frac{\tilde{\mathcal{N}}_n(x)}{\mathcal{D}_n(x)}.
\end{aligned} \tag{21}$$

As a result, we obtain the approximation $\tilde{G}_n^{(1)}$ to $K_\nu(x, y)$ by recursively solving $\tilde{\mathcal{N}}_n(x)$ and $\mathcal{D}_n(x)$ (Gaudreau et al., 2012). The detailed expressions of $\tilde{\mathcal{N}}_n(x)$ and $\mathcal{D}_n(x)$ are given in Slevinsky & Safouhi (2010). In addition, we follow Nestler et al. (2015) to further optimize the approximation of the incomplete Bessel function when $\nu = 0$ and x, y are both small, e.g. $x^2 + y^2 < 1$. In this case, the remaining part of the Taylor expansion of $K_0(x, y)$ is small and we can approximate $K_0(x, y)$ by the first m terms of Taylor series such that

$$K_0(x,y) \approx \sum_{n=0}^m \frac{(-1)^n}{n!} x^n y^n \Gamma(-n, x). \tag{22}$$

The detailed error bound by this expansion is shown in Nestler et al. (2015).

B.2 Convergence of Incomplete Bessel Function Summation

We define the summation of incomplete Bessel functions on a lattice as

$$\sum_{\boldsymbol{k} \in \mathbb{Z}^d} K_\nu(\alpha |\boldsymbol{L}\boldsymbol{k} + \boldsymbol{v}|^2 + \gamma, \beta), \tag{23}$$

where $\nu, \alpha, \beta, \gamma \in \mathbb{R}$ are constants, $\alpha > 0$, $\beta \geq 0$, $\gamma \geq 0$, $\boldsymbol{v} \in \mathbb{R}^d$ is a vector inside a unit cell and $\boldsymbol{L} \in \mathbb{R}^{d \times d}$ is the lattice matrix. We aim to prove the summation of incomplete Bessel functions is convergent and can be approximated with an error bounded by the Gaussian Lattice Sum introduced in Appendix A.

Proof. The incomplete Bessel function has an upper bound such that

$$\big|K_\nu(x,y)\big| = \Big| \int_1^\infty t^{-\nu-1} e^{-xt-y/t} dt \Big| \leq \Big| \int_1^\infty t^{-\nu-1} e^{-xt} dt \Big| = \big| x^{-\nu} \Gamma(-\nu, x) \big|, \tag{24}$$

where $x > 0$ and Γ is the incomplete Gamma function described in Appendix A. Based on this, we obtain

$$\big|K_\nu(\alpha|\boldsymbol{L}\boldsymbol{k} + \boldsymbol{v}|^2 + \gamma, \beta)\big| \leq \Big| \frac{\Gamma(-\nu, \alpha|\boldsymbol{L}\boldsymbol{k} + \boldsymbol{v}|^2 + \gamma)}{(\alpha|\boldsymbol{L}\boldsymbol{k} + \boldsymbol{v}|^2 + \gamma)^\nu} \Big|, \tag{25}$$

where $\alpha|\boldsymbol{L}\boldsymbol{k} + \boldsymbol{v}|^2 + \gamma > 0$. As shown by Borwein & Chan (2009), $|\frac{\Gamma(z,x)}{x^z}|$ has an upper bound such that

$$\begin{aligned}
\Big|\frac{\Gamma(z,x)}{x^z}\Big| &= \Big| e^{-x} \int_0^\infty e^{-xs} (1+s)^{z-1} ds \Big| \\
&\leq \begin{cases} \frac{e^{-x}}{x-z+1}, & z > 1 \\ \frac{e^{-x}}{x}, & z \leq 1 \end{cases},
\end{aligned} \tag{26}$$

where $z \in \mathbb{R}$, $x \in \mathbb{R}$ and $x > 0$. For a prefixed value $R \in \mathbb{R}$, $R^2 > -\nu$, $R^2 > 1$ and $R^2 > \gamma$, we have

$$\begin{aligned}\epsilon(R) &= \sum_{\boldsymbol{k}\in\mathbb{Z}^d, \alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma\geq R^2} K_\nu(\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma,\beta) \\ &\leq \sum_{\boldsymbol{k}\in\mathbb{Z}^d, \alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma\geq R^2} |K_\nu(\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma,\beta)| \\ &\leq \sum_{\boldsymbol{k}\in\mathbb{Z}^d, \alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma\geq R^2} e^{-\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2-\gamma} \\ &\leq e^{-\gamma} G_\Omega(\boldsymbol{L},\boldsymbol{v},\alpha),\end{aligned} \tag{27}$$

where $G_\Omega(\boldsymbol{L},\boldsymbol{v},\alpha)$ is the Gaussian Lattice Sum as described in Appendix A. Therefore, the incomplete Bessel function summation can be divided into two parts such that

$$\sum_{\boldsymbol{k}\in\mathbb{Z}^d} K_\nu(\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma,\beta) = \sum_{\boldsymbol{k}\in\mathbb{Z}^d, \alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma\leq R^2} K_\nu(\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2+\gamma,\beta) + \epsilon(R), \tag{28}$$

where the first part is a finite part inside an ellipsoid with a size of $\sqrt{R^2/\alpha - \gamma/\alpha}$, and the second part $\epsilon(R)$ is bounded by the Gaussian Lattice Sum $G_\Omega(\boldsymbol{L},\boldsymbol{v},\alpha)$ which is convergent. Thus, the incomplete Bessel function summation is convergent. Consequently, to approximate the incomplete Bessel function summation, we choose to evaluate the summation inside an ellipsoid with the size of $\sqrt{R^2/\alpha - \gamma/\alpha}$ for a prefixed $R \in \mathbb{R}$, such that $R^2 > -\nu$, $R^2 > 1$ and $R^2 > \gamma$. Then the error $\epsilon(R)$ is bounded by Gaussian Lattice Sum $G_\Omega(\boldsymbol{L},\boldsymbol{v},\alpha)$. We can further bound the error by the inequality (11) of Gaussian Lattice Sum as introduced in Appendix A such that

$$\epsilon(R) \leq \sum_{\boldsymbol{k}\in\mathbb{Z}^d, \sqrt{\alpha}|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|\geq\sqrt{R^2-\gamma}} e^{-\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2} \leq \frac{d}{2}(\frac{2}{\rho})^d \Gamma(\frac{d}{2}, (\sqrt{R^2-\gamma} - \frac{\rho}{2})^2), \tag{29}$$

where $\rho = \min\{\sqrt{\alpha}|\boldsymbol{L}\boldsymbol{k}| \; |\boldsymbol{k} \in \mathbb{Z}^d, \boldsymbol{k} \neq \boldsymbol{0}\}$. This completes the proof. □

C Fast Algorithm of Lattice Sum

C.1 Integral Transformation

Here, we prove that the summation of the three introduced potentials can be transformed into an integral form as

$$\mathcal{S}(\boldsymbol{L},\boldsymbol{v}) = D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt, \tag{30}$$

where A, B, C, D are constants derived from the corresponding specific potential forms.

Proof. We first prove that both the potential forms $V(\boldsymbol{L},\boldsymbol{v}) = 1/|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^{2p}$ and $V(\boldsymbol{L},\boldsymbol{v}) = e^{-\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|}$ can be written in the following integral form

$$V(\boldsymbol{L},\boldsymbol{v}) = D \int_0^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt. \tag{31}$$

1). For the potentials in the form of $V(\boldsymbol{L},\boldsymbol{v}) = 1/|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^{2p}$, we apply the Mellin transform such that

$$\begin{aligned} M\{V\}(\boldsymbol{L},\boldsymbol{v}) &= \int_0^\infty t^{p-1} e^{-t|\boldsymbol{L}\boldsymbol{n}+\boldsymbol{v}|^2} dt \\ &= \frac{\Gamma(p)}{|\boldsymbol{L}\boldsymbol{n}+\boldsymbol{v}|^{2p}}. \end{aligned} \tag{32}$$

Thus, we obtain

$$V(\boldsymbol{L},\boldsymbol{v}) = \frac{1}{|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^{2p}} = \frac{1}{\Gamma(p)} \int_0^\infty t^{p-1} e^{-t|\boldsymbol{L}\boldsymbol{n}+\boldsymbol{v}|^2} dt. \tag{33}$$

Apparently, we can obtain $A = 1/\pi$, $B = 0$, $C = p$ and $D = 1/\Gamma(p)$ for the integral form of $V(\boldsymbol{L}, \boldsymbol{v}) = 1/|\boldsymbol{L}\boldsymbol{k} + \boldsymbol{v}|^{2p}$.

2). For the potentials in the form of $V(\boldsymbol{L}, \boldsymbol{v}) = e^{-\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|}$, we consider the inverse Laplace transform on $e^{-\alpha\sqrt{s}}$ as shown by Bateman (1954), such that

$$\mathcal{L}^{-1}\{e^{-\alpha\sqrt{s}}\} = \frac{a}{2\sqrt{\pi}} t^{-\frac{3}{2}} e^{-\frac{\alpha^2}{4t}}. \tag{34}$$

Therefore, we can apply the Laplace transform in Eqn. (34) to derive the integral form of $e^{-\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|}$:

$$\begin{aligned} V(\boldsymbol{L}, \boldsymbol{v}) &= e^{-\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|} \\ &= \frac{a}{2\sqrt{\pi}} \int_0^\infty t^{-\frac{3}{2}} e^{-|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{\alpha^2}{4t}} dt \\ &= \frac{\alpha}{2\pi} \int_0^\infty t^{-\frac{3}{2}} e^{-\pi t|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 - \frac{\alpha^2}{4\pi t}} dt \, (t \leftarrow \pi t). \end{aligned} \tag{35}$$

Apparently, we can obtain $A = 1$, $B = \frac{\alpha^2}{4\pi}$, $C = -\frac{1}{2}$ and $D = \frac{\alpha}{2\pi}$ for the integral form of $V(\boldsymbol{L}, \boldsymbol{v}) = e^{-\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|}$.

Finally, we conduct a summation for these two types of potentials $V(\boldsymbol{L}, \boldsymbol{v})$ in the space $\boldsymbol{k} \in \mathbb{Z}^d$ as

$$S(\boldsymbol{L}, \boldsymbol{v}) = \sum_{\boldsymbol{k}\in\mathbb{Z}^d} V(\boldsymbol{L}, \boldsymbol{v}). \tag{36}$$

This completes the proof. □

C.2 Calculating Integral Summation

As shown in Sec. 3.4, $\mathcal{S}(\boldsymbol{L}, \boldsymbol{v})$ can be written as the summations in the Euclidean space directly and then Fourier space:

$$\begin{aligned} \mathcal{S}(\boldsymbol{L}, \boldsymbol{v}) &= D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt \\ &= D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^1 t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt + D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_1^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt \\ &= S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v}) + S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v}), \end{aligned} \tag{37}$$

where $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v}) = D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^1 t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt$ denotes the summation in Fourier space, and $S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v}) = D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_1^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt$ denotes the summation in direct space. Apparently, $S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v})$ is already the form of the incomplete Bessel function summation. We apply the analytic continuation to $S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v})$ to expand the domain of C in $S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v})$ as detailed in Appendix C.3 and we have

$$S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v}) = D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} K_{-C}(A\pi|\boldsymbol{L}\boldsymbol{k} + \boldsymbol{v}|^2, B) \tag{38}$$

for all constant $C \in \mathbb{R}$. Below, we prove that $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v})$ can be deduced into the incomplete Bessel function summation.

Proof. Inspired by the Ewald summation (Ewald, 1921; Crandall, 1998), we consider $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v})$ on the reciprocal lattice using the Poisson summation (Crandall, 1998):

$$\sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{2\pi i \boldsymbol{w}\cdot\boldsymbol{L}\boldsymbol{k} - \pi t|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2} = \frac{t^{-\frac{d}{2}} e^{2\pi i \boldsymbol{w}\cdot\boldsymbol{v}}}{\det \boldsymbol{L}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{2\pi i \boldsymbol{L}'\boldsymbol{k}\cdot\boldsymbol{v} - \frac{\pi}{t}|\boldsymbol{L}'\boldsymbol{k}+\boldsymbol{w}|^2}, \tag{39}$$

where $\boldsymbol{w} \in \mathbb{R}^d$ is a vector and $\boldsymbol{w} = \mathbf{0}$ in our case, and $\boldsymbol{L}' = \boldsymbol{L}(\boldsymbol{L}^T\boldsymbol{L})^{-1}$ is the lattice matrix for the reciprocal lattice. As a result, We obtain

$$
\begin{aligned}
S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v}) &= D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^1 t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt \\
&= \frac{D}{A^C} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^A t^{C-1} e^{-\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{AB}{t}} dt (t \leftarrow \frac{t}{A}) \\
&= \frac{1}{\det \boldsymbol{L}} \frac{D}{A^C} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^A t^{C-\frac{d}{2}-1} e^{2\pi i \boldsymbol{L}'\boldsymbol{n}\cdot\boldsymbol{v} - \frac{\pi}{t}\left|\boldsymbol{L}'\boldsymbol{k}\right|^2 - \frac{AB}{t}} dt \quad \text{(Eqn. (39))} \\
&= \frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^1 t^{C-\frac{d}{2}-1} e^{2\pi i \boldsymbol{L}'\boldsymbol{k}\cdot\boldsymbol{v} - \frac{\pi}{At}\left|\boldsymbol{L}'\boldsymbol{k}\right|^2 - \frac{B}{t}} dt (t \leftarrow At) \\
&= \frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_1^\infty t^{\frac{d}{2}-C-1} e^{2\pi i \boldsymbol{L}'\boldsymbol{n}\cdot\boldsymbol{v} - \frac{\pi t}{A}\left|\boldsymbol{L}'\boldsymbol{k}\right|^2 - Bt} dt (t \leftarrow \frac{1}{t}) \\
&= \frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{2\pi i \boldsymbol{L}'\boldsymbol{k}\cdot\boldsymbol{v}} K_{C-\frac{d}{2}}(\frac{\pi\left|\boldsymbol{L}'\boldsymbol{k}\right|^2}{A} + B, 0).
\end{aligned}
\tag{40}
$$

We also apply the analytic continuation to $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v})$ in the last step. Apparently, $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v})$ is deduced into the incomplete Bessel function summation, and this completes the proof. □

Therefore, both $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v})$ and $S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v})$ can be expressed by the incomplete Bessel function summation forms:

$$
\begin{aligned}
S(\boldsymbol{L}, \boldsymbol{v}) &= S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v}) + S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v}) \\
&= D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} K_{-C}(A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2, B) \\
&\quad + \frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{2\pi i \boldsymbol{L}'\boldsymbol{k}\cdot\boldsymbol{v}} K_{C-\frac{d}{2}}(\frac{\pi\left|\boldsymbol{L}'\boldsymbol{k}\right|^2}{A} + B, 0).
\end{aligned}
\tag{41}
$$

As shown in Appendix B.2, the incomplete Bessel function summation $\sum_{\boldsymbol{k}\in\mathbb{Z}^d} K_\nu(\alpha|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 + \gamma, \beta)$ is convergent and can be approximated. Therefore, $S(\boldsymbol{L}, \boldsymbol{v})$ can also be approximated.

C.3 Analytic Continuation of Potential Summations

To represent the series that is not convergent, including the inverse summation $\sum_{\boldsymbol{k}\in\mathbb{Z}^d} \frac{1}{|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|}$ as shown in Sec. 3.4, we need to investigate the analytic continuation of potential summation. As shown in Appendix C.2, the potential summation can be written as

$$
\begin{aligned}
\mathcal{S}(\boldsymbol{L}, \boldsymbol{v}) &= D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt \\
&= D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^1 t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt + D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_1^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt.
\end{aligned}
\tag{42}
$$

These two parts can be deduced into two incomplete Bessel function summations by analytic continuation. Analytic continuation is a technique to extend the domain P of a given analytic function $f(x)$. Consequently, we denote $\hat{f}(x)$ as an analytic continuation of $f(x)$ to Q, where we denote the domain Q containing P, and denote a function $\hat{f}(x)$ that is analytic on Q, and $\hat{f}(x) = f(x)$ holds for all x in P. As shown by Kung & Yang (2003), the analytic continuation is unique and satisfies the permanence of functional relationships, i.e., the equations holding for $f(x)$ will also hold for $\hat{f}(x)$. In our case, we can expand the domain of constant C in $\mathcal{S}(\boldsymbol{L}, \boldsymbol{v})$ to $C \in \mathbb{R}$ such that $\mathcal{S}(\boldsymbol{L}, \boldsymbol{v})$ is well-defined for any $C \in \mathbb{R}$. For example, for summation of the potentials in the form of $\sum_{\boldsymbol{k}\in\mathbb{Z}^d} 1/|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^p$, we have $C = p$ as shown in Appendix C.1. Analytic continuation enables us

to compute the summation when $p = 1$, which is initially divergent. Formally, we assume that C is originally defined in the domain $P \subset \mathbb{R}$. As derived in Appendix C.1, for $C \in P$, we have

$$\begin{aligned}\mathcal{S}(\boldsymbol{L}, \boldsymbol{v}) =& D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_0^1 t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt + D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} \int_1^\infty t^{C-1} e^{-A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2 t - \frac{B}{t}} dt \\ =& D \sum_{\boldsymbol{k}\in\mathbb{Z}^d} K_{-C}(A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2, B) \\ &+ \frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{2\pi i \boldsymbol{L}'\boldsymbol{k}\cdot\boldsymbol{v}} K_{C-\frac{d}{2}}(\frac{\pi\,|\boldsymbol{L}'\boldsymbol{k}|^2}{A} + B, 0).\end{aligned} \tag{43}$$

Furthermore, we have $\nu = -C$ and $\nu = C - \frac{d}{2}$ in the incomplete Bessel function, which is analytically continued to $\nu \in \mathbb{R}$ (Jones, 2007). That is, P is contained by ν's domain $\mathbb{R}$. Therefore, the incomplete Bessel function summations are the analytic continuation of the corresponding potential summations.

To explain the advantage of analytic continuation, we consider the infinite potential summation inside a unit cell such that

$$\mathcal{S} = \sum_{\boldsymbol{a}\in\boldsymbol{A}} \sum_{\boldsymbol{b}\neq\boldsymbol{a}, \boldsymbol{b}\in\widetilde{\boldsymbol{A}}} V(\boldsymbol{a}, \boldsymbol{b}). \tag{44}$$

For a non-convergent $\mathcal{S}$, analytic continuation will result in an unusual value such as negative energy instead of infinity. This is useful in practice since the infinity will cause a numerical explosion in training. And for a convergent series $\mathcal{S}$, due to the permanence of functional relationships, analytic continuation will also produce a reasonable result. Specifically, if we use an analytically continued function to approximate convergent summation $\mathcal{S}(\boldsymbol{a}, \boldsymbol{b}) = \sum_{\boldsymbol{c}\neq\boldsymbol{a}, \boldsymbol{c}\in\boldsymbol{A}_{\boldsymbol{b}}} V(\boldsymbol{c}, \boldsymbol{a})$, we can approximate $\mathcal{S}$ by deriving

$$\mathcal{S} = \sum_{\boldsymbol{a}\in\boldsymbol{A}} \sum_{\boldsymbol{b}\neq\boldsymbol{a}, \boldsymbol{b}\in\widetilde{\boldsymbol{A}}} V(\boldsymbol{a}, \boldsymbol{b}) = \sum_{\boldsymbol{a}\in\boldsymbol{A}} \sum_{\boldsymbol{b}\in\boldsymbol{A}} \sum_{\boldsymbol{c}\neq\boldsymbol{a}, \boldsymbol{c}\in\boldsymbol{A}_{\boldsymbol{b}}} V(\boldsymbol{c}, \boldsymbol{a}) = \sum_{\boldsymbol{a}\in\boldsymbol{A}} \sum_{\boldsymbol{b}\in\boldsymbol{A}} \mathcal{S}(\boldsymbol{a}, \boldsymbol{b}). \tag{45}$$

As for non-convergent summation $\mathcal{S}(\boldsymbol{a}, \boldsymbol{b}) = \sum_{\boldsymbol{c}\neq\boldsymbol{a}, \boldsymbol{c}\in\boldsymbol{A}_{\boldsymbol{b}}} V(\boldsymbol{c}, \boldsymbol{a})$, due to the permanence of functional relationships of analytic continuation, the equation $\sum_{\boldsymbol{a}\in\boldsymbol{A}} \sum_{\boldsymbol{b}\neq\boldsymbol{a}, \boldsymbol{b}\in\widetilde{\boldsymbol{A}}} V(\boldsymbol{a}, \boldsymbol{b}) = \sum_{\boldsymbol{a}\in\boldsymbol{A}} \sum_{\boldsymbol{b}\in\boldsymbol{A}} \mathcal{S}(\boldsymbol{a}, \boldsymbol{b})$ still holds. A famous example of using analytic continuation for crystal energy prediction is the Madelung constant of NaCl, which is derived by the summation of Coulomb potentials among Na and Cl ions and calculated by analytic continuation of an absolutely convergent series (Borwein et al., 1985).

C.4 Implementation and Numerical Examples of Approximation

Here we describe the implementation of our algorithm as in Eqn. (41). Considering the fact that solving the inverse of the incomplete Gamma function is complicated, instead, we provide a proper value R and then calculate its corresponding error bound ϵ based on Eqn. (29). Given a lattice matrix $\boldsymbol{L} \in \mathbb{R}^{d\times d}$, a vector $\boldsymbol{v} \in \mathbb{R}^d$ inside a unit cell, and the constants A, B, C, D derived from specific potential functions as described in Appendix C.1, we aim to evaluate these two parts $D\sum_{\boldsymbol{k}\in\mathbb{Z}^d} K_{-C}(A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2, B)$ and $\frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{2\pi i \boldsymbol{L}'\boldsymbol{k}\cdot\boldsymbol{v}} K_{C-\frac{d}{2}}(\frac{\pi|\boldsymbol{L}'\boldsymbol{k}|^2}{A} + B, 0)$.

To evaluate $S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v}) = D\sum_{\boldsymbol{k}\in\mathbb{Z}^d} K_{-C}(A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2, B)$, we derive the following steps.

Step 1: Determine the value R such that $R^2 > C$, $R^2 > 1$, and calculate the error bound $\epsilon = \frac{d}{2}(\frac{2}{\rho})^d \Gamma(\frac{d}{2}, (R - \frac{\rho}{2})^2)$, where $\rho = \min\{\sqrt{A\pi}|\boldsymbol{L}\boldsymbol{k}| \; |\boldsymbol{k} \in \mathbb{Z}^d, \boldsymbol{k} \neq \boldsymbol{0}\}$.

Step 2: Select points inside an ellipsoid such that $\boldsymbol{P} = \{\boldsymbol{k}| \; |\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}| \leq R/\sqrt{A\pi}\}$.

Step 3: Evaluate the incomplete Bessel function summation $S_{\text{direct}}(\boldsymbol{L}, \boldsymbol{v})$ by calculating the every term $DK_{-C}(A\pi|\boldsymbol{L}\boldsymbol{k}+\boldsymbol{v}|^2, B)$ based on Appendix B.1.

To evaluate $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v}) = \frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} \sum_{\boldsymbol{k}\in\mathbb{Z}^d} e^{2\pi i \boldsymbol{L}'\boldsymbol{k}\cdot\boldsymbol{v}} K_{C-\frac{d}{2}}(\frac{\pi|\boldsymbol{L}'\boldsymbol{k}|^2}{A} + B, 0)$, we derive the following steps.

Table 6: Numerical examples of our algorithm. Here, $\zeta(x) = \sum_{n=1}^{\infty} 1/n^x$. Implementation details can be found in Appendix C.4.

Target	Evaluation	Error Bound	Time
$2 * \zeta(2) = 3.28986813$	3.28068288	8e-1	0.002 s
$2 * \zeta(2) = 3.28986813$	3.28984070	6e-2	0.003 s
$2 * \zeta(2) = 3.28986813$	3.28986812	7e-4	0.003 s
$2 * \zeta(2) = 3.28986813$	3.28986813	1e-6	0.003 s
$2 * \zeta(3) = 2.40411381$	2.40411381	1e-6	0.003 s
$2 * \zeta(4) = 2.16464647$	2.16464647	1e-6	0.003 s
$\sum_{\boldsymbol{n}\in\mathbb{Z}^2,\boldsymbol{n}\neq\boldsymbol{0}} \frac{1}{\lvert\boldsymbol{n}\rvert^4} = 6.02681204$	5.99068949	3	0.003 s
$\sum_{\boldsymbol{n}\in\mathbb{Z}^2,\boldsymbol{n}\neq\boldsymbol{0}} \frac{1}{\lvert\boldsymbol{n}\rvert^4} = 6.02681204$	6.02670959	2e-1	0.003 s
$\sum_{\boldsymbol{n}\in\mathbb{Z}^2,\boldsymbol{n}\neq\boldsymbol{0}} \frac{1}{\lvert\boldsymbol{n}\rvert^4} = 6.02681204$	6.02681199	3e-3	0.003 s
$\sum_{\boldsymbol{n}\in\mathbb{Z}^2,\boldsymbol{n}\neq\boldsymbol{0}} \frac{1}{\lvert\boldsymbol{n}\rvert^4} = 6.02681204$	6.02681204	5e-6	0.003 s
$\sum_{n\in\mathbb{Z}} e^{-\lvert n\rvert} = 2.16395341$	2.16395326	4e-1	0.002 s
$\sum_{n\in\mathbb{Z}} e^{-\lvert n\rvert} = 2.16395341$	2.16395341	3e-4	0.004 s
$\sum_{\boldsymbol{n}\in\mathbb{Z}^3} e^{-\lvert\boldsymbol{n}\rvert} = 25.39268269$	25.39268214	2.5	0.003 s
$\sum_{\boldsymbol{n}\in\mathbb{Z}^3} e^{-\lvert\boldsymbol{n}\rvert} = 25.39268269$	25.39268269	1e-2	0.003 s

Step 1: Determine the value R such that $R^2 > \frac{d}{2} - C$, $R^2 > 1$, $R^2 > B$, and calculate the error bound $\epsilon > \frac{d}{2}(\frac{2}{\rho})^d \Gamma(\frac{d}{2}, (\sqrt{R^2 - B} - \frac{\rho}{2})^2)$, where $\rho = \min\{\sqrt{\frac{\pi}{A}}|\boldsymbol{L'k}| \; |\boldsymbol{k} \in \mathbb{Z}^d, \boldsymbol{k} \neq \boldsymbol{0}\}$.

Step 2: Select points inside an ellipsoid such that $\boldsymbol{P} = \{\boldsymbol{k}| \; |\boldsymbol{Lk} + \boldsymbol{v}| \leq \sqrt{A(R^2 - B)/\pi}\}$.

Step 3: Evaluate the incomplete Bessel function summation $S_{\text{Fourier}}(\boldsymbol{L}, \boldsymbol{v})$ by calculating every term $\frac{1}{\det \boldsymbol{L}} \frac{D}{A^{\frac{d}{2}}} e^{2\pi i \boldsymbol{L'k}\cdot\boldsymbol{v}} K_{C-\frac{d}{2}}(\frac{\pi\left|\boldsymbol{L'k}\right|^2}{A} + B, 0)$ based on Appendix B.1.

Our implementation is based on Cython and GNU Scientific Library (Galassi et al., 2002), in which the native incomplete Gamma function and Bessel function are used to implement the incomplete Bessel function. We conduct numerical experiments on Intel Xeon Gold 6258R CPU. We show the evaluation examples in Table 6 with the corresponding error bound and evaluation time. The running time is at the scale of milliseconds.

D Network Architecture

The employed network architecture is shown in Fig. 2. Since our major contribution is to consider infinite interatomic potentials, we simply design our network architecture following the commonly used settings. Specifically, existing methods for 3D graphs (Xie & Grossman, 2018; Schütt et al., 2017; Klicpera et al., 2020b;a; Liu et al., 2022c; Gasteiger et al., 2021; Schütt et al., 2021) share a similar architecture, which usually contains an input block, an interaction block, and an output block. Without loss of generality, we take the updating process for node i as an example to illustrate the network.

The Inputs contain atomic features and potentials. $\mathbf{z}_i$ is the 92-dimensional atomic feature for any atom i following CGCNN (Xie & Grossman, 2018). Below we denote d as the interatomic distances. As explained in Sec. 4.1, $\mathbf{e}^c = -\epsilon_1'/d$ denotes the Coulomb potentials for the local crystal graph. $\mathbf{s}^c = -\sum_d \epsilon_1/d, \mathbf{s}^l = -\sum_d \epsilon/d^6, \mathbf{s}^p = \sum_d e^{-\alpha d}$ denote the summations of Coulomb potentials, London dispersion potentials, and Pauli potentials for the infinite crystal graph, and we set $\epsilon_1' = 4.0$ for $\mathbf{e}^c$, $\epsilon_1 = -1.0$ for $\mathbf{s}^c$, $\epsilon = 4.0$ for $\mathbf{s}^l$ and $\alpha = 3.0$ for $\mathbf{s}^p$, respectively. For simplicity, we apply them to the RBF embeddings with the same cutoff. Since $\mathbf{e}^c$ and $\mathbf{s}^l$ are negative values and have values exceeding the RBF cutoff, we use an exponential function to make their absolute values smaller.

The input block contains a ***Linear*** layer and an ***Embedding*** layer. For each node i, the ***Linear*** layer is employed to generate a 256-dimensional vector as the input node features to the first interac-

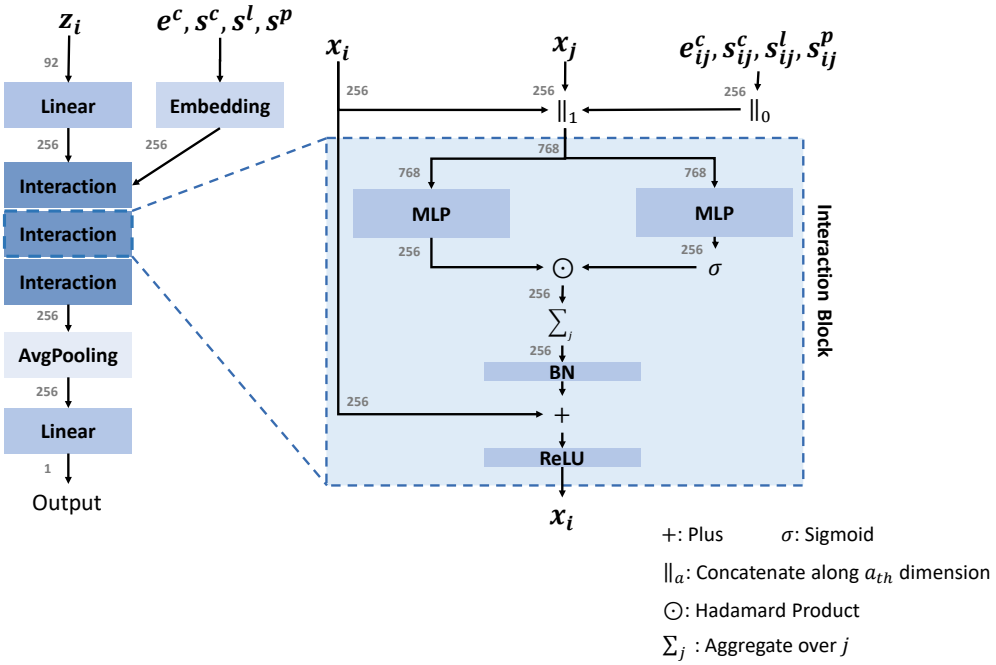

Figure 2: The developed network architecture for PotNet.

tion layer. For each edge, the ***Embedding*** layer is employed to map the Coulomb potentials and summations of Coulomb potentials, London dispersion potentials, and Pauli repulsion potentials to 256-dimensional embeddings by using 256 RBF kernels with centers from -4.0 to 4.0.

The Interaction block contains several ***Interaction*** layers. Each layer updates the feature vector of node i based on features of the neighboring nodes and potential embeddings of the connected edges. Particularly, for any of the neighboring node j of node i, the corresponding potential embeddings $\mathbf{e}^c_{ij}$, $\mathbf{s}^c_{ij}$, $\mathbf{s}^l_{ij}$, and $\mathbf{s}^p_{ij}$ for edge ij are all produced by the ***Embedding*** layer.

The Readout block contains an ***AvgPooling*** layer and another ***Linear*** layer. We first use the ***Avg-Pooling*** layer to aggregate features from all nodes in a graph and then use the ***Linear*** layer that maps the hidden dimension of 256 to the final output which is a scalar.